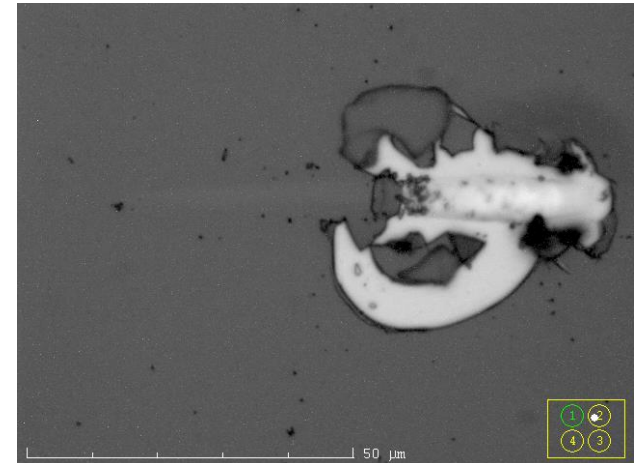
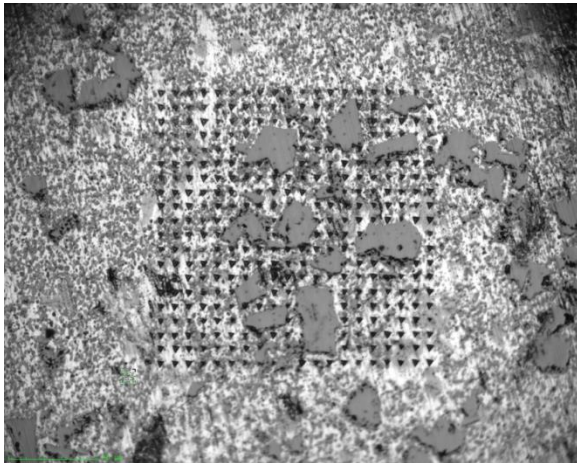


## Micromechanics of thin films

Dr. Thomas Chudoba

ASMEC Advanced Surface Mechanics GmbH



# Content

## 1) Motivation for mechanical testing

## 2) Mechanical material parameters

Hardness and modulus measurements by nanoindentation

Test methods for hardness and modulus

Determination of the area function

Yield strength measurements

Low cycle fatigue

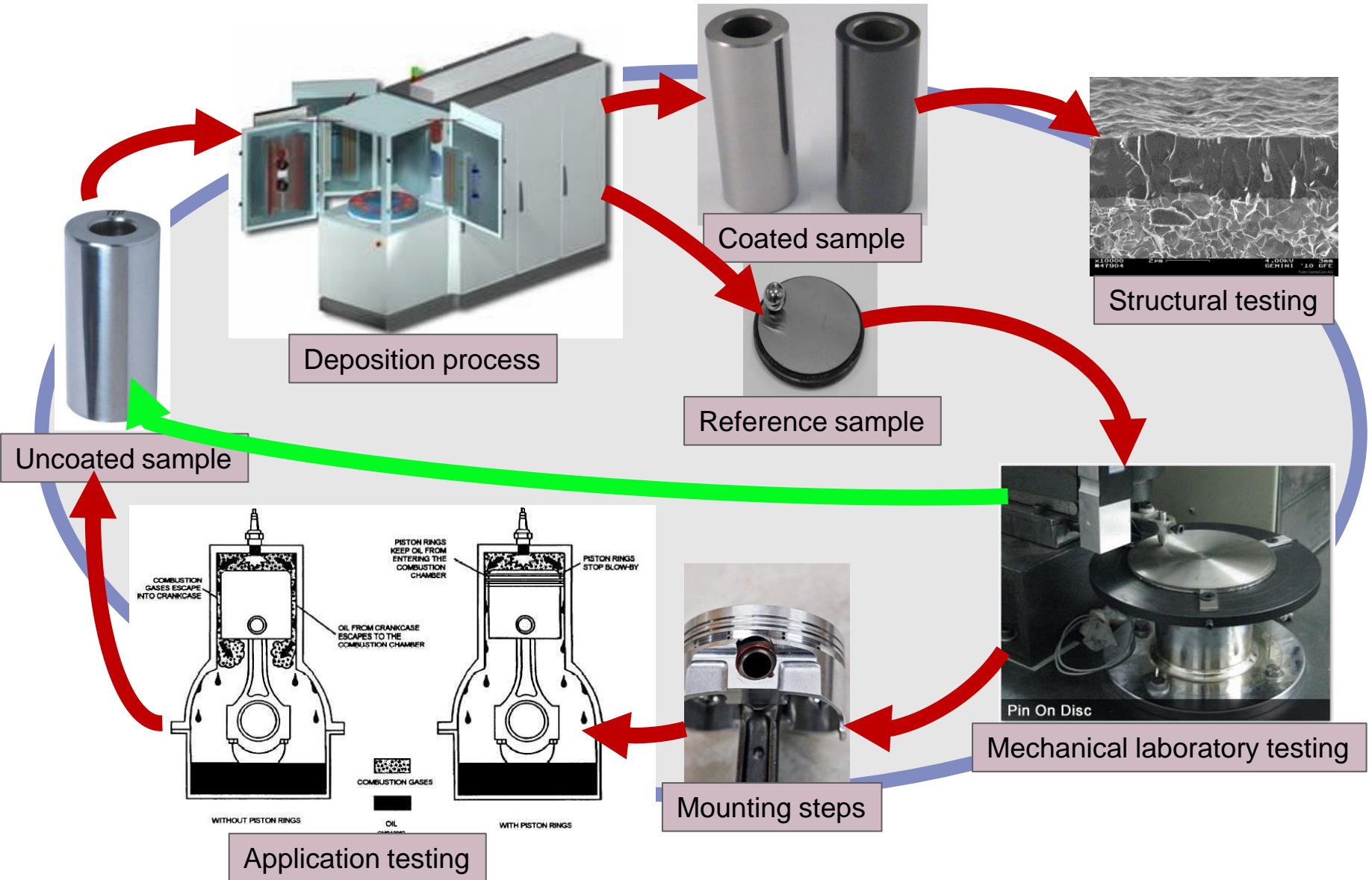
Micro wear test

Micro scratch test as an adhesion test

Mapping of mechanical properties

## 4) Conclusions

# 1) Motivation for mechanical testing



Trial and error development process in a multi parameter space

Example: Double layer coating

Testing of:

- 10 materials for each single layer (10 x 10)
- 10 process parameters per material
- 10 film thickness combinations

= 10 x 10 x 10 x 10 combinations

= **10000** samples

## Problems of surface mechanics for coated systems

The search for the optimum coating material is  
**time consuming and expensive.**

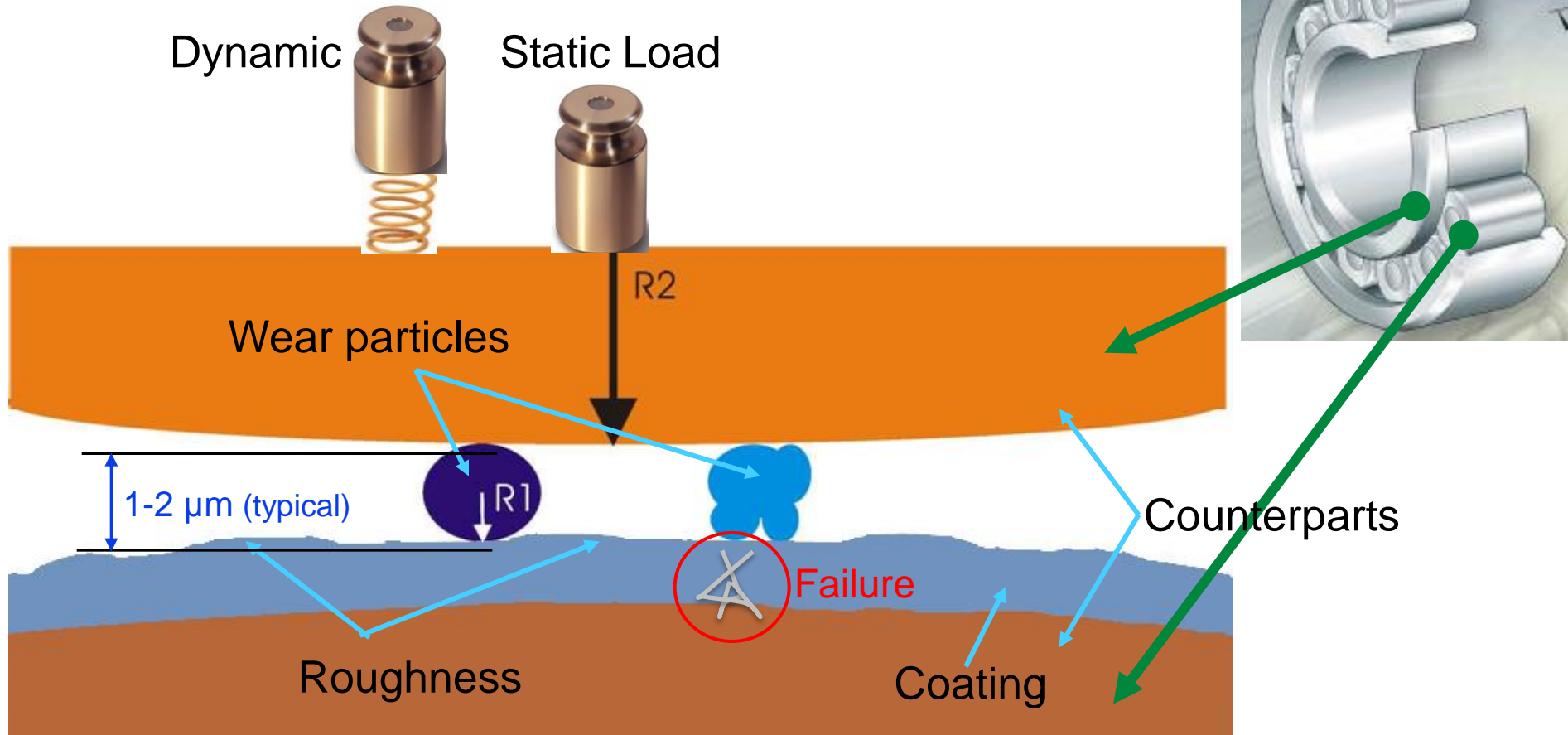
Mechanical parameters of thin coatings and small structures are  
**difficult to measure.**

Laboratory measurements and simulations often  
**differ from the conditions in a real application.**

## Steps for a reduction of the development effort

- 1) Identify the most critical conditions in the application, which cause failures
- 2) Identify the most critical mechanical (physical) material parameters
- 3) Combine tests with a higher level of modelling and calculations

## Analysis of characteristic loading conditions



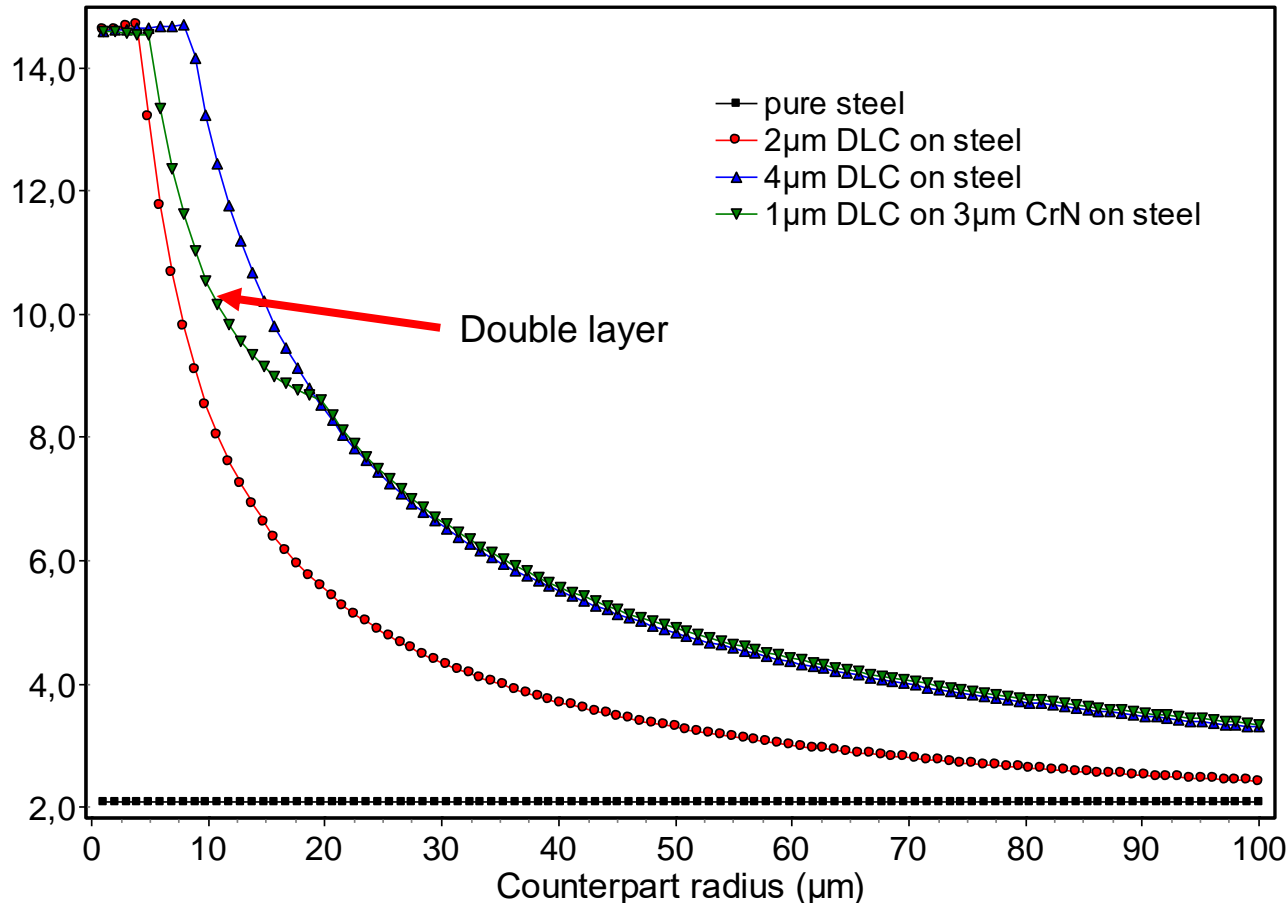
To understand the failure reasons one has to go down to the dimension of roughness and wear particles → This requires high resolution.



## Example for modelling

Load carrying capacity of different coating systems on steel

Load carrying capacity



Input parameter required:  
 E – Young's modulus  
 $\nu$  – Poisson's ratio  
 Y – Yield strength  
 (T – Tensile strength)

	Steel	DLC	CrN	
E	200	270	300	GPa
$\nu$	0.3	0.2	0.25	
Y	2	15	8	GPa

## 2) Mechanical material parameters

Mechanical behavior	Parameter	Test methods for coatings
Elastic behavior	Young's modulus, Poisson's ratio	Nanoindentation+LFU, ultrasonic surface waves, AFAM, impulse excitation technique
Plastic behavior	Hardness, Yield strength, Stress-strain-curve, hardening exponent	Nanoindentation+LFU
Brittle behavior	Fracture toughness, tensile strength	Nanoindentation+LFU, 4-point bending
Time dependent behavior	Creep, fatigue resistance, strain rate dependence	creep test, impact test, cavitation test, fatigue test (cyclic contact loading)
Frictional and wear behavior	Friction coefficient (not intrinsic) Wear coefficient (not intrinsic) Stribeck curve (not intrinsic)	Friction test, wear test, nanoindentation+LFU
Adhesion	Adhesive strength Scratch resistance (not intrinsic)	Scratch test (LFU), peel test, centrifuge test, cavitation test, Rockwell test

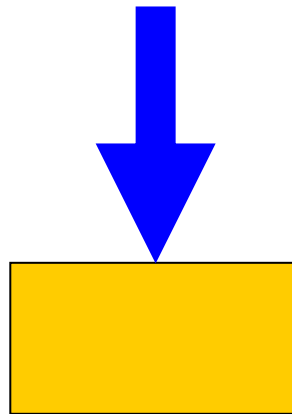
For all parameter it also has to be considered:

**LFU** = Lateral Force Unit

- Internal stress
- Dependency on temperature
- Dependency on sample homogeneity: failure density, gradients, thickness constancy, roughness

## Closer to application conditions

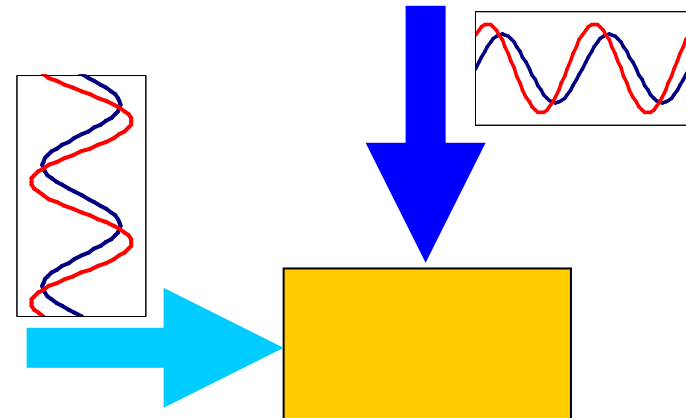
### Default Nanoindenter



1 degree of freedom:

- Normal load-displacement-curve

### Universal nanomechanical tester



4 degrees of freedom:

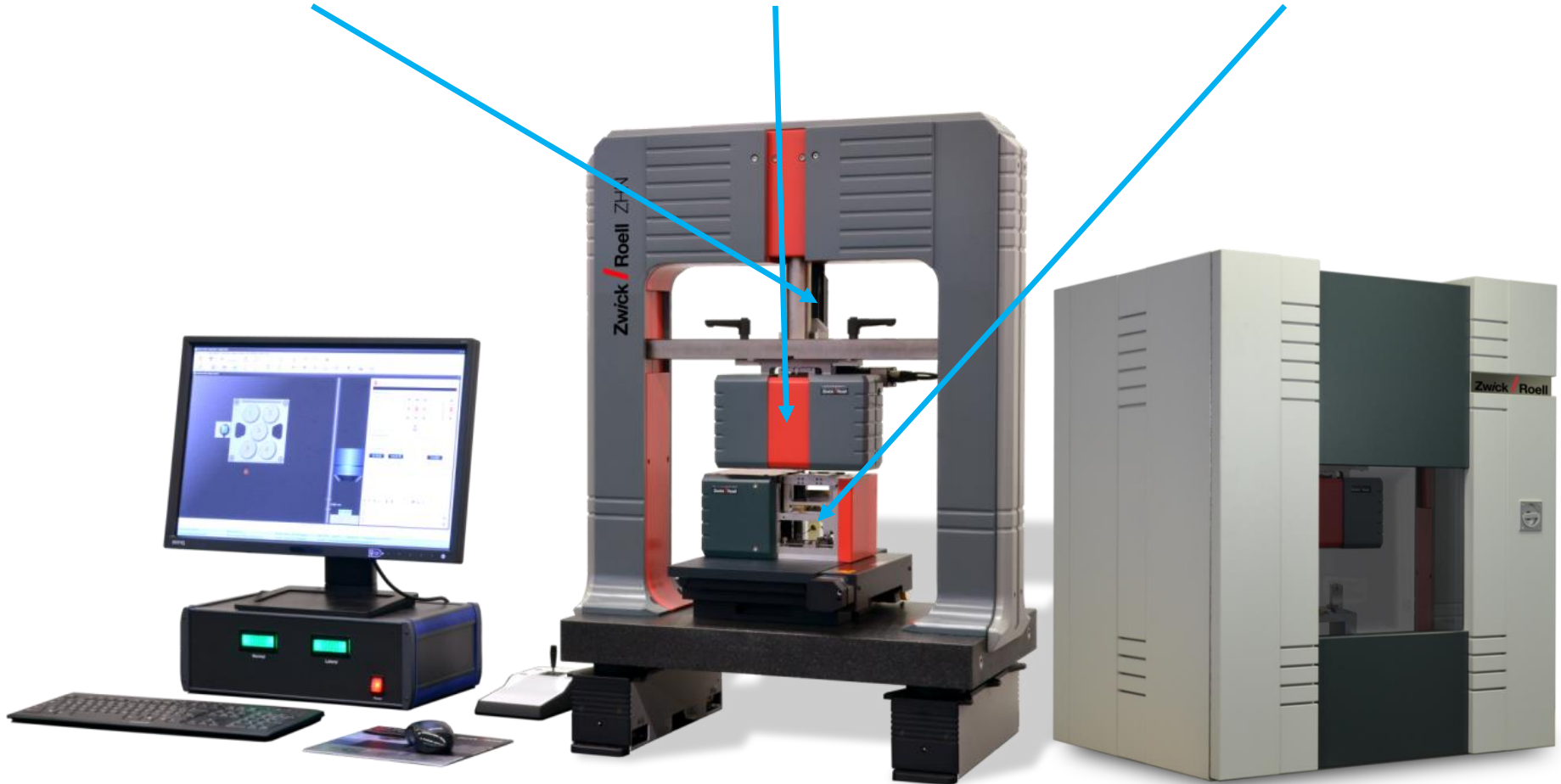
- Normal load-displacement-curve
- Lateral load-displacement-curve
- Oscillation normal (dynamic mode)
- Oscillation lateral (dynamic mode)

