



3D micro scratch tests in combination with a comprehensive stress analysis – a new tool for the understanding of surface failures

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Agenda

- Motivation of micro scratch tests
- Determination of real tip geometry
- Analysis of repeated scratch tests at same position with increasing load
 - Detection of elastic plastic transition (first failure)
 - Detection of severe fracture
- Stress analysis
- Conclusions







It depends on the sharpness of the counterpart if coatings can be damaged or not.

This shall be analyzed in laboratory with scratch tests.







Scratch test with common Rockwell-C indenter, linearly increasing load 0– 100 N on soft ASA polymer substrate and thin hard layers (a) 0.52µm CrN and (b) 1.51µm TiN.

T. Sander, S. Tremmel, S. Wartzack Surf. Coat. Techn. 206 (2011) 1873-1878 Scratches evaluated in scratch test (CSM Revetest-RST) showing the influence of the nitriding temperature (520 °C (b) and 560 °C (c) respectively) on the crack behavior depending on the generated surface hardness. (a) shows scratch pattern on the untreated reference.

H. Paschke, M.Weber, P.Kaestner, G.Braeuer Surf. Coat. Techn. 205 (2010) 1465-1469

Conventional scratch tests produce coating failures but one can not understand why the failures occur







Failure modes for a standard scratch test with Rockwell C indenter



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Von Mises stress field in the surface of a 2µm thick hard coating on steel for a scratch test with tips of different radius.



The first failure of a conventional scratch test occurs normally in the substrate.







To understand the failure reasons one has to go down to the dimension of roughness and wear particles \rightarrow This requires high resolution.





Micro scratch test



| Parameter (typical) | Micro scratch test | Conventional scratch test |
|---------------------|------------------------|------------------------------|
| Indenter radius | 2 – 20 µm | 200 µm (Rockwell) |
| Normal force | 0,01 – 2 N | 1 – 100 N |
| Scratch length | 10 - 500 µm | 5000 µm |
| Scratch speed | 1-20 µm/s | 300 µm/s |
| Normal displacement | Available | Not available |
| Stress maximum | In coating / Interface | In substrate |
| Roughness influence | Low | High |
| Tip wear | Medium | High |





\approx 2µm DLC on steel, tip radius 7.2µm



Example for bad adhesion

The delamination can be seen in the curve by a step in the displacement curve and a minimum in the friction curve





UNAT – Universal Nanomechanical Tester







UNAT principle

Other principles



High lateral stiffness in scratch direction

Shaft bending in nm range is determined and corrected in the data

Low lateral stiffness in scratch direction

Shaft bending in μ m range prevent correct determination if tip position on surface





Determination of the real tip geometry by using fully elastic indentations into two different reference materials



Fully elastic load-displacement curves, measured on fused silica + sapphire







Area function of indenter S10-6, nominal radius 10µm, obtained from elastic measurements on fused silica + sapphire







Radius function of indenter S10-6, nominal radius 10 μ m, obtained from elastic measurements on fused silica + sapphire The effective radius at the outermost tip is only 7 μ m



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Investigation of two samples with **a:C:H:W** coatings (DLC)

Sample #60 = gradient coating Sample #62 = homogeneous coating thickness 6.5µm thickness 7.8µm



2 examples for an optimized layer structure for a certain roughness range and loading condition







Indentation modulus as function of depth, measured with dynamic QCSM method of ASMEC







Indentation hardness as function of depth, measured with dynamic QCSM method of ASMEC





Measurement sequence







Sample surface after three 350mN tests





Sample #62





Sample surface after three 800mN tests



Sample #60

Sample #62





Sample surface after two 1500mN tests



Sample #60

Sample #62







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Sample #60 350mN







Comparison of surface position before and after test Peak height is reduced

Zwick Roell





Comparison of surface position before and after test Peak height is further reduced







Peak height is further reduced





Test 1

800mN

#60



Plastic deformation is clearly visible







Comparison of surface position before and after test Plastic deformation starts at 207mN, position 75µm







Comparison of surface position before and after test Plastic deformation starts again at 263mN, position 97.5µm

Zwick Roell





Comparison of surface position before and after test Plastic deformation starts again at 367mN, position 137µm







Sample #60; comparison of Test 1-3 with 800mN





Test 1 0.00-0.100 0.10 350mN 0.20 0.075 Depth (hm) #62 0.050 0.50 0.60 0.025 0.70 #62 - Scratch depth #62 - Pre-scan depth #62 - Post-scan depth #62 - Friction 0.80 -0.000 50 100 150 200 250 300 Ó

Distance (µm)

Friction coefficient







Comparison of surface position before and after test Plastic deformation starts at 173mN, position 147.5µm







Comparison of surface position before and after test Plastic deformation starts again at 176mN







Comparison of surface position before and after test Plastic deformation starts at 177mN







Comparison of depth under load and remaining depth from 800mN tests on sample #60 and 62









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At present roughness information was only available in one direction \rightarrow Pseudo 3D

Stress analysis is done in the region of beginning plastic deformation



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 $168 \text{mN} \rightarrow 274 \text{mN}$



Comparison of stress profiles from different positions Plastic deformation starts at von Mises stress of 14.5 GPa





 $153mN \rightarrow 199mN$



Comparison of stress profiles from different positions The stress is not steadily increasing with force Plastic deformation starts at von Mises stress of 15.8 GPa



Sample #60





The tensile stress is considerably higher in regions with increasing surface profile



Failure type: fracture





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Comparison of depth under load and remaining depth from 1500mN tests on sample #60 and 62













Sample# 60, radial XX-stress just before fracture







Sample# 60, XZ shear stress just before fracture





Comparison of 5 successive scratch tests with constant load on sample #62 distance $300\mu m$, force 1000mN



Development of the depth under load from test to test

Development of the depth change from test to test (profile after-before test) The surface after the test is higher than before the test





Summary

| Sample | #60 gradient | #62 homogeneous |
|---------------------------------|--------------|-----------------|
| Hardness | 11.5 GPa | 14.5 GPa |
| Modulus at surface | 120 GPa | 150 GPa |
| Elastic-plastic transition | 207 mN | 173 mN |
| Max. von Mises stress | 14.5 GPa | 15.8 GPa |
| First fracture in test 2 | 1341 mN | 1180 mN |
| Max. tensile stress at fracture | 20 GPa | |
| Max. shear stress at fracture | 8 GPa | |
| | | |

The stress calculation did not consider the real 3D surface profile and the multilayer structure of the coating Therefore the stresses are overestimated here.





Conclusions

- In micro scratch tests it is possible to detect the elastic plastic transition and to follow a peak reduction in repeated tests at same position
- Roughness is reduced but not removed after plastic deformation in a scratch track
- The stress is not steadily increasing with increasing force but depends on surface profile
- At peaks the von Mises stress is considerably increased
- At increasing flanks of peaks the tensile stress is considerably increased
- ➤ Severe coating failure may occur after repeated loading of the same position → a single scratch my not be enough to evaluate the failure probability of coatings
- The gradient layer is more failure resistant than a homogeneous layer
- Detailed stress calculations help to understand coating failures and to optimize coatings. This requires an accurate determination of indenter radius.





Thank you for your attention !