## ZHN – Zwick Universal Nanomechanical Tester

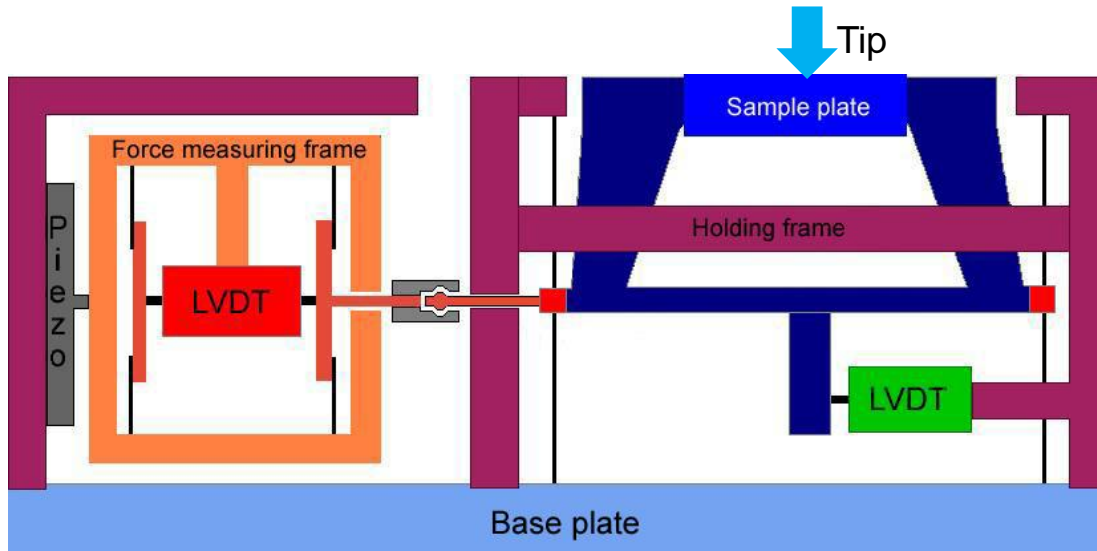
Optics behind the head

Normal force unit NFU

Lateral force unit LFU

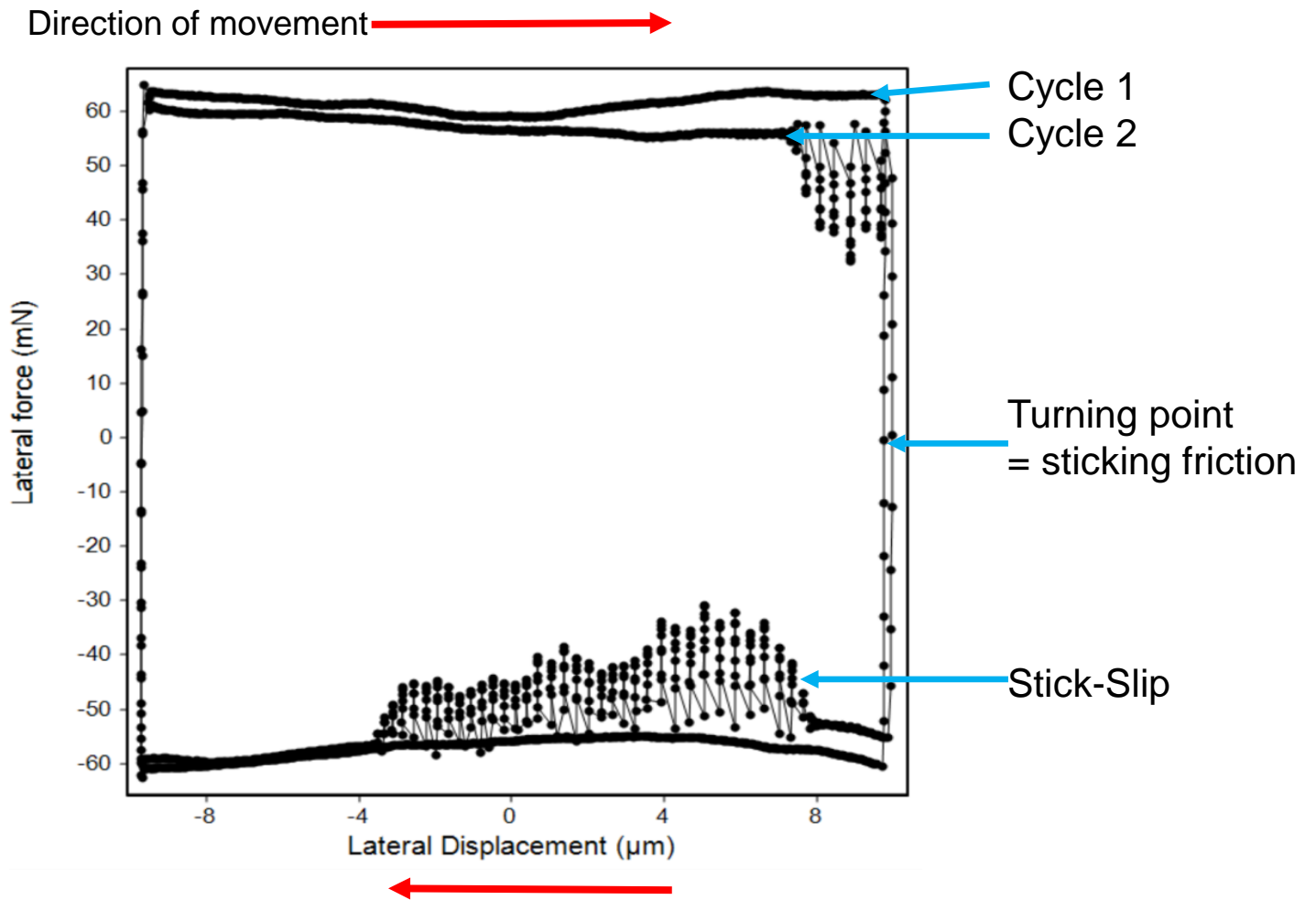


## Lateral force unit – the additional component



- Nanometer resolution like in normal direction
- The sample is moved laterally (not the tip)
- High stiffness in normal direction
- No height change during lateral movement
- Force generation independent on movement

- A force can be applied and measured without any movement of tip or sample
- No rolling motion of the tip due to bending of the indenter shaft
- Transition sticking- sliding friction highly resolved



Example: Diamond on diamond in water

# Hardness and modulus measurements by nanoindentation



## Depth sensing indentation = instrumented indentation = IIT

Standardized since **2002**: **DIN EN ISO 14577 Part 1-3**,  
**2007**: **Part 4: Coatings**

### Metallic materials - Instrumented indentation test for hardness and materials parameters

Macro-range: Force  $2 \text{ N} < F < 30.000 \text{ N}$

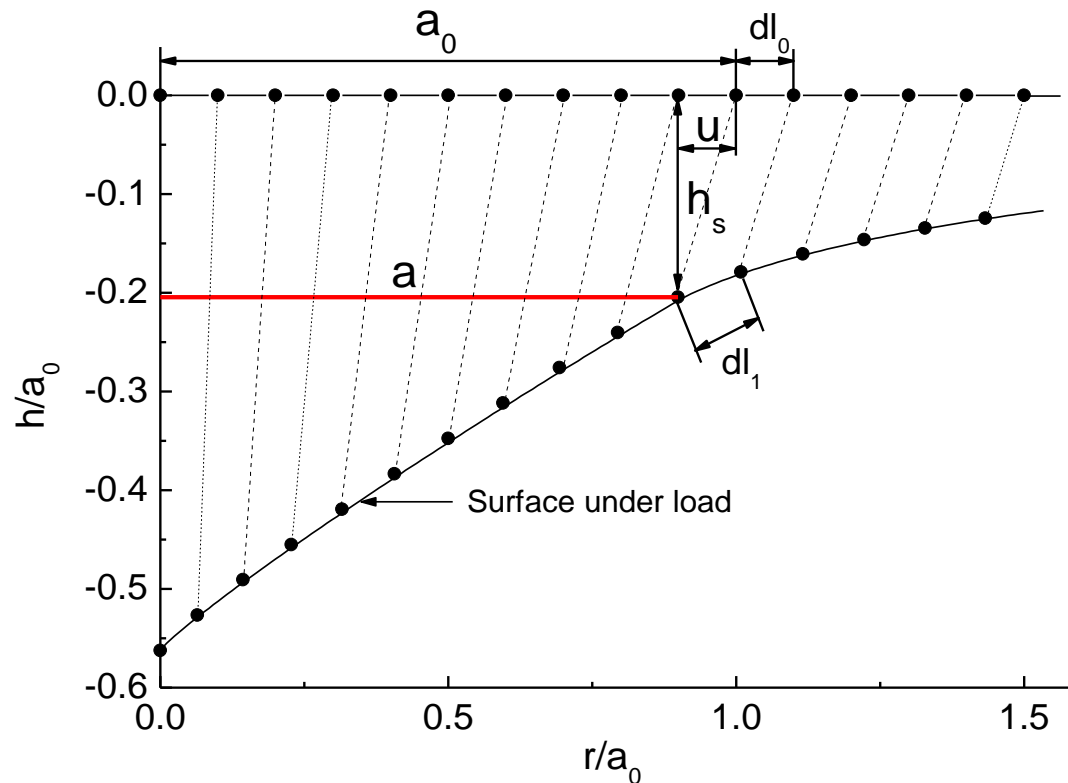
Micro-range: Force  $F < 2 \text{ N}$

Nano-range: Depth  $h < 200 \text{ nm}$

**Indenters:** **Pyramids** Vickers (4 edges)  
Berkovich (3 edges)

## Additional correction: radial displacement correction

Included in the new revision of the standard ISO 14577



Example for influence:

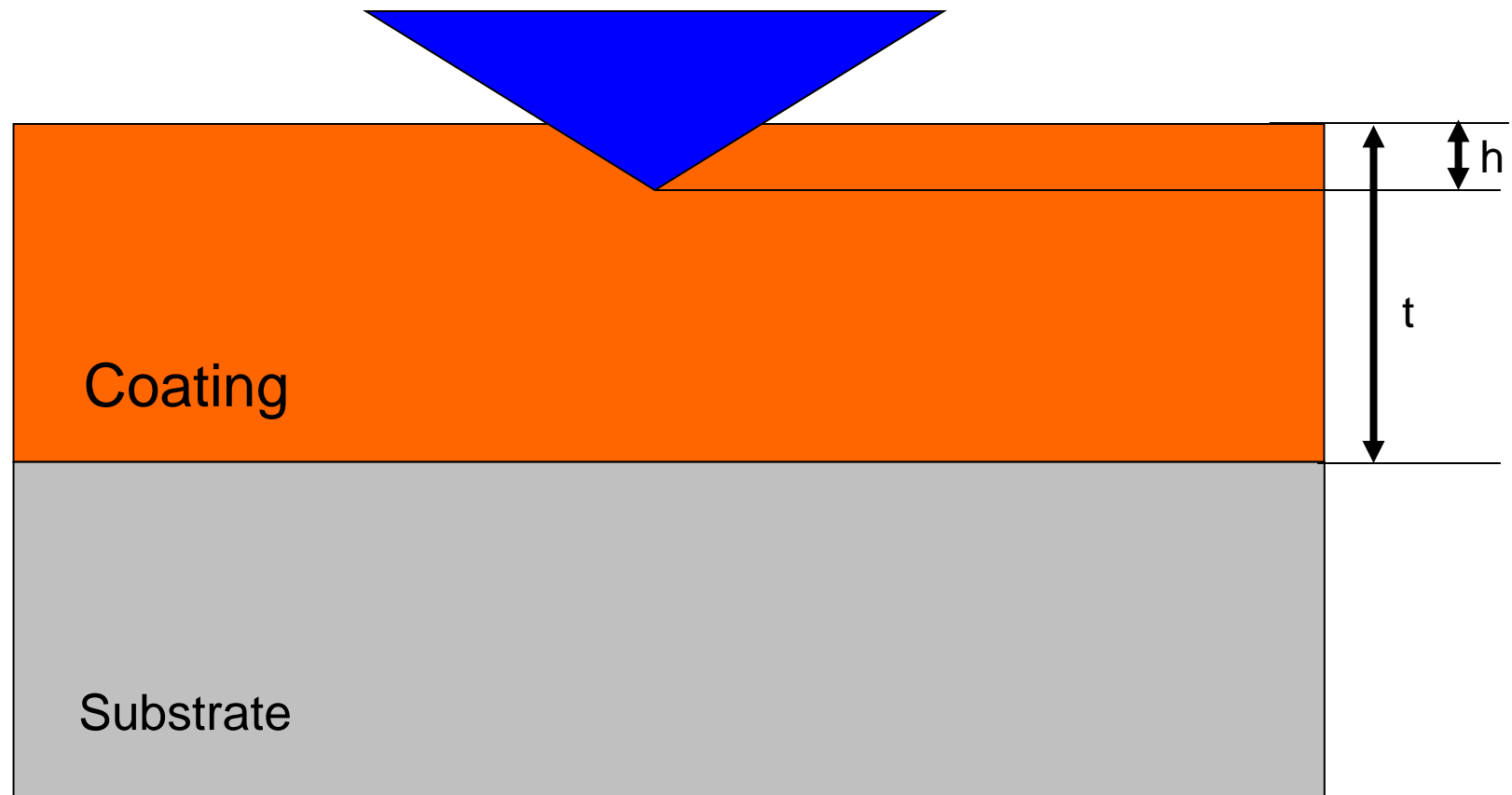
	with	without correction
H:	30 GPa	→ 31,5 GPa
E:	400 GPa	→ 413 GPa

$$u_r = \frac{(1-2\nu)(1+\nu)}{2} \frac{F}{E \cdot a^2} \cos(\arctan(\frac{h_0}{a}))$$

T. Chudoba, N. M. Jennett, Higher accuracy analysis of instrumented indentation data obtained with pointed indenters, J. Phys. D: Appl. Phys. 41 (2008)

## Upper depth limit for the hardness of coatings

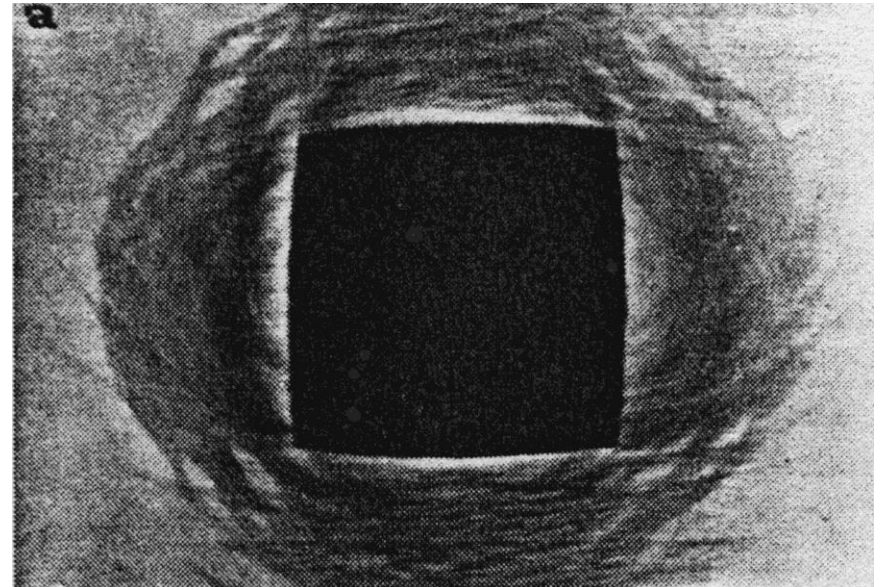
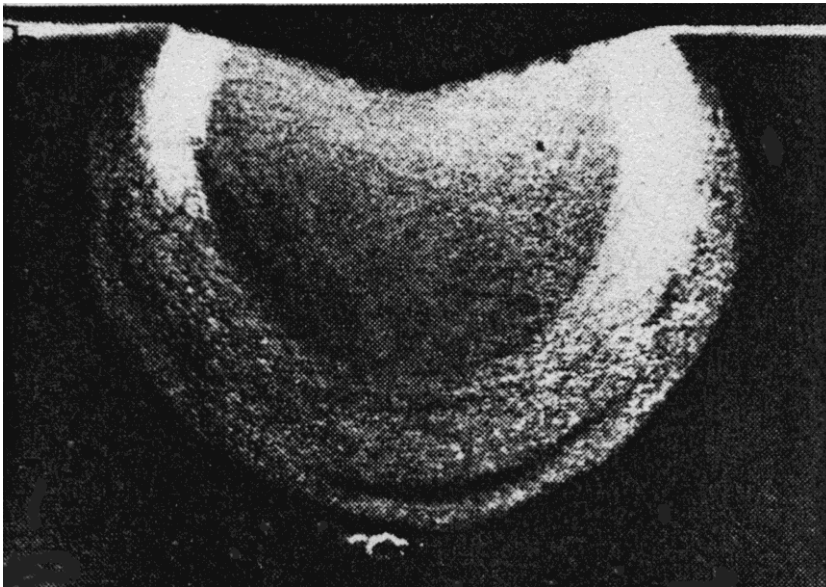
One tenth rule  $h < \frac{t}{10}$



## Reason for depth limit

The plastic zone is much larger and deeper than the indent.

Therefore the information in the load-displacement curve comes from a depth of up to 10 times the indentation depth.

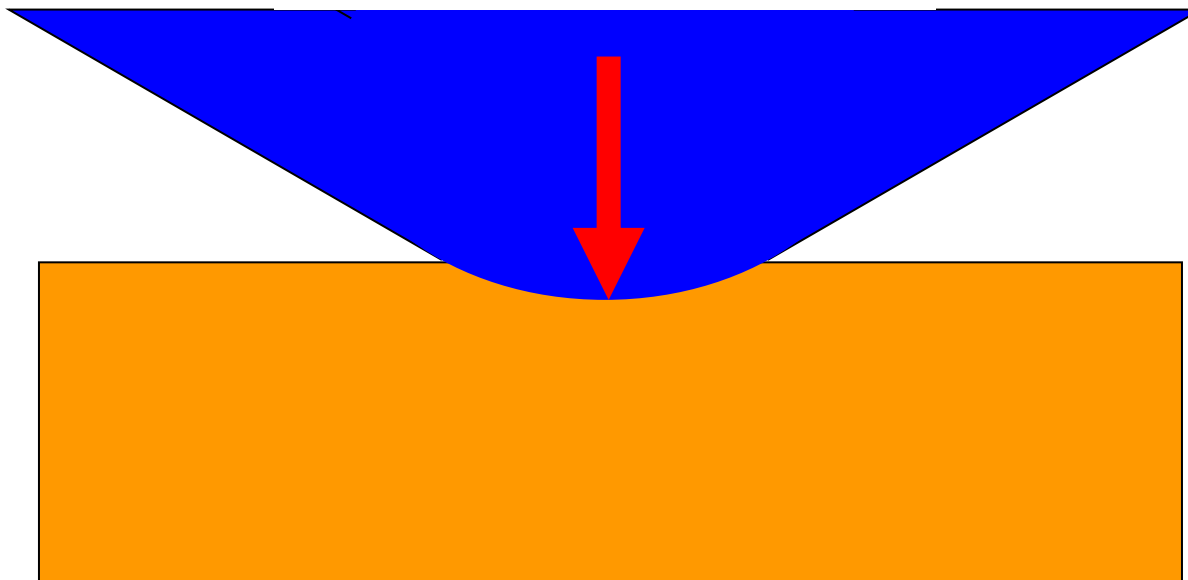


Presentation of the plastic zone in steel using a special etching technique

## Lower depth limit for hardness tests due to tip rounding

Minimum indentation depth for comparable hardness results:  
20% of tip radius

A Berkovich tip has a typical tip radius between 100 – 300 nm



Minimum depth for  
hardness tests:

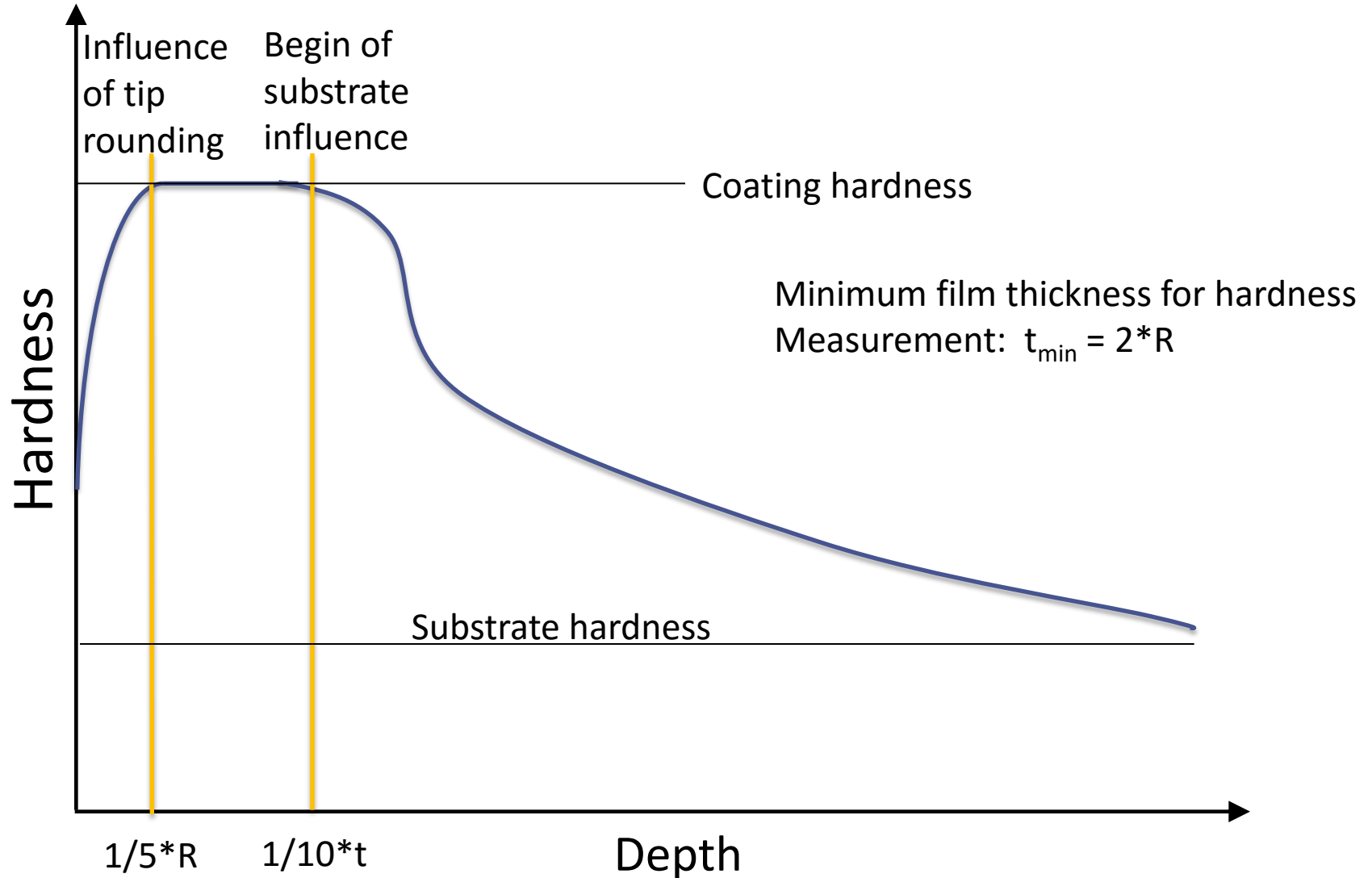
$$h_{\min} = 0.2 \cdot R$$

Minimum film thickness for **hardness** measurements  
in dependence on tip radius (not for modulus)

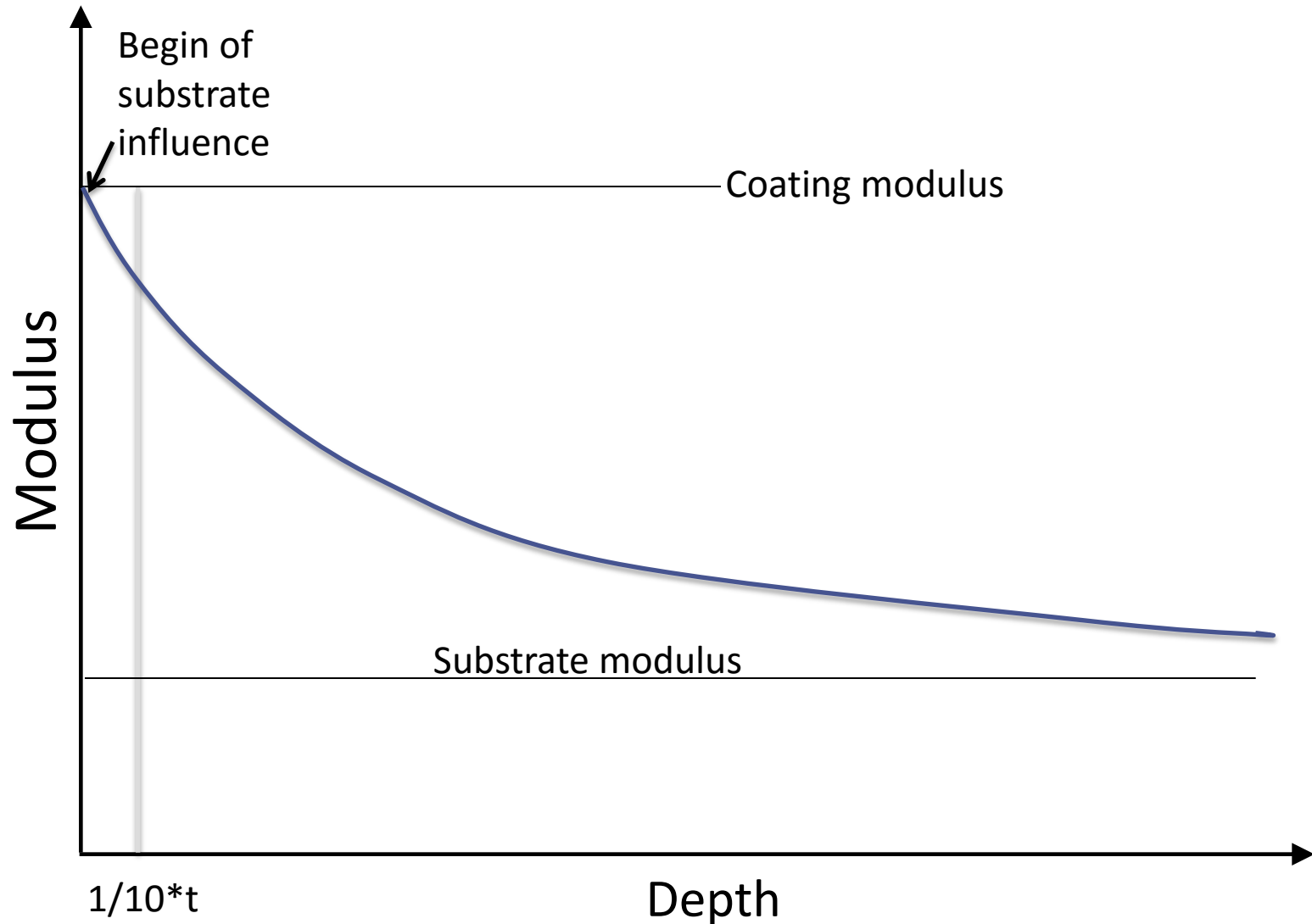
Tip radius (nm)	Minimum film thickness (nm)
50	100
75	150
100	200
150	300
200	400
250	500
300	600

typical

## Substrate influence on hardness



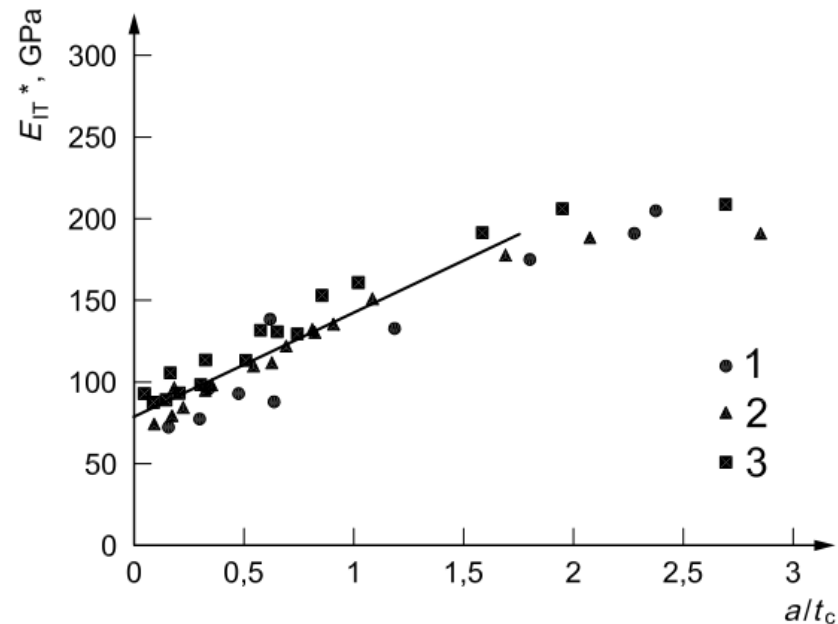
# Substrate influence on modulus





## ISO 14577 Part 4

In the case of soft/ductile coatings, indentation force or displacement and indenter geometry shall be chosen such that data shall be obtained in the region where  $alt_c < 1,5$ . The plane strain indentation modulus of the coating  $E_C^*$  is obtained by taking a series of measurements at different indentation depths and extrapolating a linear fit to plane strain indentation modulus vs.  $alt_c$  to zero, see Figure 4.

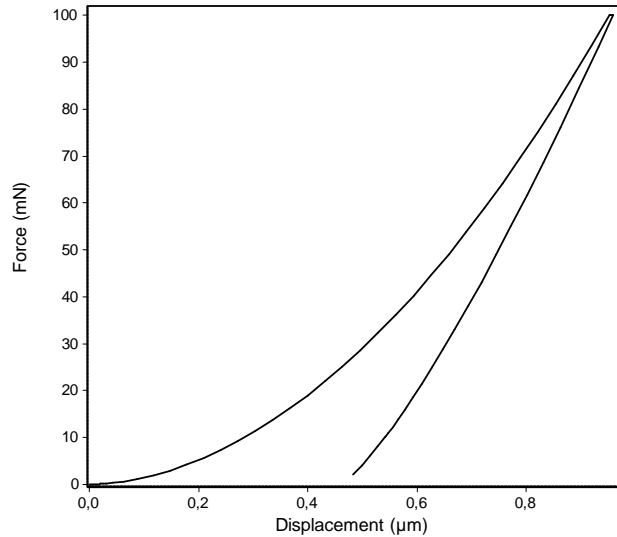


### Key

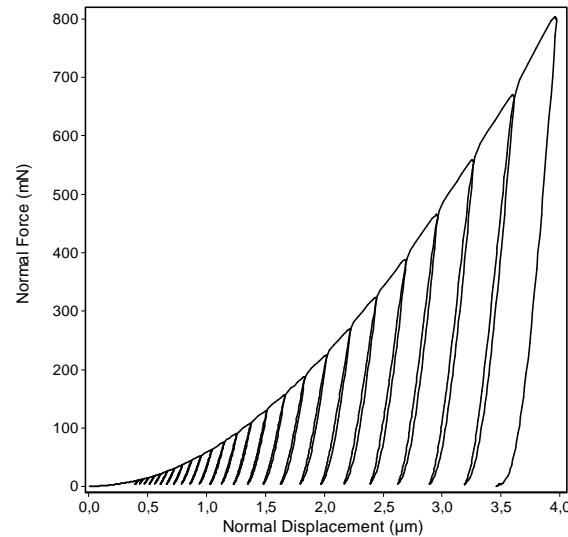
- 1 spherical indenter
- 2 Berkovich indenter
- 3 Vickers indenter

## Test methods for hardness and modulus

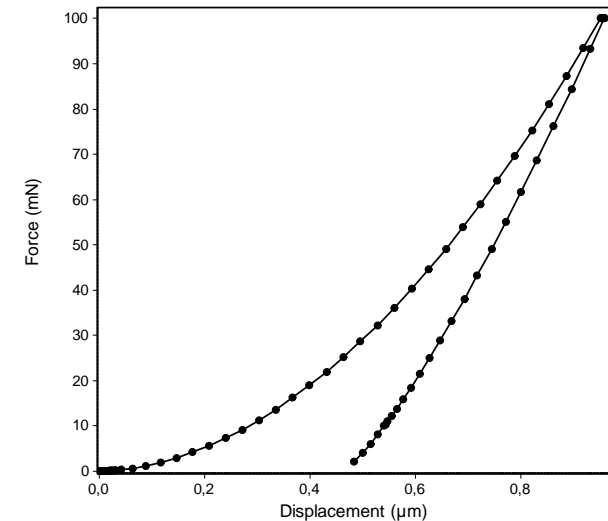
## Single test



## Cyclic test



## Dynamic test (QCSM)

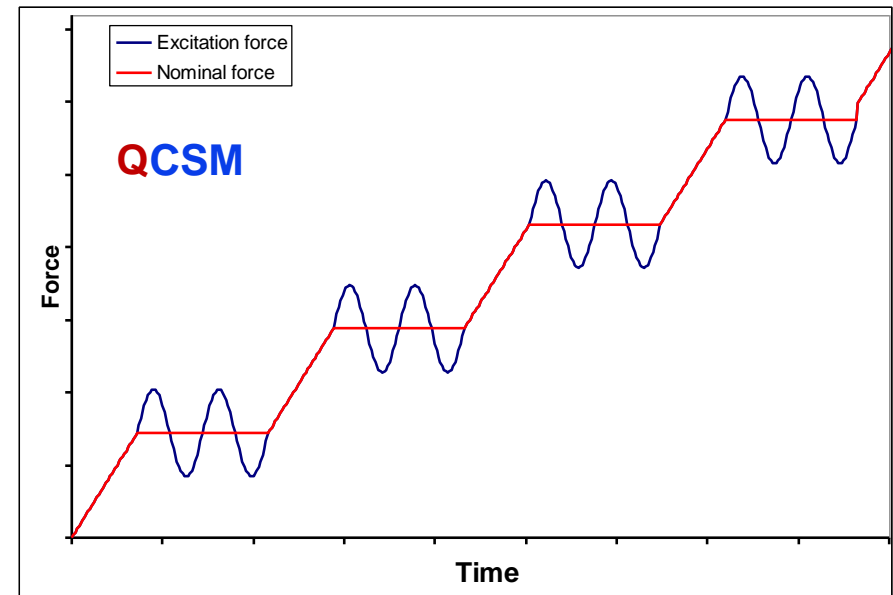
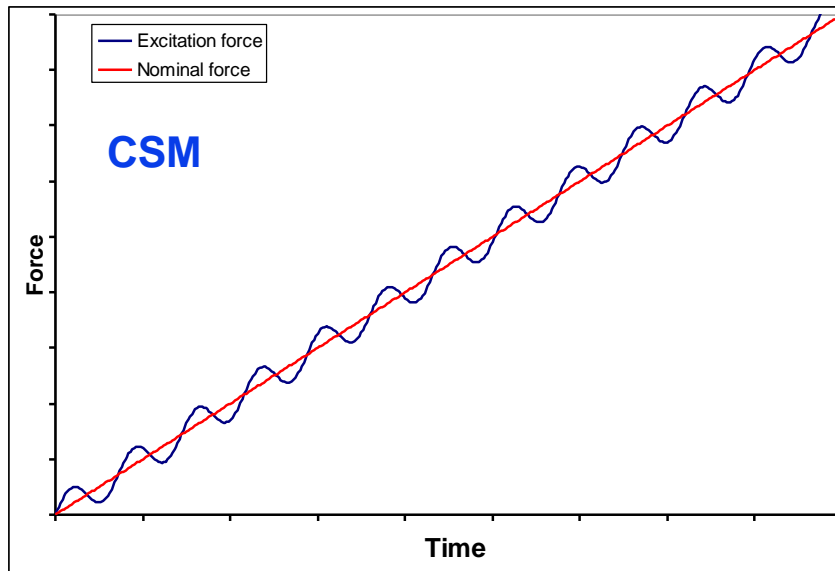


1 Result for maximum depth  
Typical test time **20s**  
Medium accuracy  
Results after creep

About 20 results over depth  
Typical Test time **250s**  
Lowest accuracy  
Results after creep

About 35 results over depth  
Typical test time **125s**  
Highest accuracy  
Results during creep

## Continuous and quasi continuous stiffness measurement

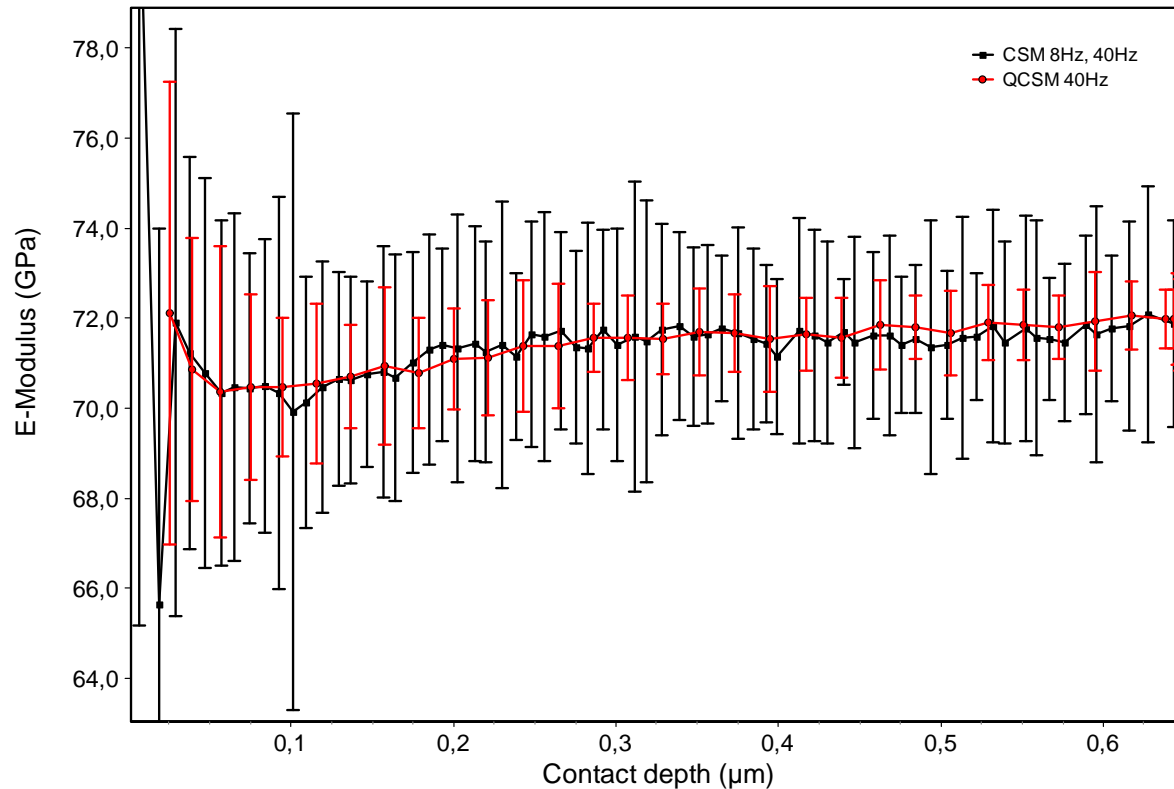


**QCSM:** sinusoidal oscillation only on (0,5 – 3s) when static force is constant.  
First 20% of data per force step are neglected to reduce creep influence.

### Advantages:

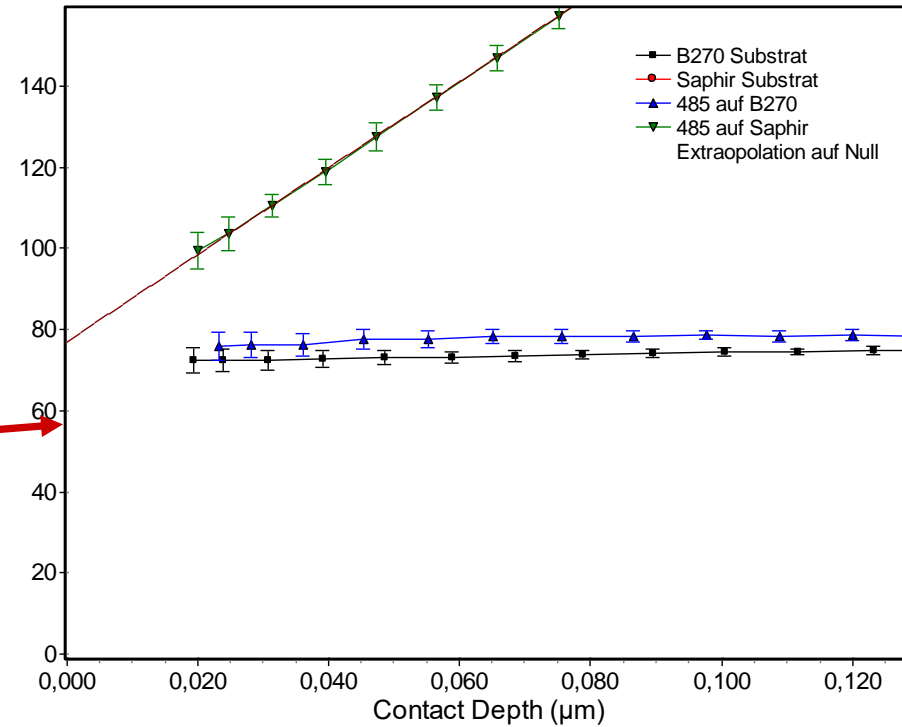
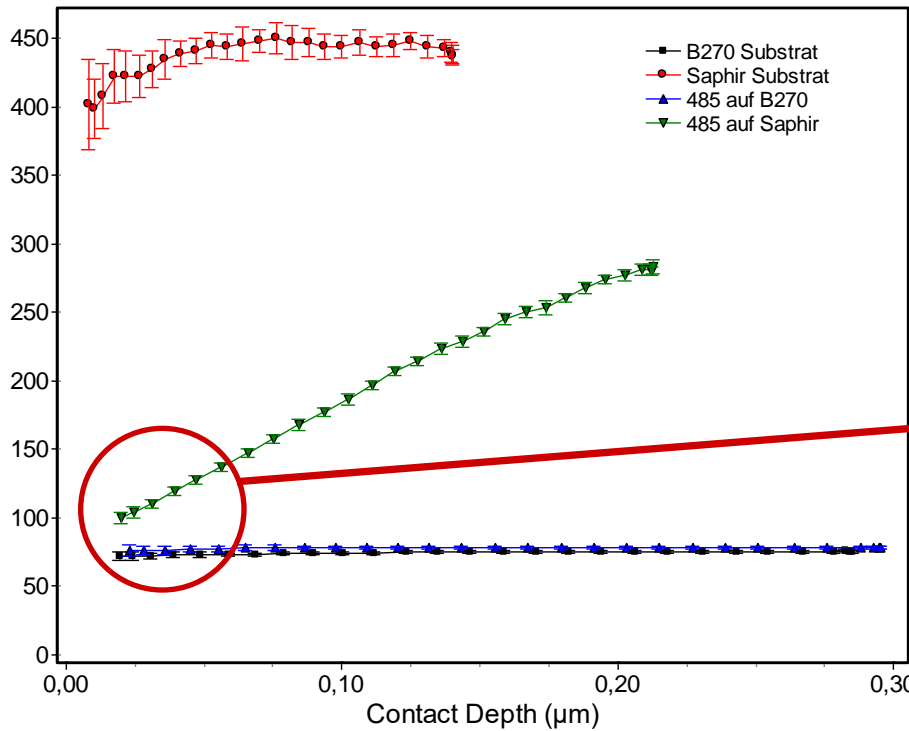
- Averaging of several oscillations at same conditions possible without problems → more accurate.
- Creep influence is reduced.
- Static forces are well defined for every oscillation result.

## Comparison of accuracy between CSM and QCSM



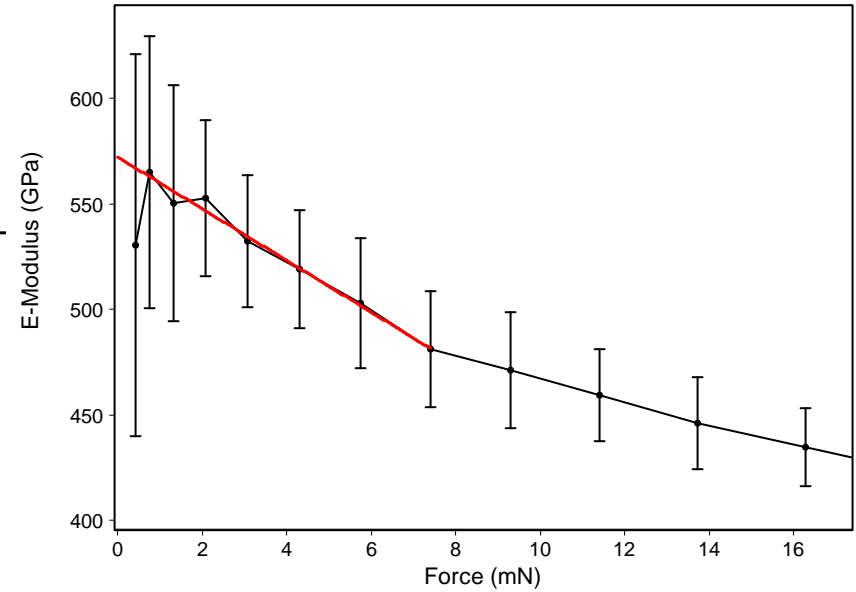
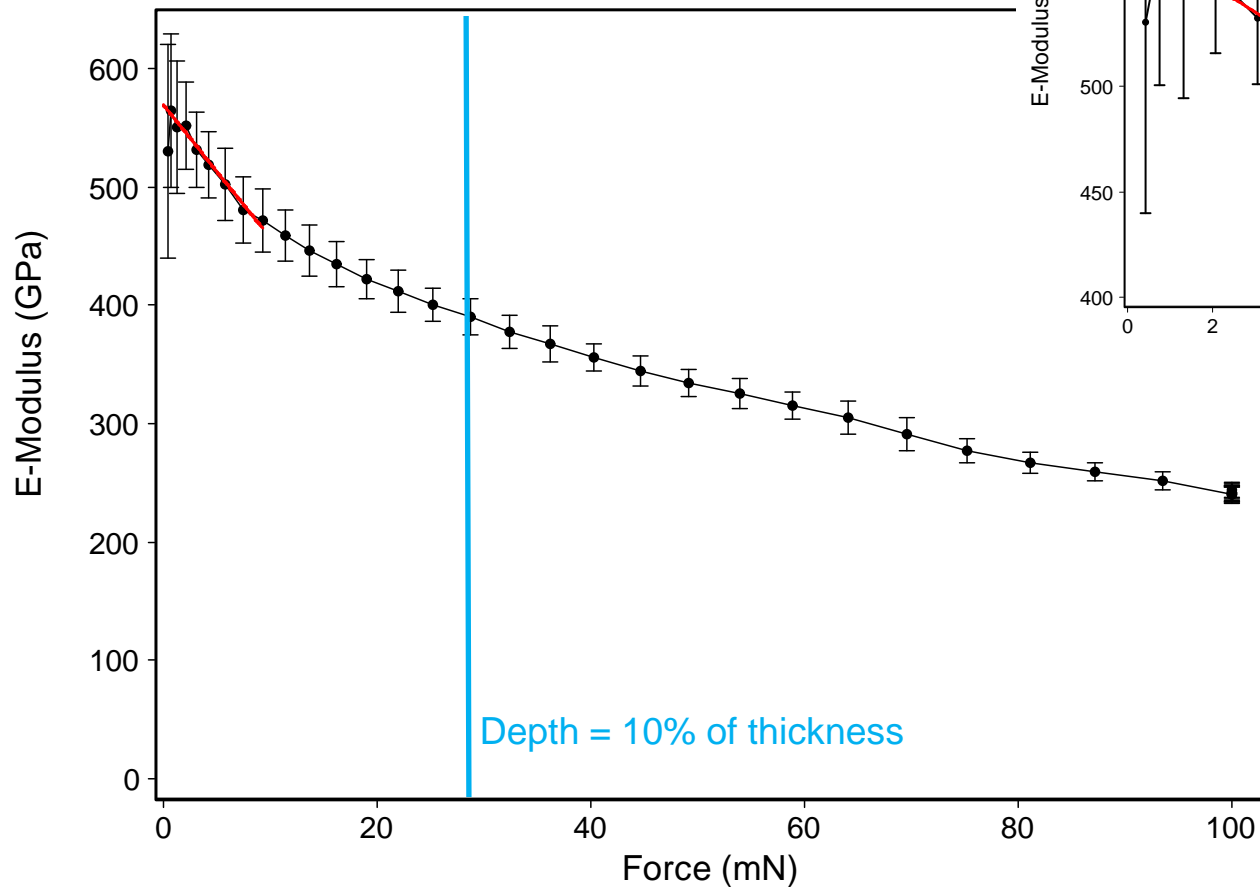
Error bars give statistical error for average of 6 measurements  
Error of QCSM tests < 50% of error of CSM tests.

Example: equal **260nm** SiO<sub>2</sub> coatings on glass and sapphire

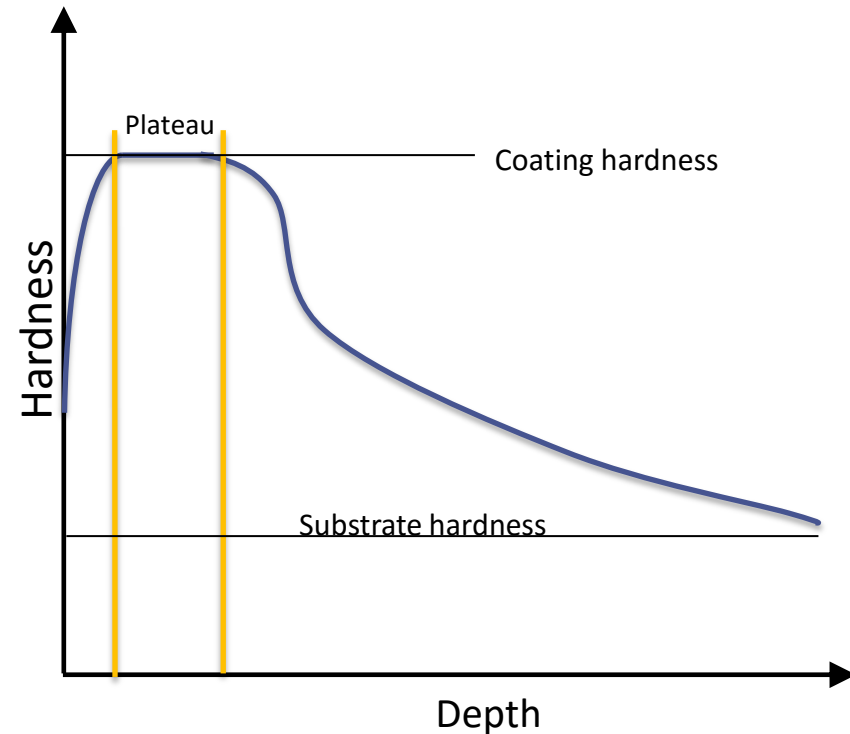
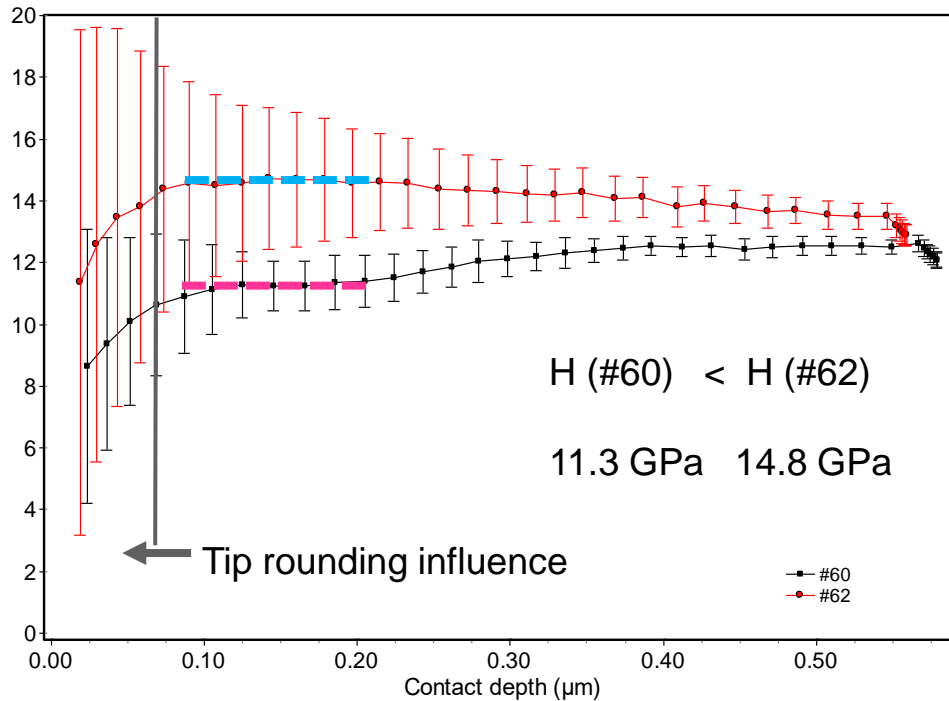


Maximum force 18mN; first point at (20 nm; 0,24 mN).

## Example: 1,26 $\mu$ m ta-C on steel

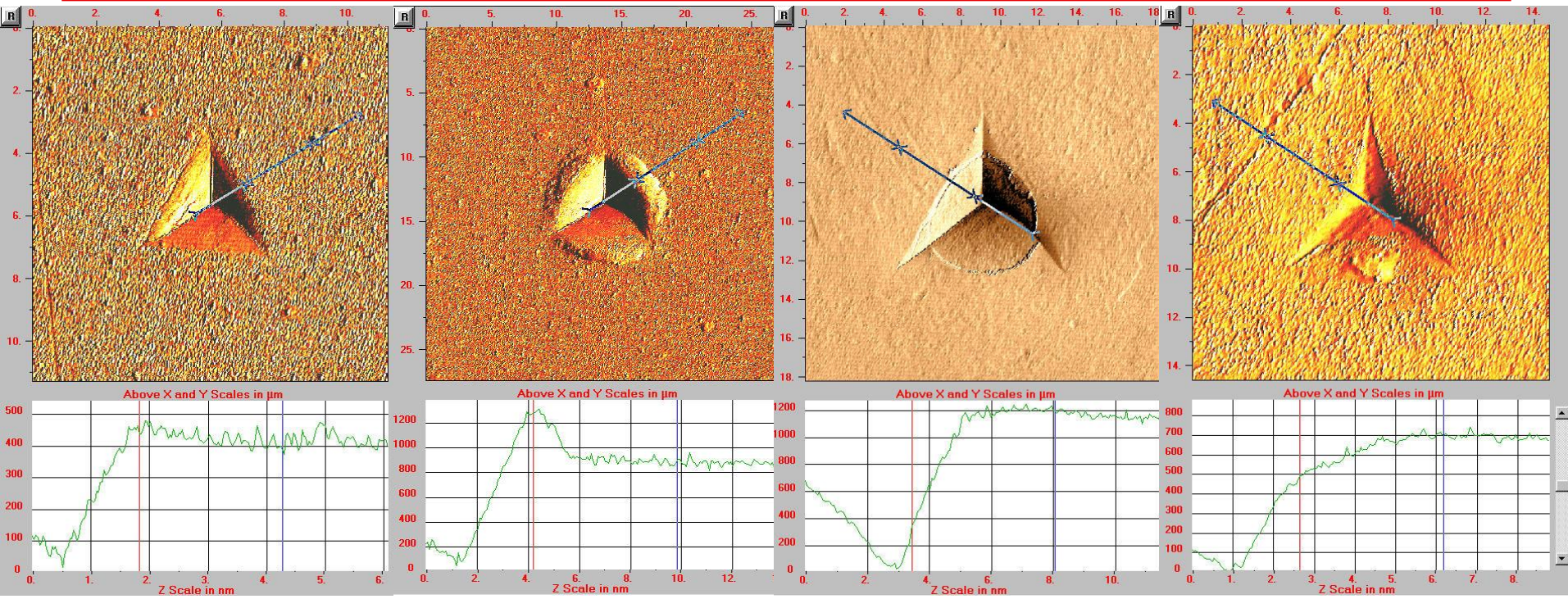


Hardness: finding a plateau without substrate influence



Indentation hardness of two different **a:C:H:W** coatings (DLC) as function of depth, measured with dynamic QCSM method





## normal

Al on BK7 glass

0,4 μm / 1,1 μm

Load: 10 mN

## pile-up

Al on BK7 glass

0,9 μm / 1,1 μm

50 mN

## cracks

Al<sub>2</sub>O<sub>3</sub> on Nickel

1,2 μm / 0,9 μm

100 mN

## sink-in

Al<sub>2</sub>O<sub>3</sub> on Nickel

0,7 μm / 2 μm

100 mN

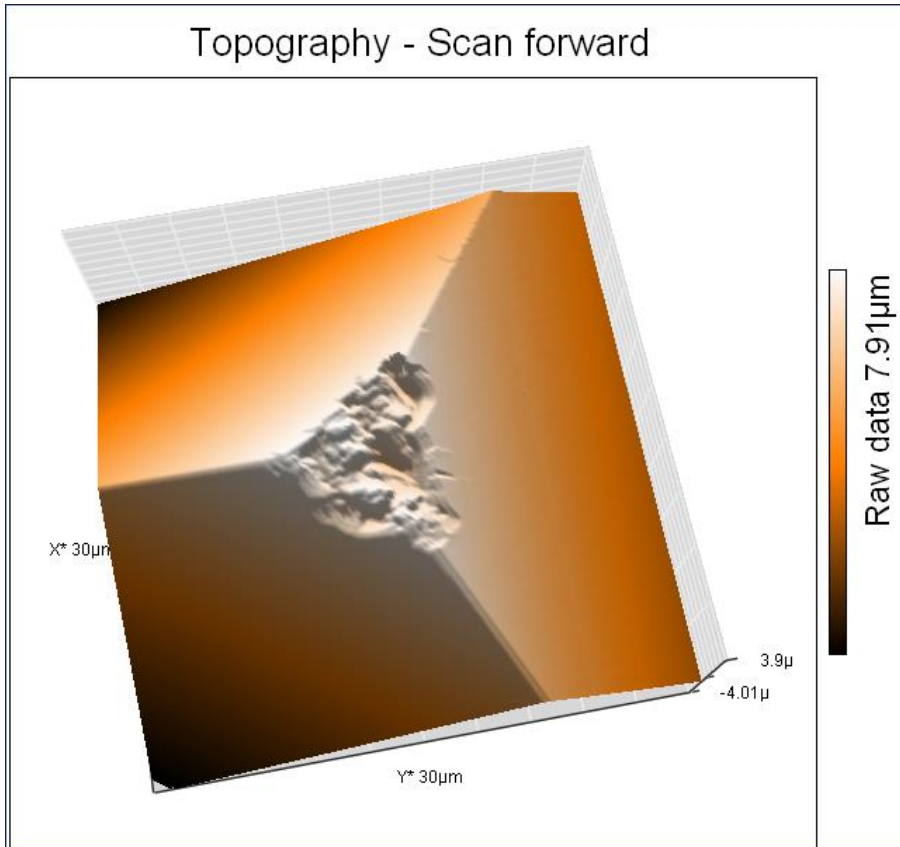
green = depth / film thickness

## Determination of area function

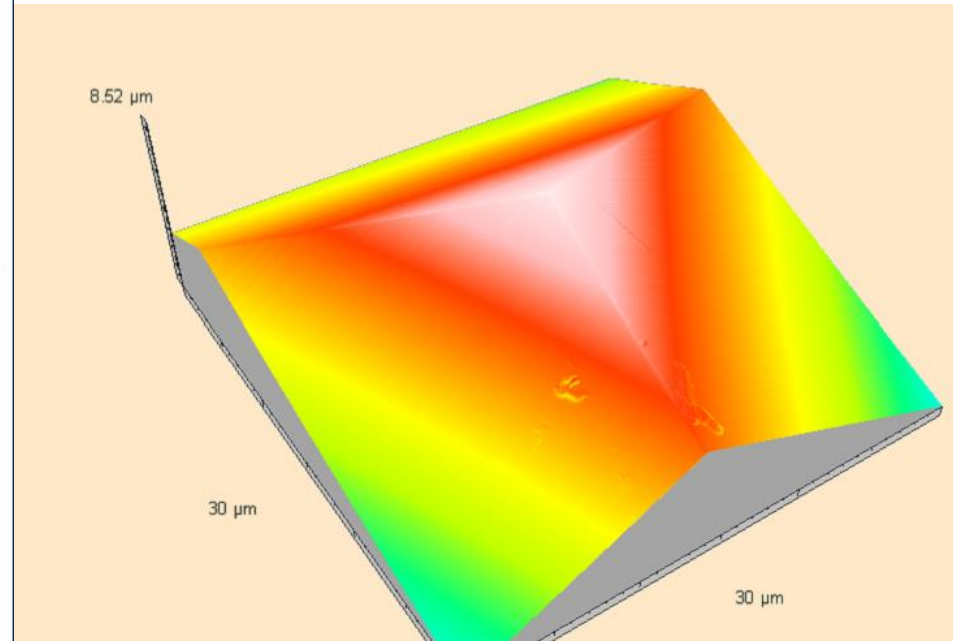
In the low depth range ( $<6\mu\text{m}$ ) an accurate determination of the real tip shape (area function) is the key for correct hardness and modulus measurements.

## Determination of area function

Direct method: AFM scan of the tips



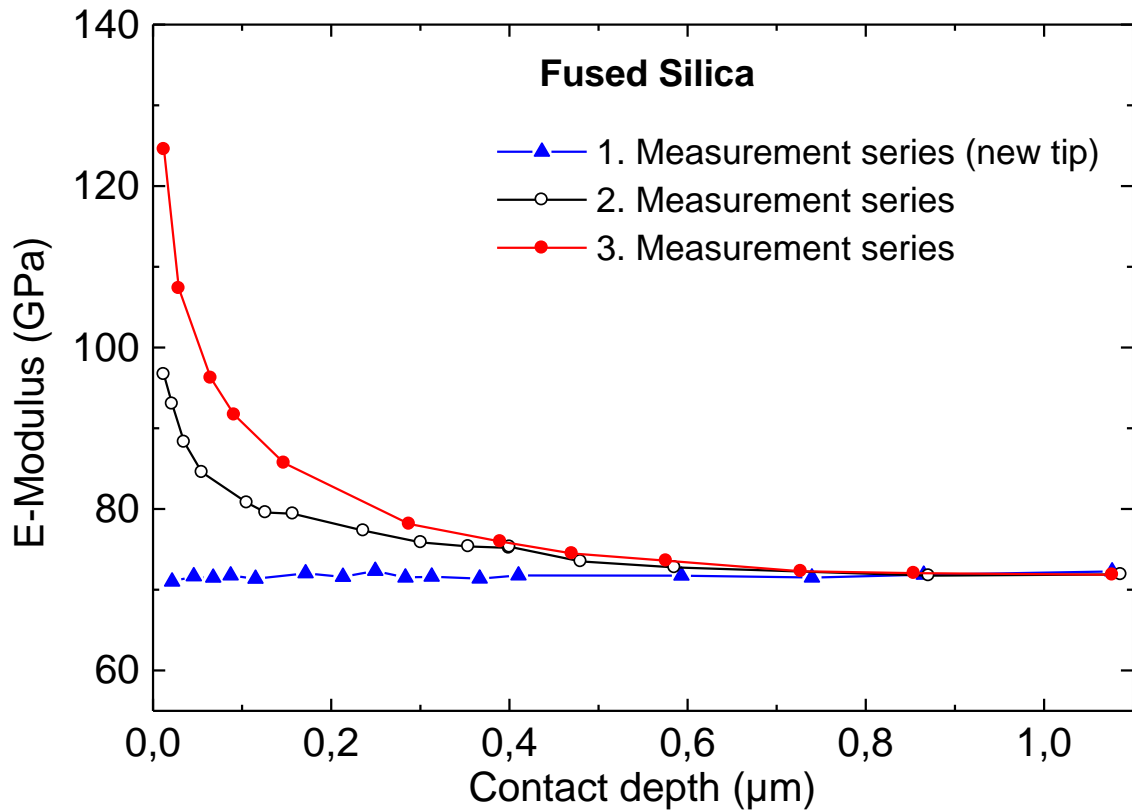
Left: broken diamond tip



Right: new tip without defects

## Error due to incorrect area function in dependence on depth

### Example: fused silica



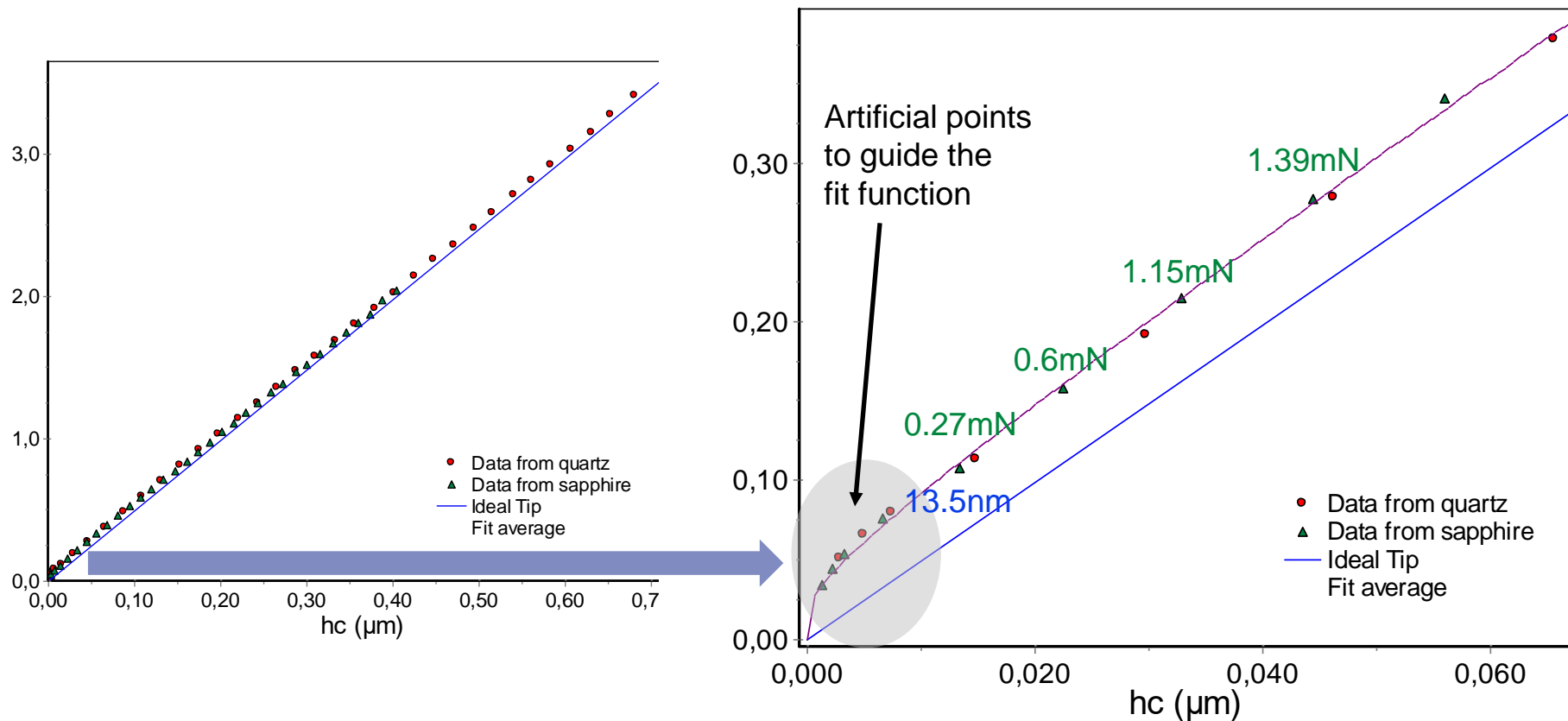
Tip radius:

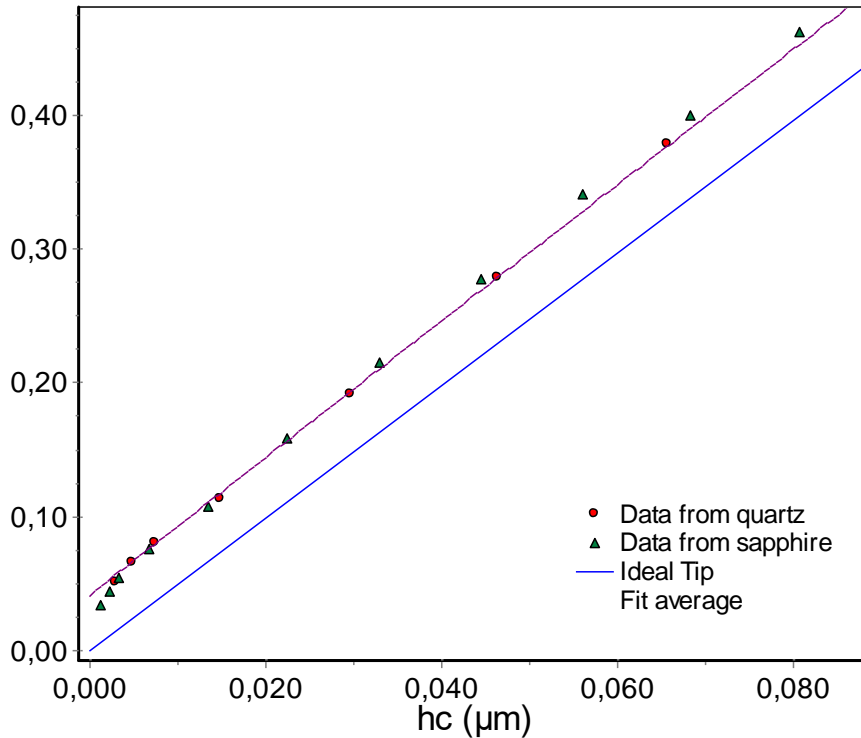
- 1.  $\approx 150$  nm (calibrated fresh tip)
- 2.  $\approx 350$  nm (no new calibration)
- 3.  $\approx 700$  nm (no new calibration)

## Determination of area function

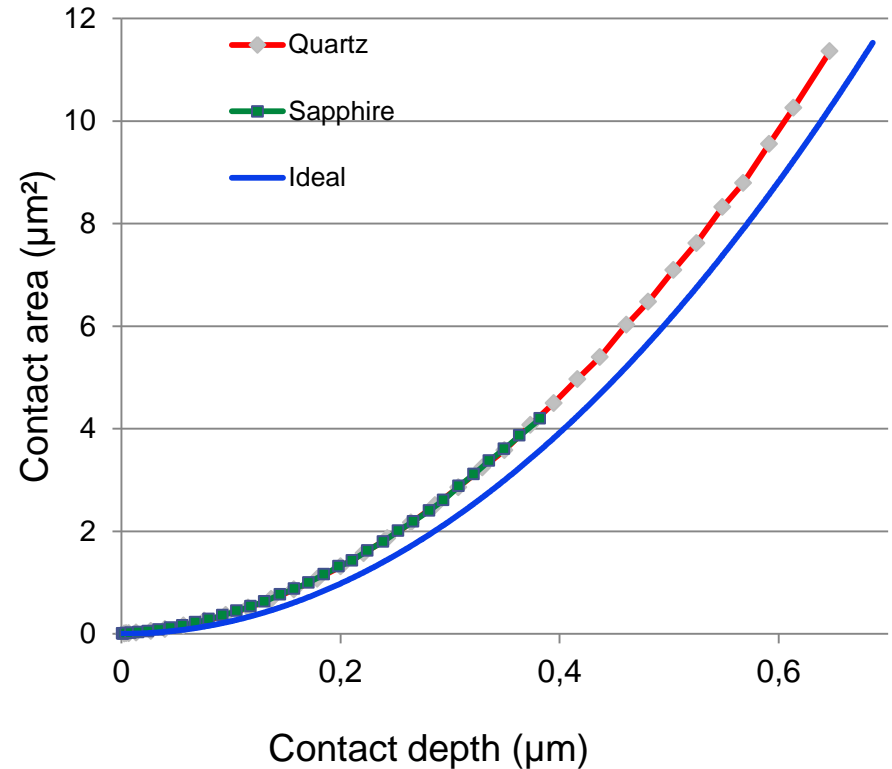
**Direct method:** Measurement with metrological AFM

**Indirect method:** Indentation measurements on homogeneous reference materials





Inadequate fit function with insufficient term number

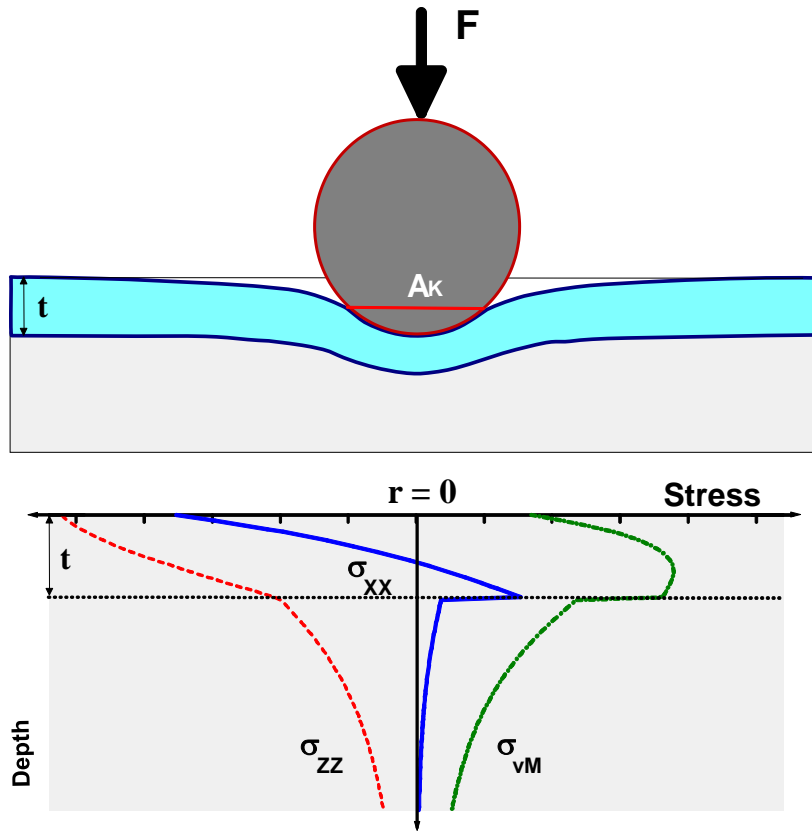


A quadratic presentation (area over depth) allows only a bad estimation of the tip quality

## **Alternative modulus test methods**

**especially for ultra thin coatings**

## Elastic indentation with spherical indenter and compensation of substrate influence



### Requirements

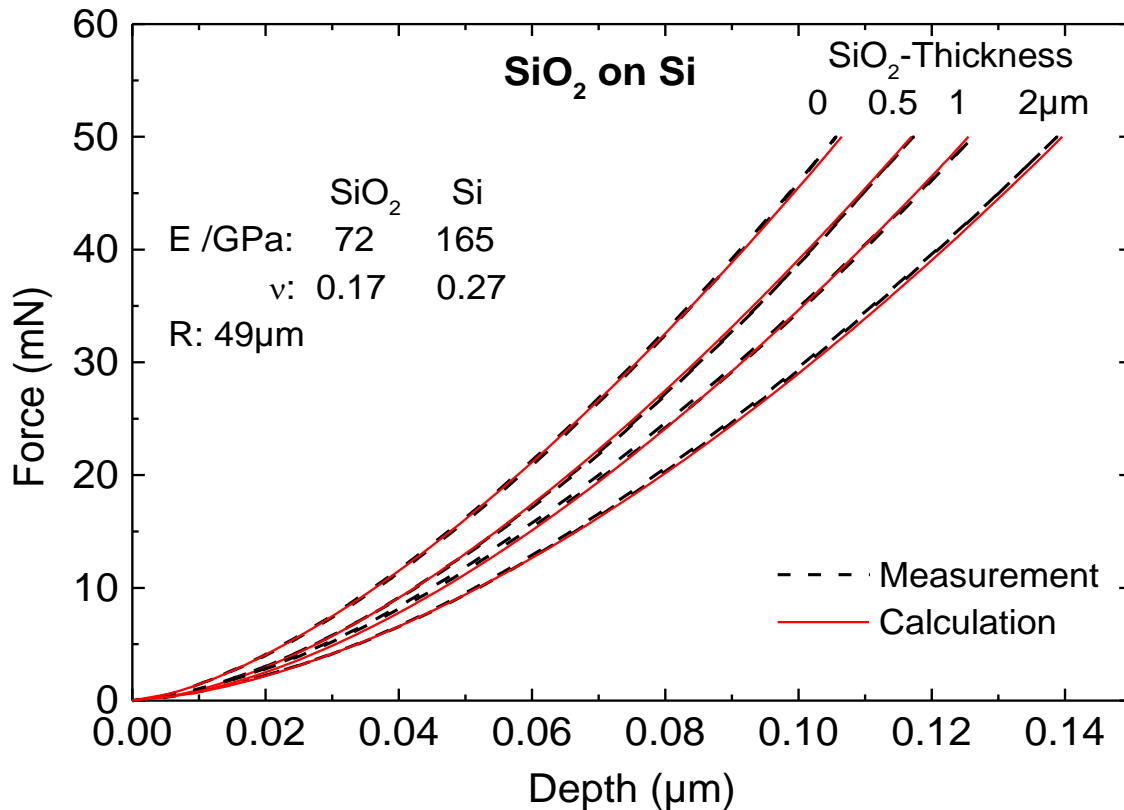
Wholly elastic indents

High accurate measurements with resolution  $< 1$  nm

Accurate knowledge of tip radius and frame compliance

Combination with elastic modelling





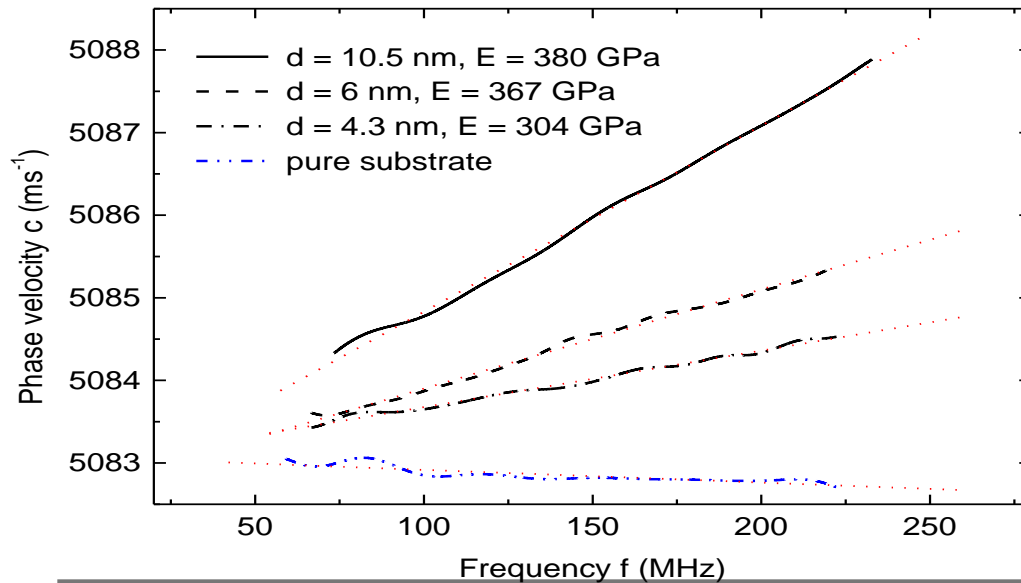
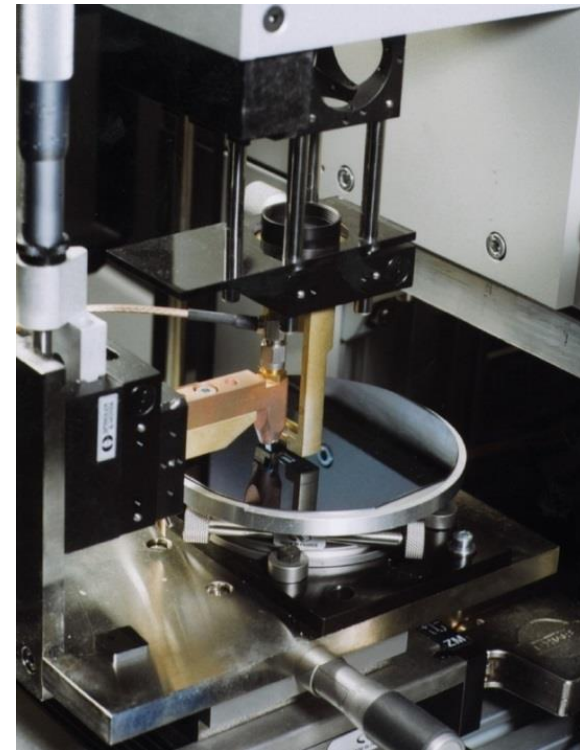
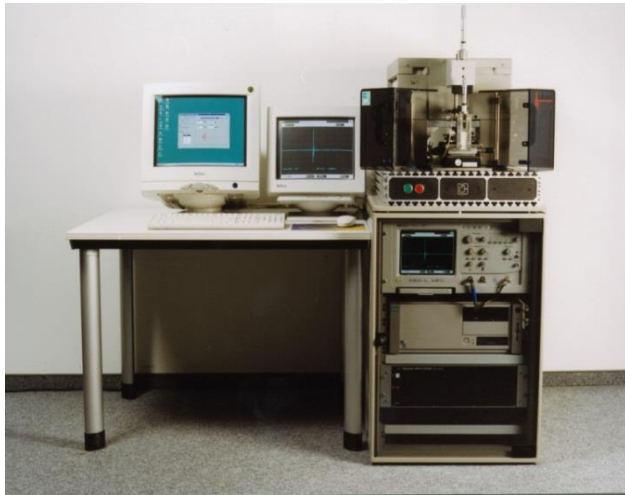
T. Chudoba, N. Schwarzer, F. Richter  
Surf. Coat. Tech. 127 (2002) 9-17

Fit of the measurement data with a theoretical load-displacement curve.

Known substrate properties; **fit parameter:** film modulus

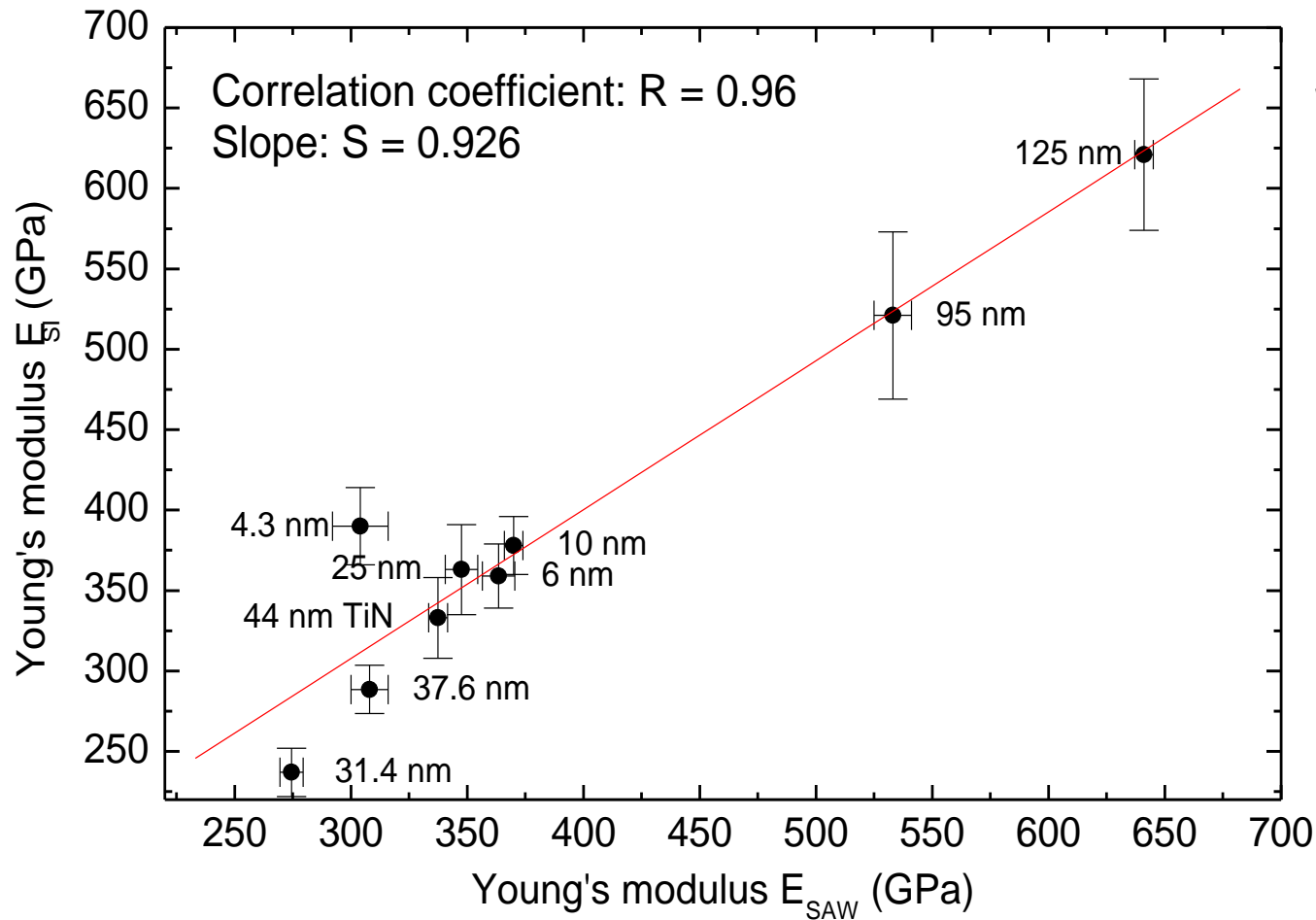
Useable software [ELASTICA](#), [FilmDoctor](#)

## Acoustic surface waves DIN 50992-1



*LA wave*

## Comparison SAW - Nanoindentation



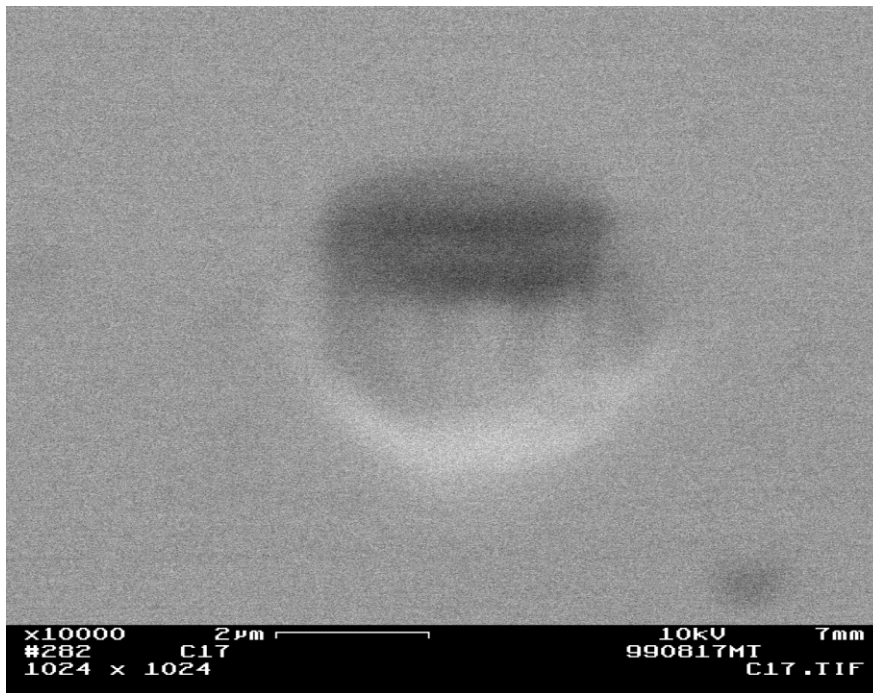
**Without 4.3 nm:**  
 **$R = 99.1$ ;  $S = 0.993$**

## Yield strength measurements

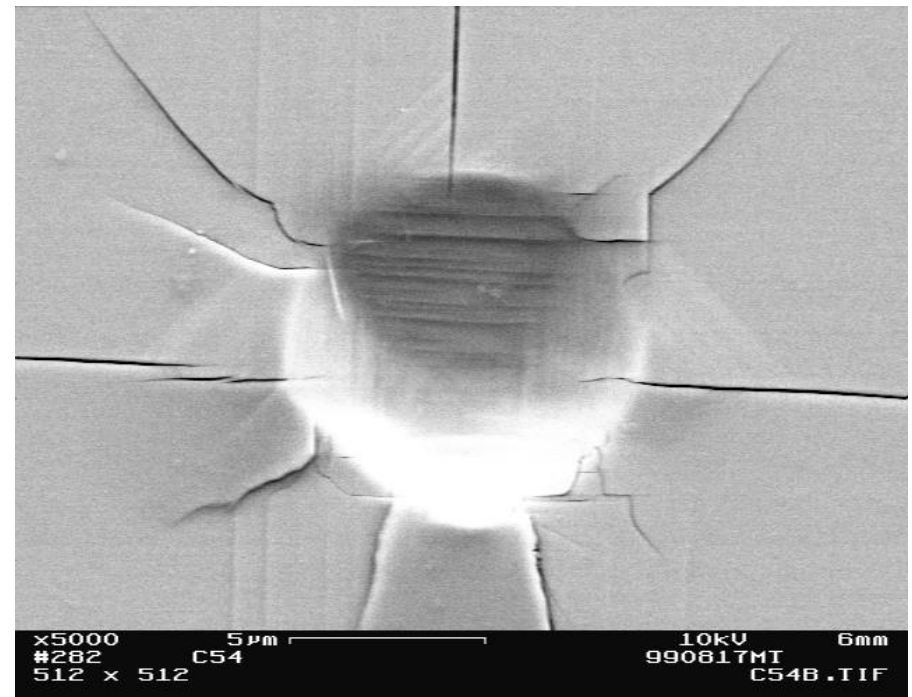
## Yield strength

Can we measure the yield strength of hard and brittle materials?

**Yes**, it is a question of dimension!



small load

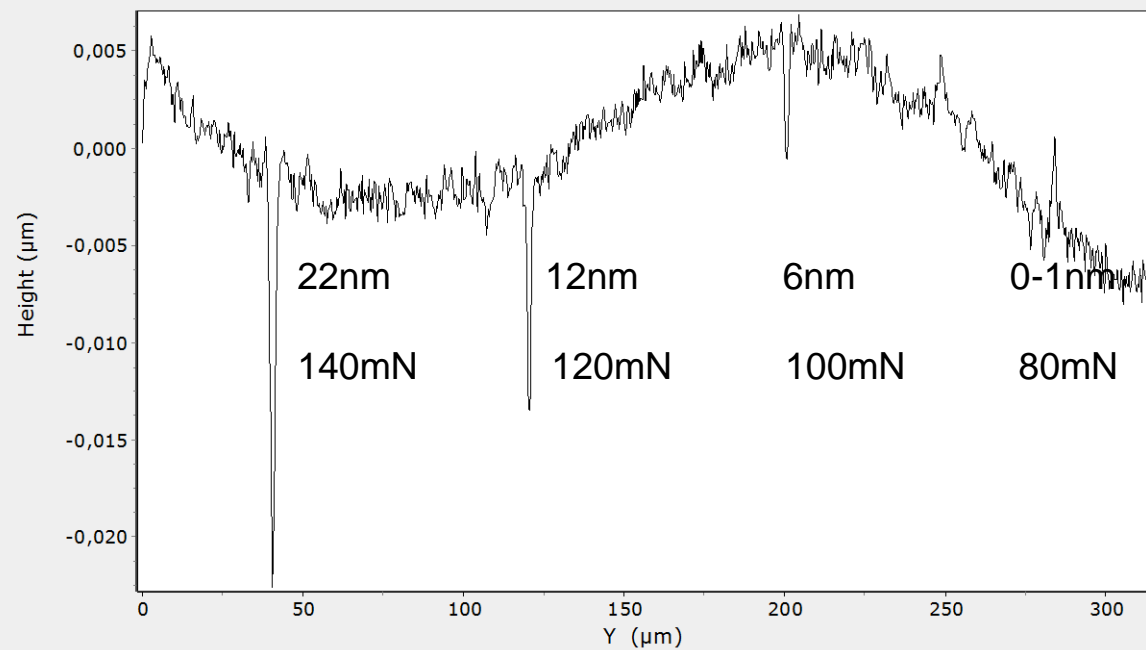


higher load

$4\ \mu\text{m}$  radius indenter in GaAs

## Wear test on fused silica (100 cycles)

Wear depth (obtained by surface scan)



5 tests in a distance of 80  $\mu\text{m}$ ;  
Time per cycle: 5s  
Forces: 50, 80, 100, 120, 140 mN

Distance:  $\pm 20 \mu\text{m} = 40 \mu\text{m}$   
Speed: 16  $\mu\text{m/s}$

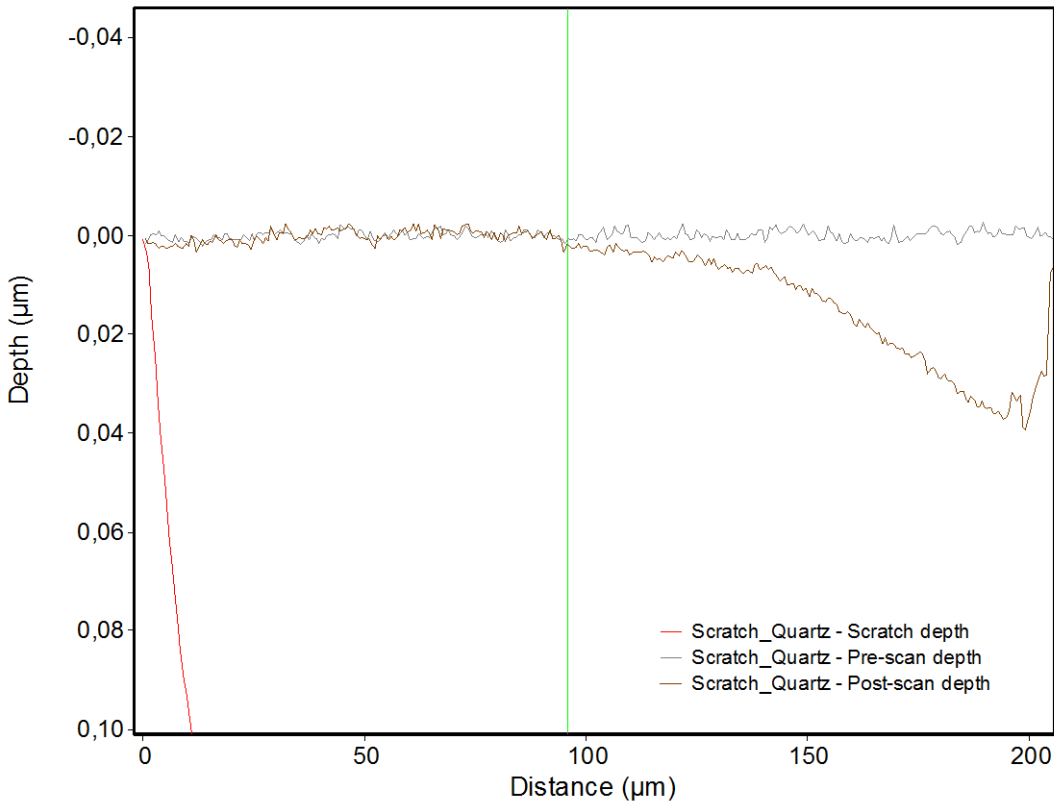
## Scratch test on fused silica

Maximum force: 300mN (increasing)

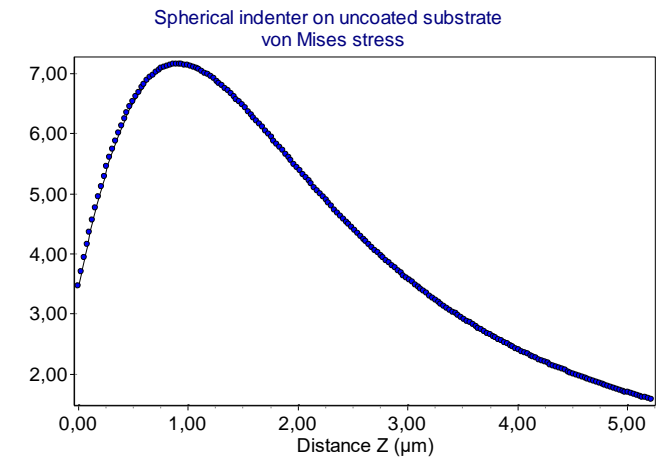
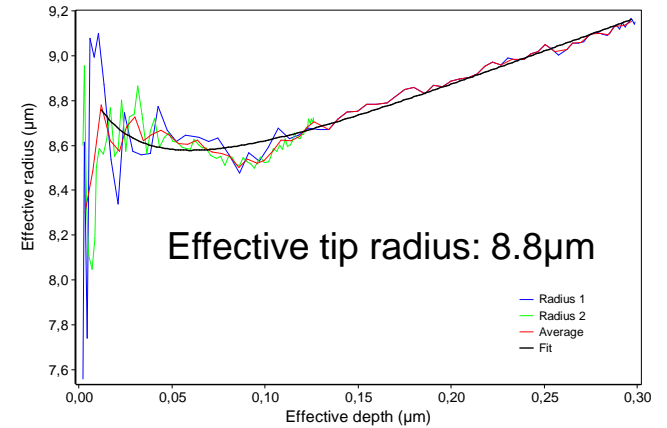
Distance: 250 $\mu$ m

Speed: 10 $\mu$ m/s





Critical force for plastic deformation: 95.7mN

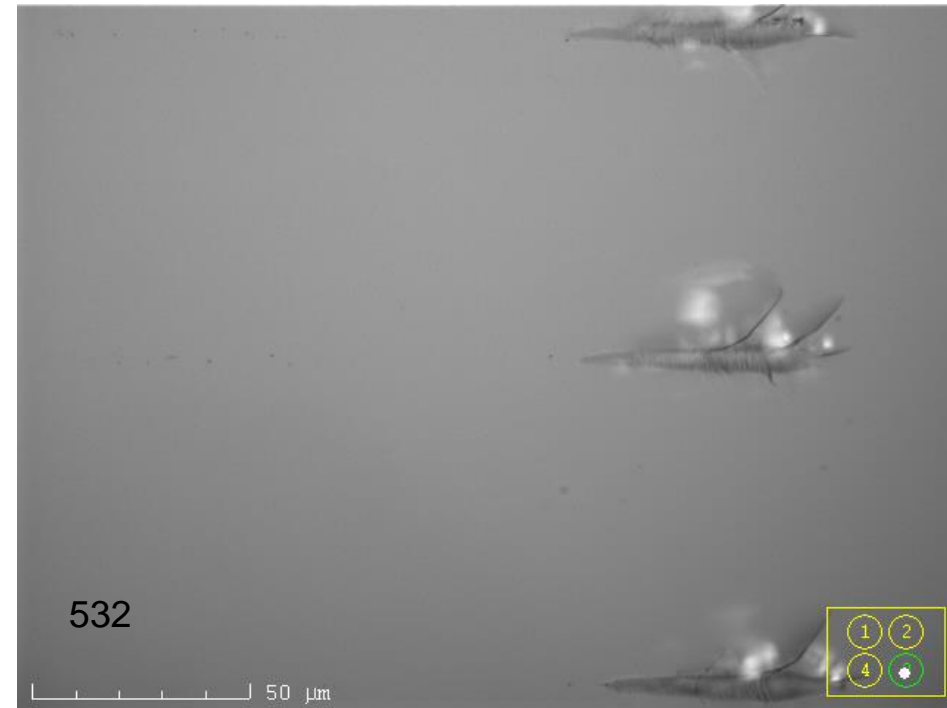
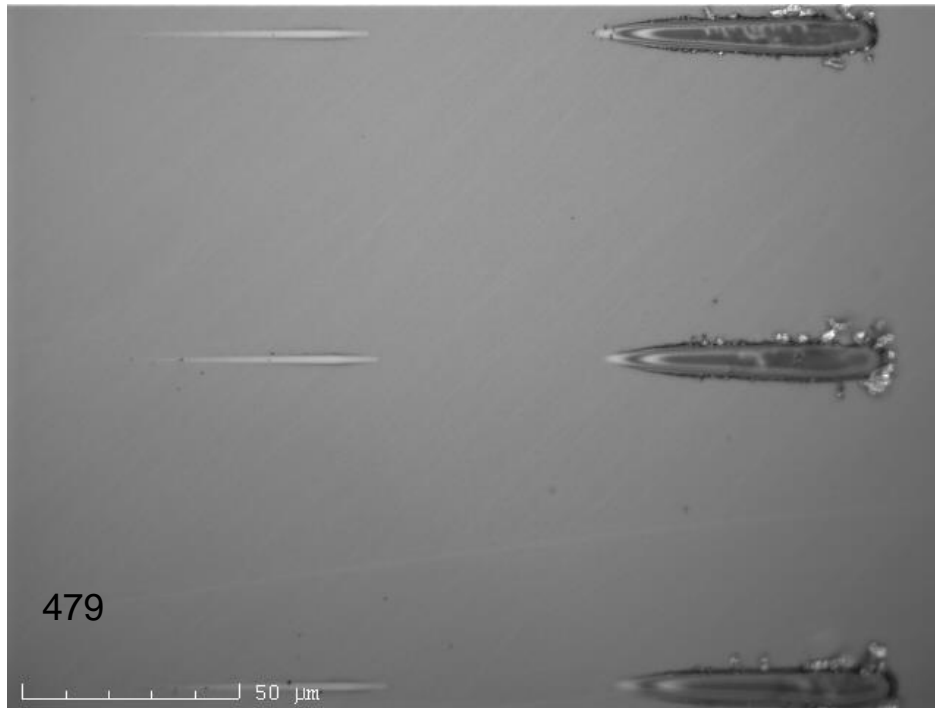


Max. von Mises stress: 7.15 GPa  
 = yield strength



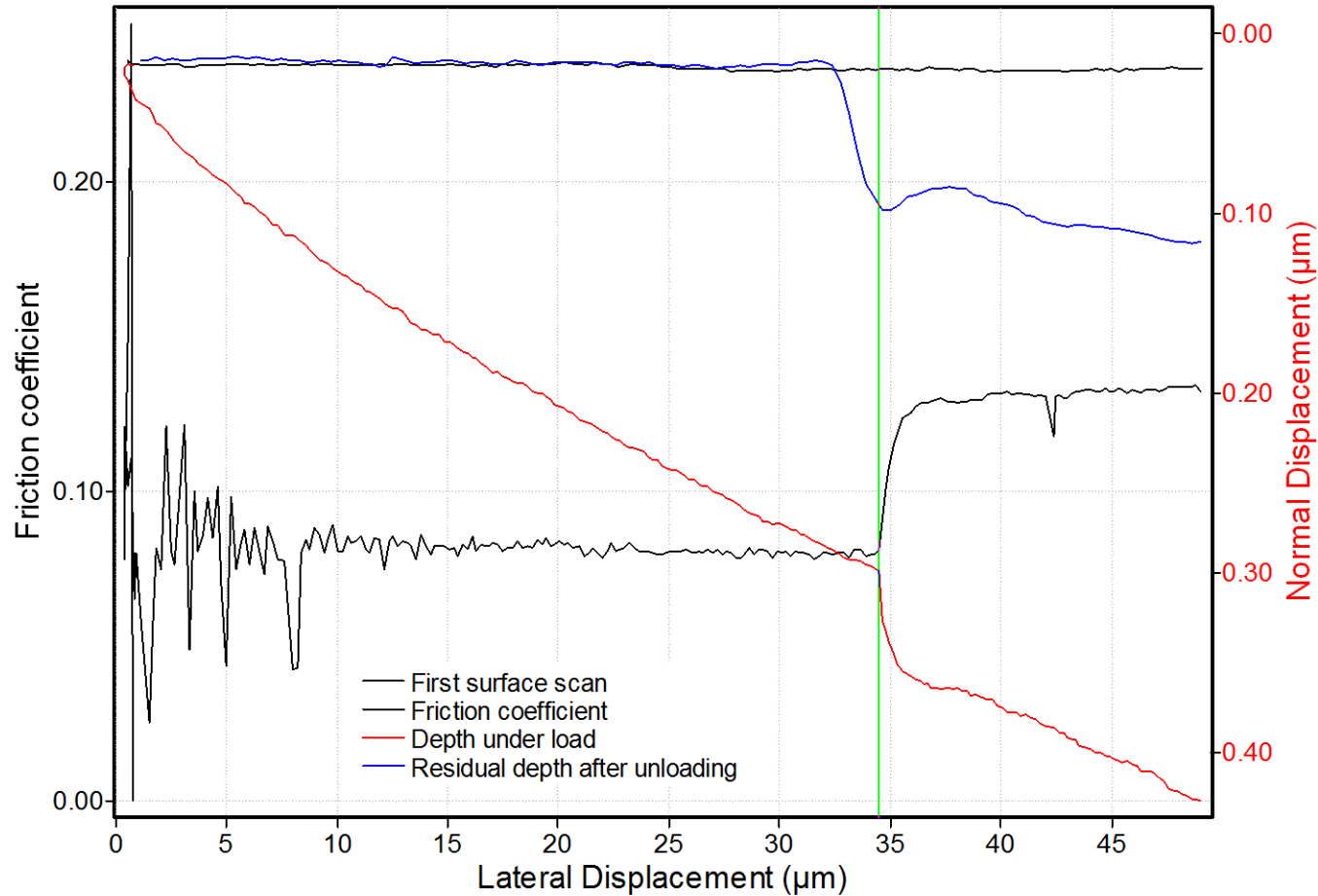
## 240 nm thick optical coatings on sapphire

Three 50mN und 700mN micro scratch tests over each other



The two coatings show different failure modes

The difference between pre-scan and post-scan of the surface allows detection the elastic-plastic transition



Failure of the coating at: X: 34.48 μm Fn: 174.94 mN μ: 0.081

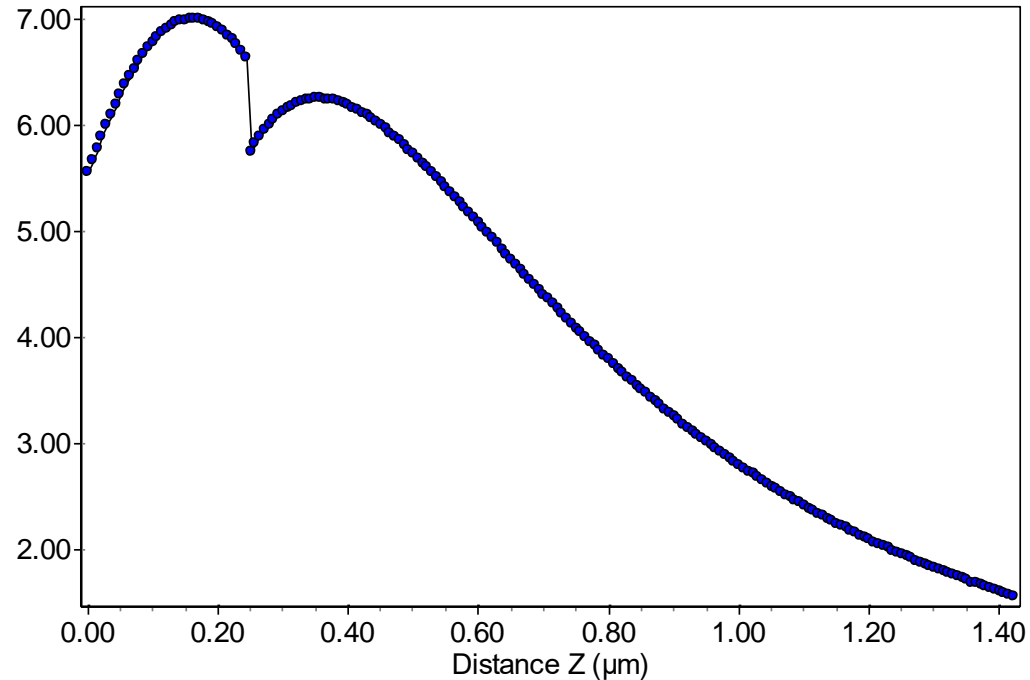
Yielding starts in the coating since the substrate was hard enough for these samples

## Yield strength results

Sample number	Fcrit mN	$\mu$	Yield strength GPa
<b>479</b>	<b>6.89</b>	<b>0.061</b>	<b>6.97</b>
486	6.60	0.067	6.5
488	8.30	0.078	7.4
489	8.49	0.088	7.64
531	12.75	0.110	8.79
<b>532</b>	<b>131.79</b>	<b>0.069</b>	<b>20.8</b>
485/Sapphire	8.93	0.065	6.31
485 /Glass	31.92	0.057	6.25
Saphir	171.500	0.084	27.7

on sapphire
on glass

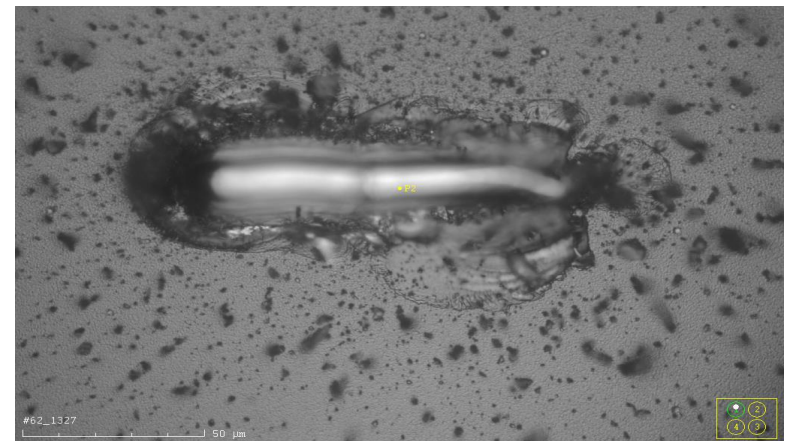
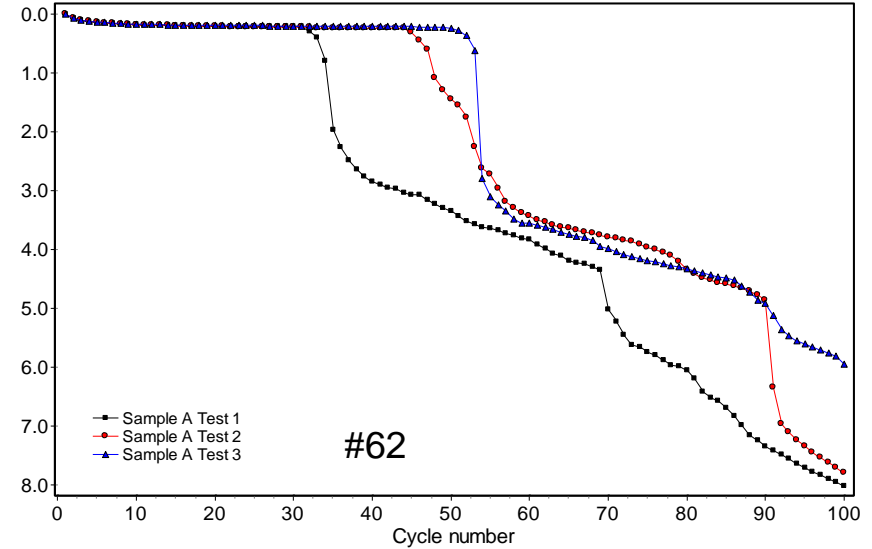
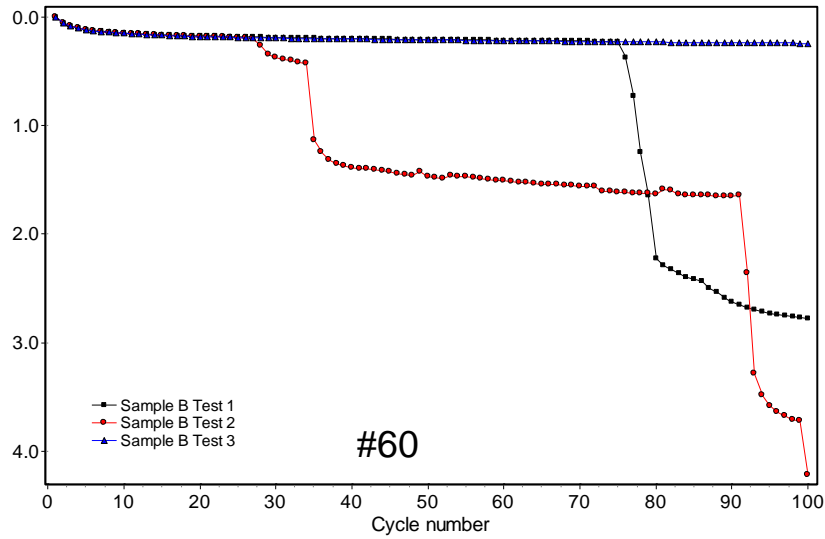
Spherical indenter on substrate with one layer  
von Mises stress

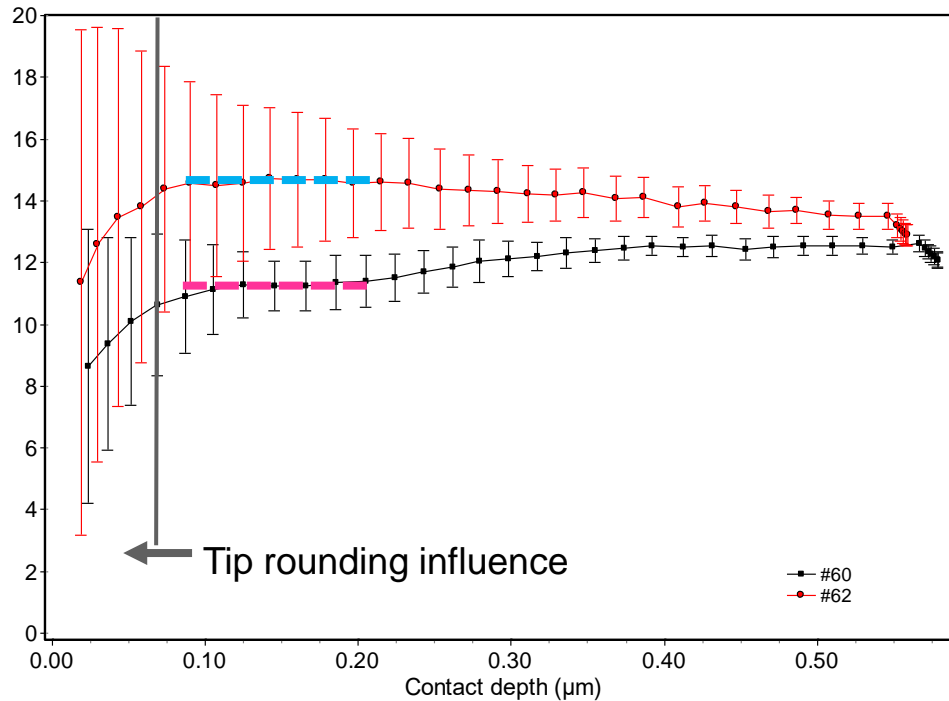


Von Mises stress profile for sample 479

## Low cycle fatigue

Wear behavior of sample #60 is better than for sample #62  
 although the hardness is lower  
 Test conditions: 500mN, 9.3 $\mu$ m radius indenter





(#60) versus (#62)

Hardness

11.3 GPa < 14.8 GPa

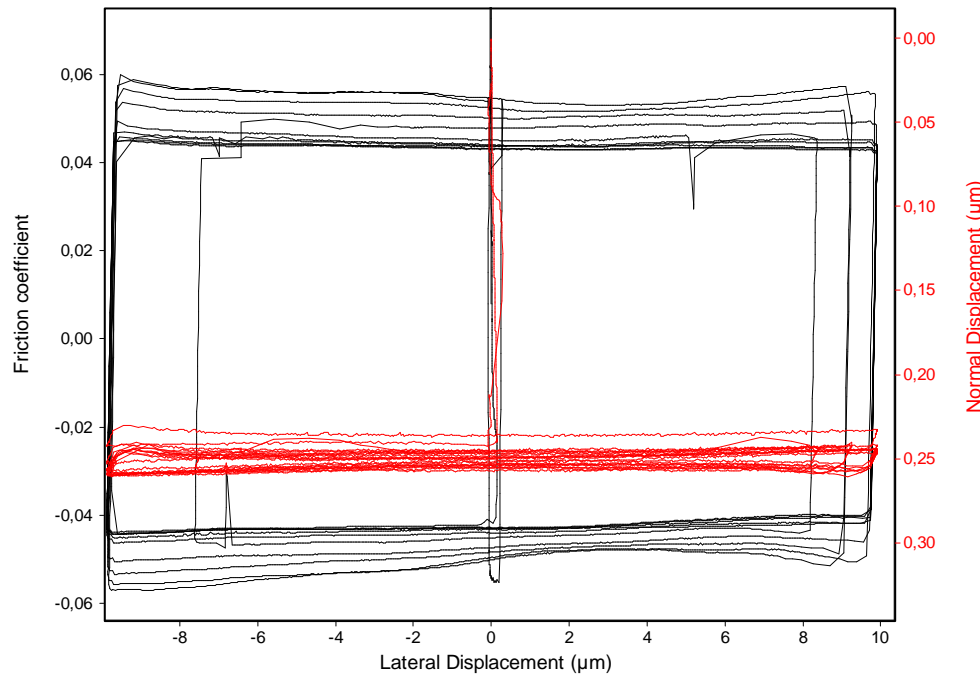
Av. failure cycle

71 > 43

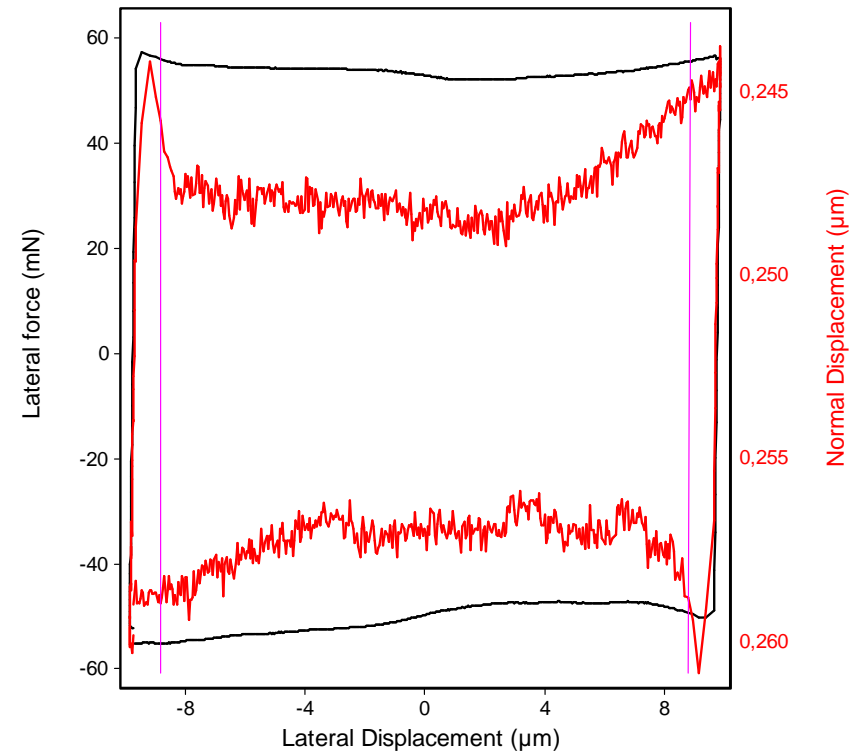
Hardness is not the only criteria for the durability of a coating.

## Micro wear tests

Example: **Dry** friction of diamond (Rockwell indenter) against diamond layer  
 Minimum friction: 0.04      Normal force: 1N

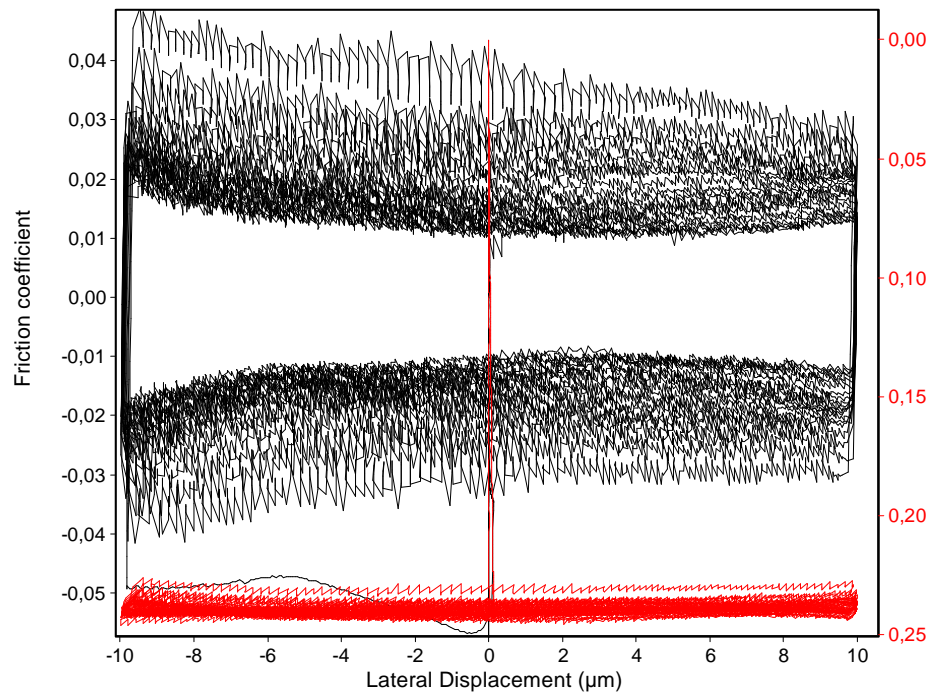


10 cycles in one graph

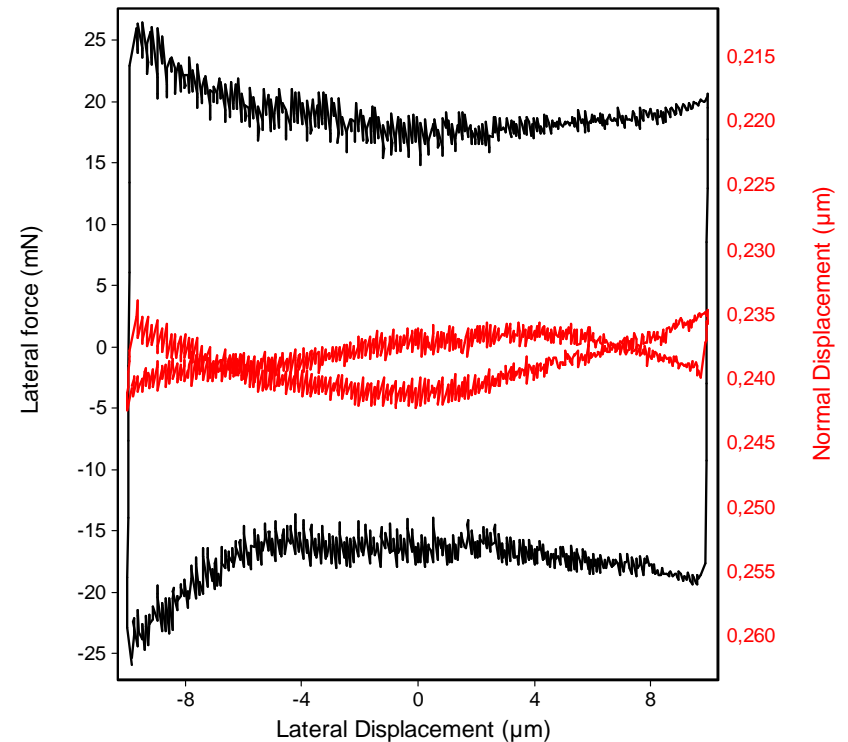


Cycle 9

Example: **Wet** friction of diamond (Rockwell indenter) against diamond layer  
 Minimum friction: 0.01      Normal force: 1N

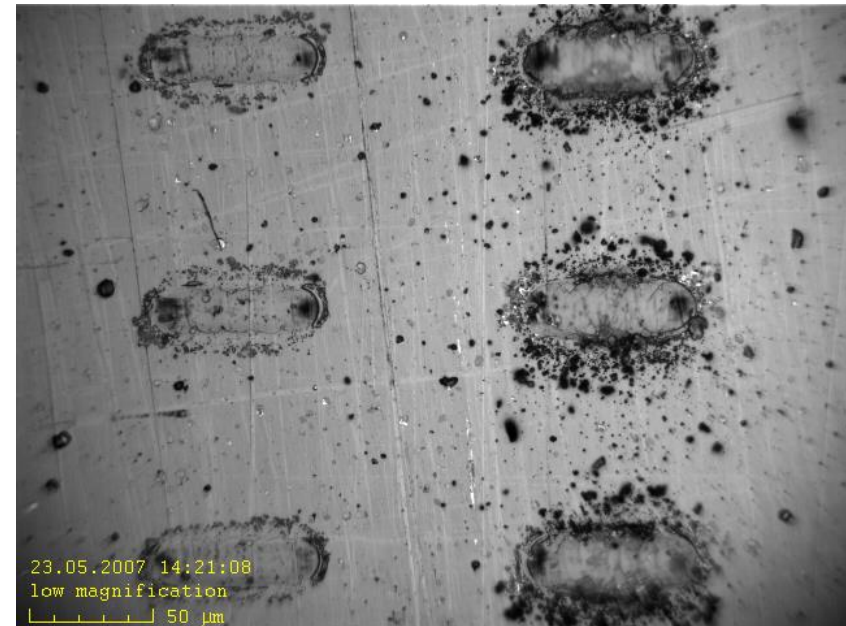


20 cycles in one graph



Cycle 9





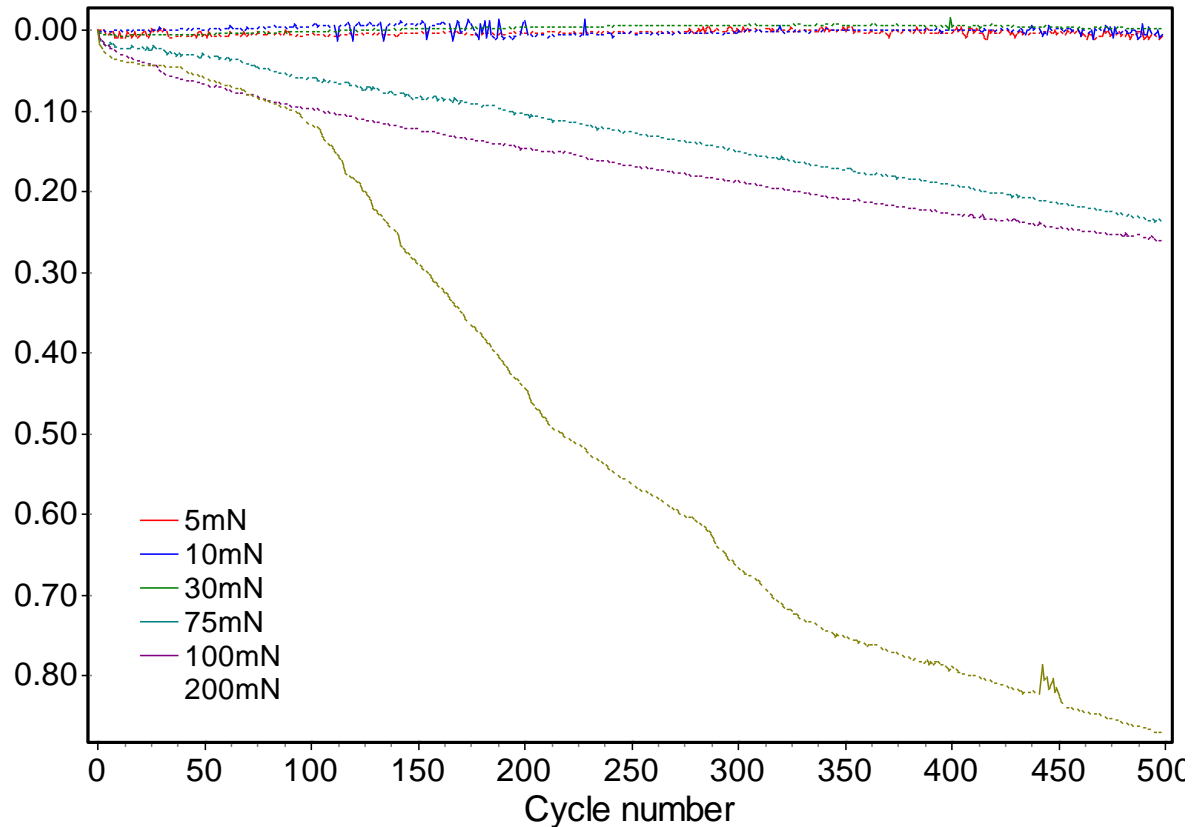
2.5 $\mu\text{m}$  DLC on steel, 55 $\mu\text{m}$  tip radius  
Amplitude 50  $\mu\text{m}$ , Measuring time 1800 s

## CVD DLC on steel

5,7 $\mu$ m film thickness, Film modulus: 100 GPa, Hardness: 12 GPa

Indenter: diamond sphere, 6  $\mu$ m radius

Displacement amplitude 80 $\mu$ m measurement time 3144s

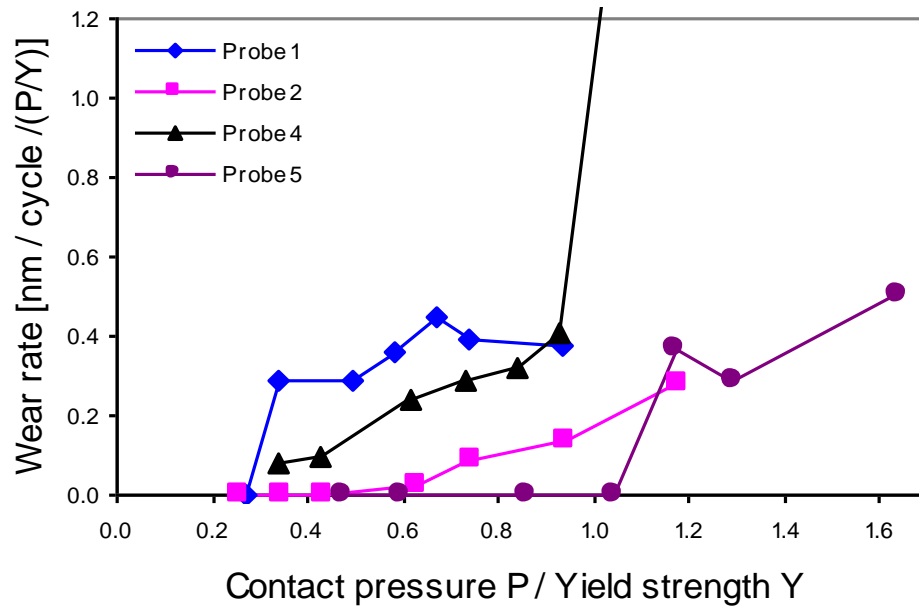


**No wear until 50mN normal force**

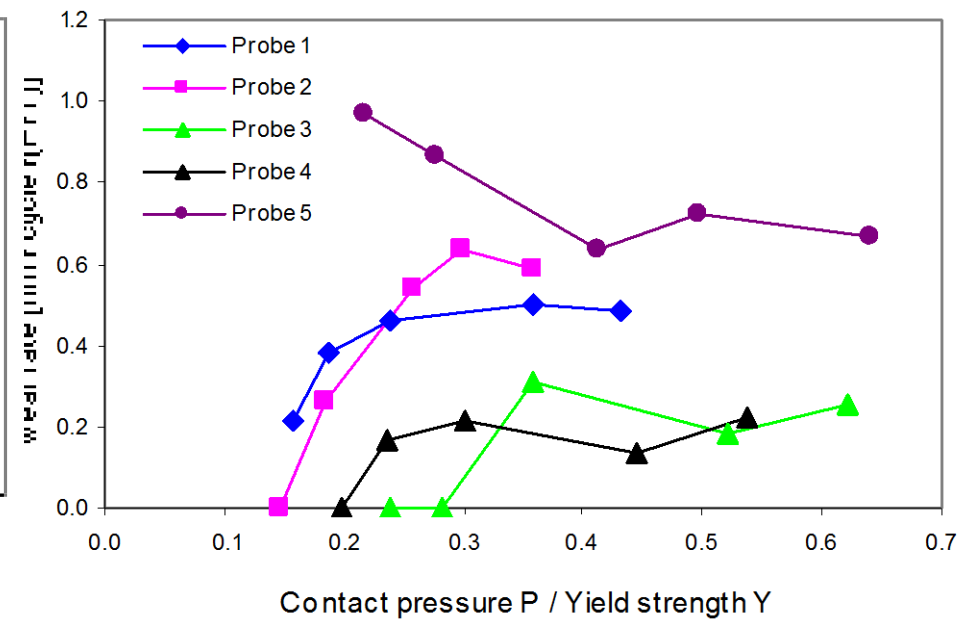
Wear rate at 75mN and 100mN: 0,4nm/cycle

## Wear tests on different DLC coatings (a-C:H and t-aC)

### Tests with diamond indenter of 6 $\mu\text{m}$ radius

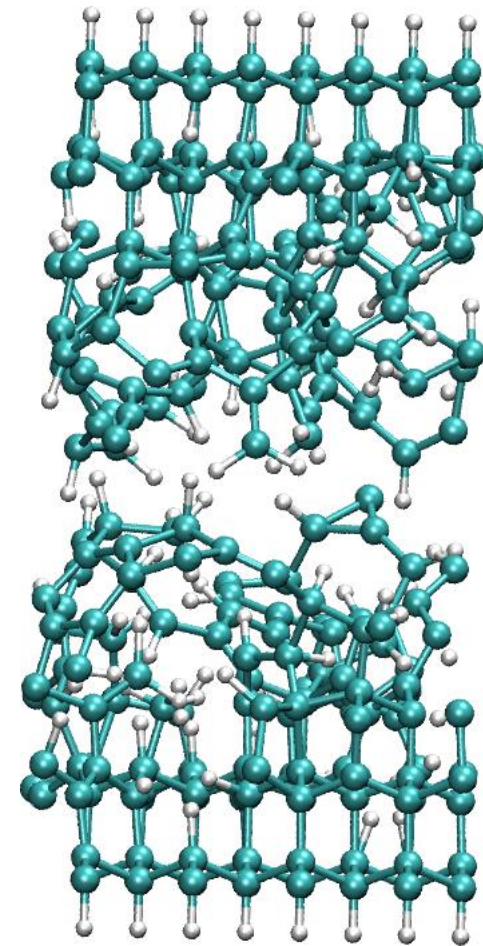
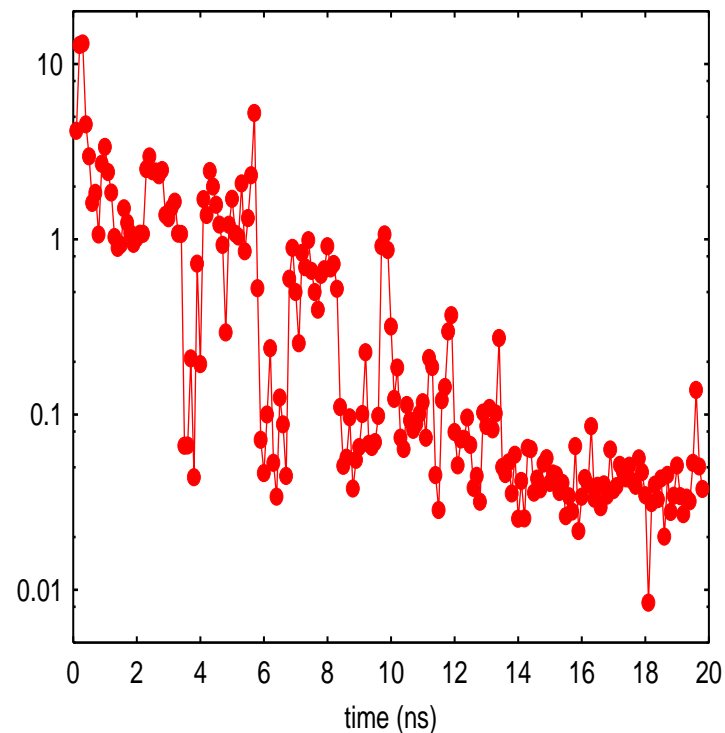
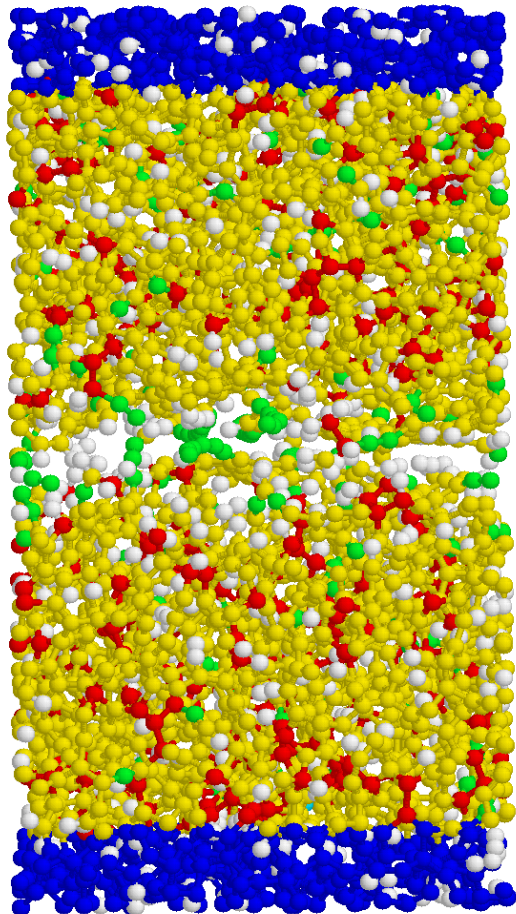


### 67 $\mu\text{m}$ radius



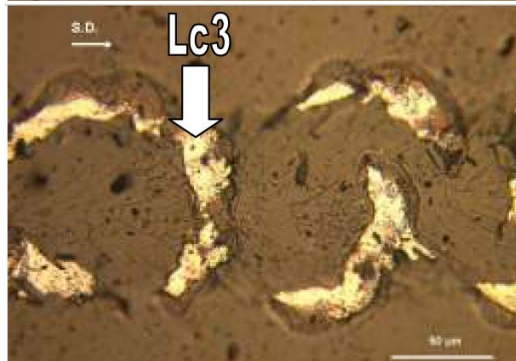
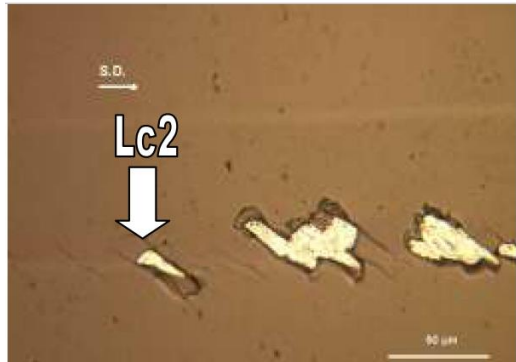
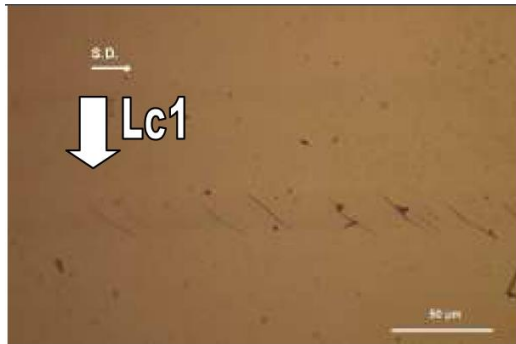
Normalization with yield strength  $Y$  (X-axis)  
and  $P/Y$ -ratio (Y-axis)

Moleculardynamic simulation of running in behavior of DLC → graphitization observed



Source: Moseler, Fh IWM, Freiburg, Germany

## **Micro scratch test as adhesion test**



## Scratch analysis via definition of critical forces

$L_{C1}$  - first cracks in the scratch track

$L_{C2}$  – local interfacial spallation

$L_{C3}$  – coating removal and visibility of substrate

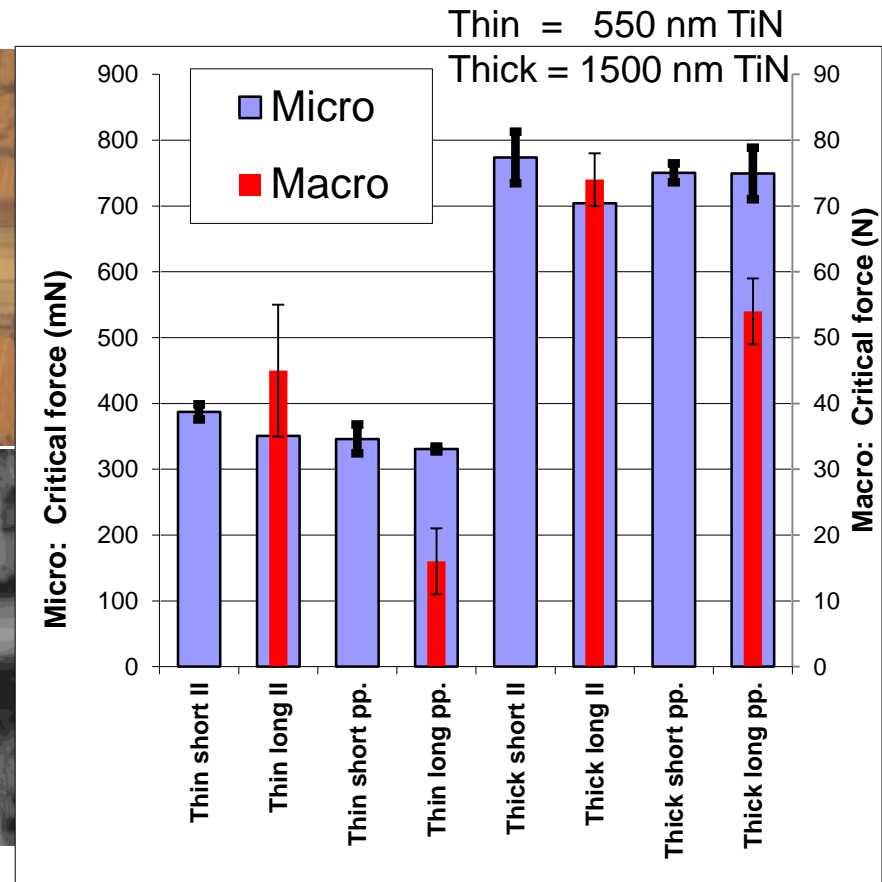
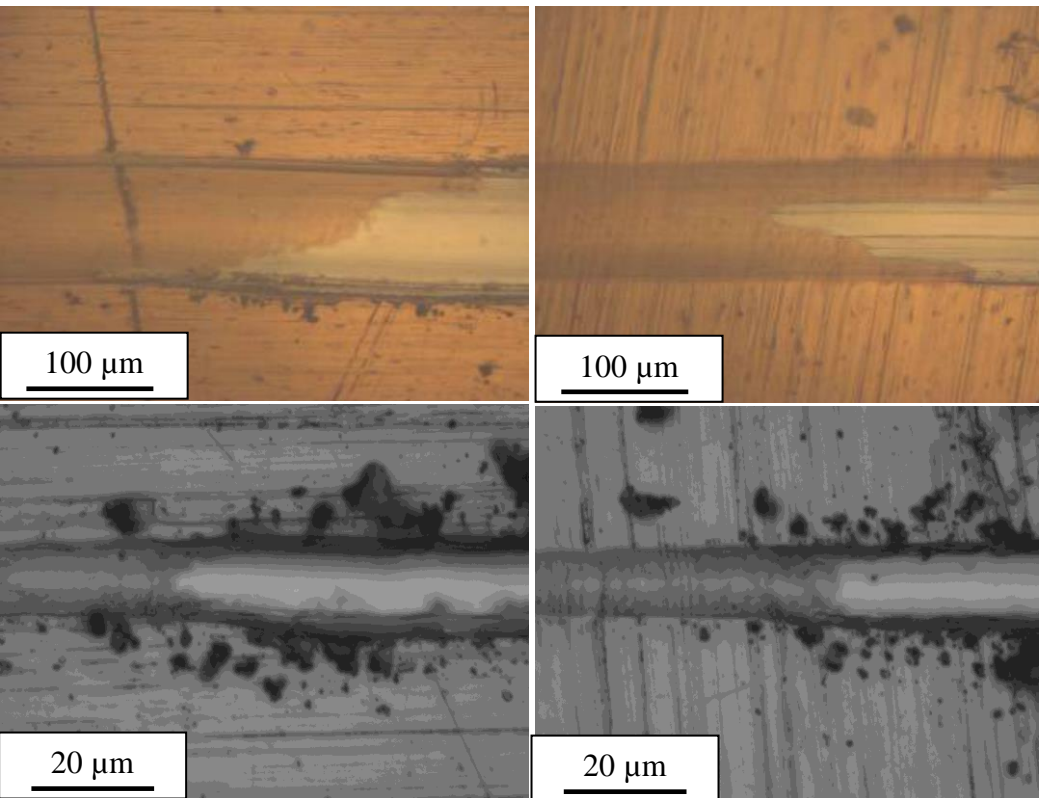
The indication  $C1 .. C3$  depends on the occurrence of the failure mode

Images from: The certification of critical coating failure loads: a reference material for scratch testing according to ENV 1071-3: 1994 IRMM (European Institute for Reference Materials)

## Macro scratch test versus micro scratch test

(from EU project Nanoindent)

550 nm TiN on steel



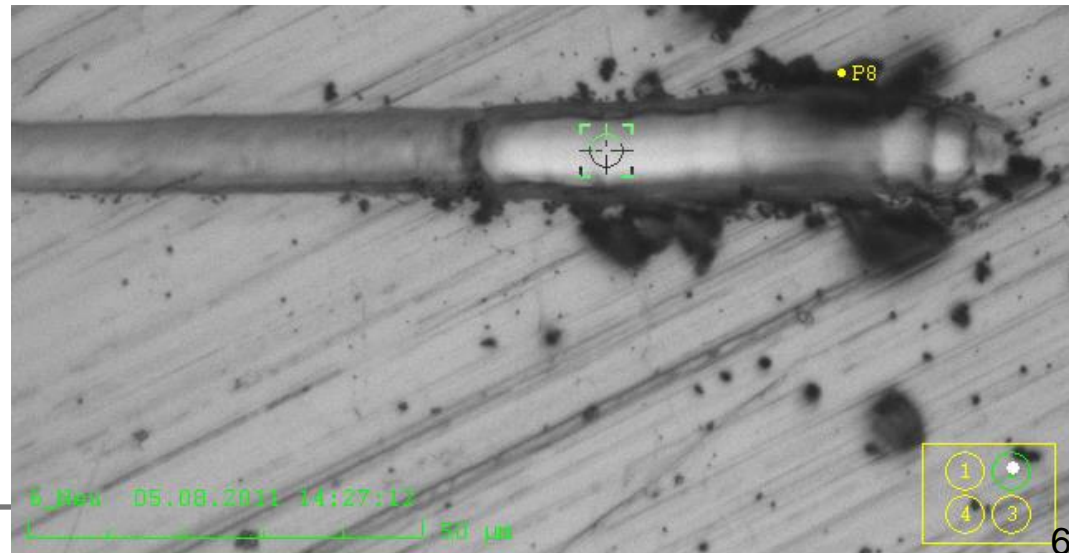
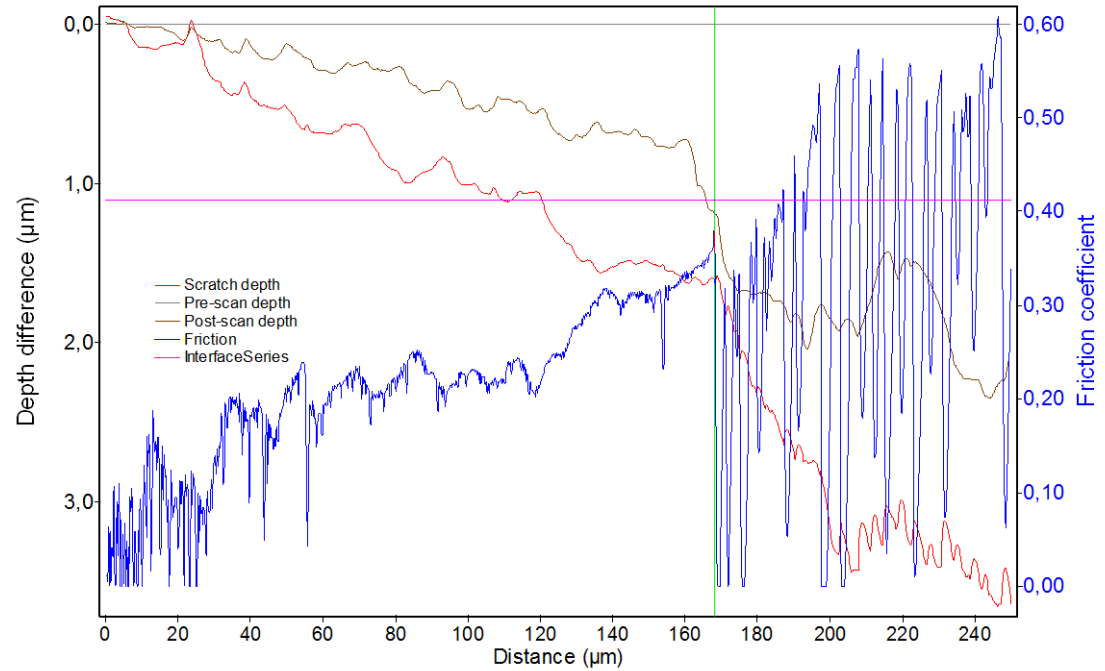
For macro scratch test there is a clear dependency from the orientation of the roughness profile.  
parallel = pp perpendicular = II.

## Example 1:

1,1  $\mu\text{m}$  TiN on steel

10,4  $\mu\text{m}$  tip radius

$F_{C3} = 662 \text{ mN}$



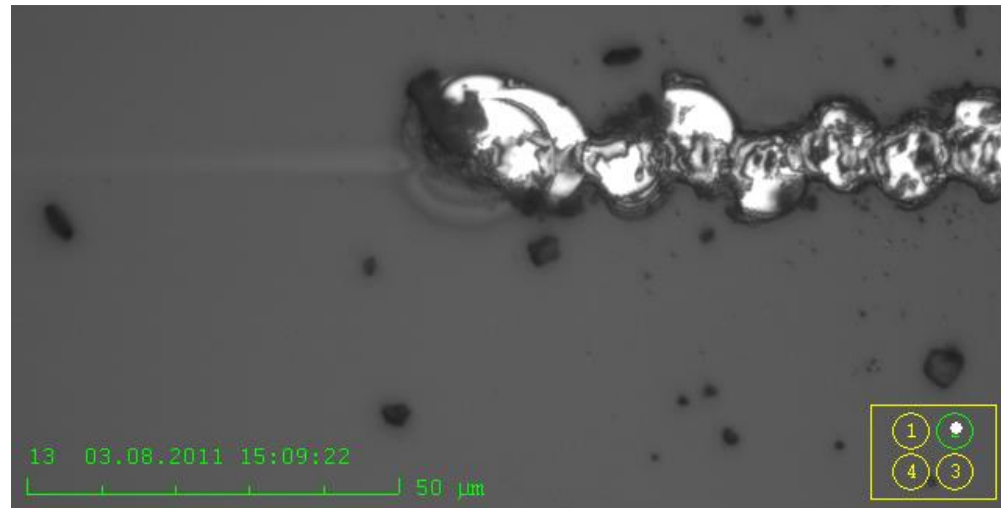
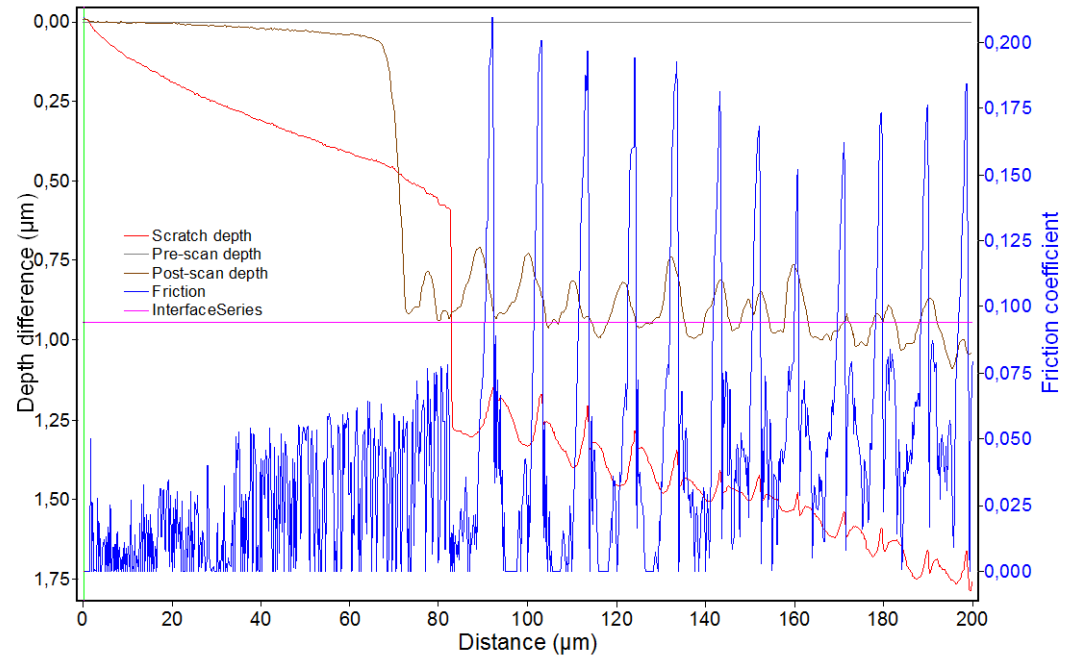


## Example 2:

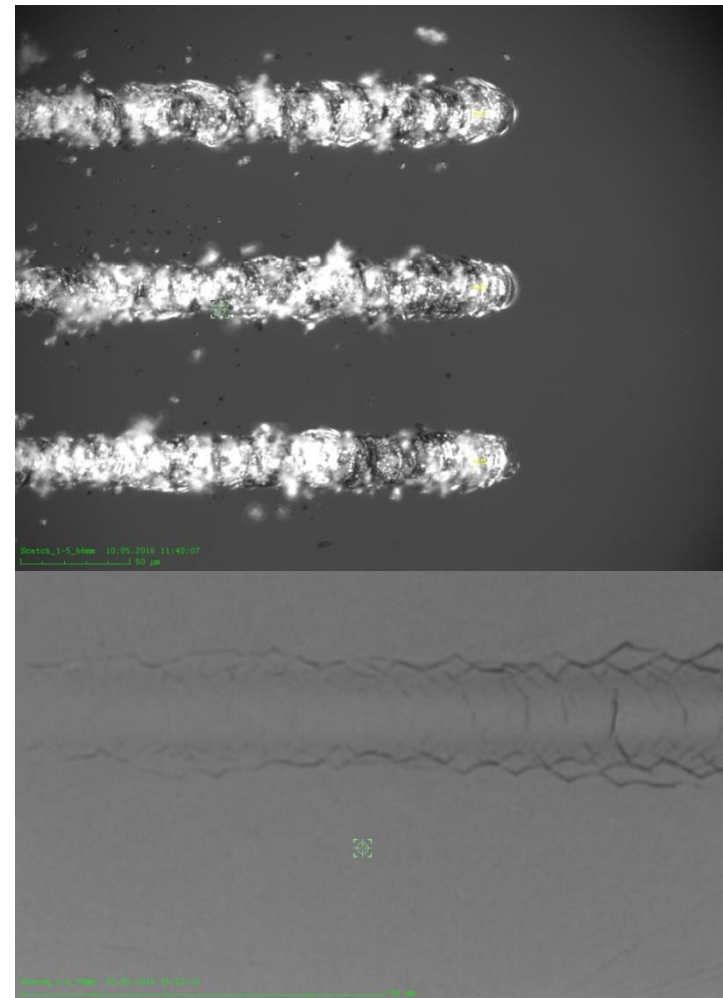
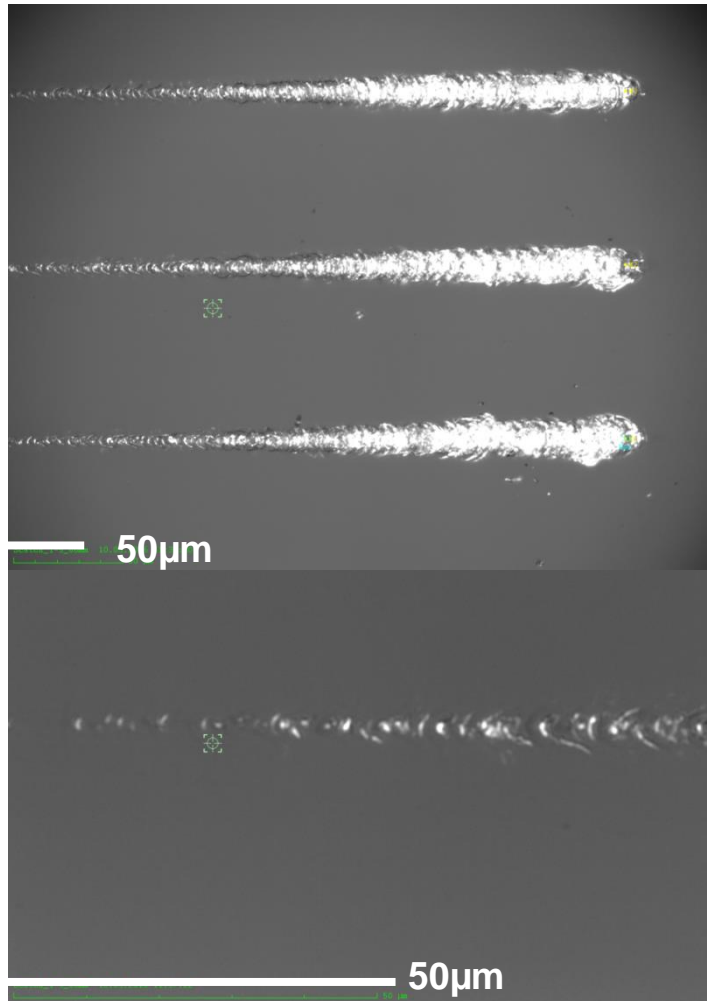
0,94  $\mu\text{m}$  soft DLC on Si

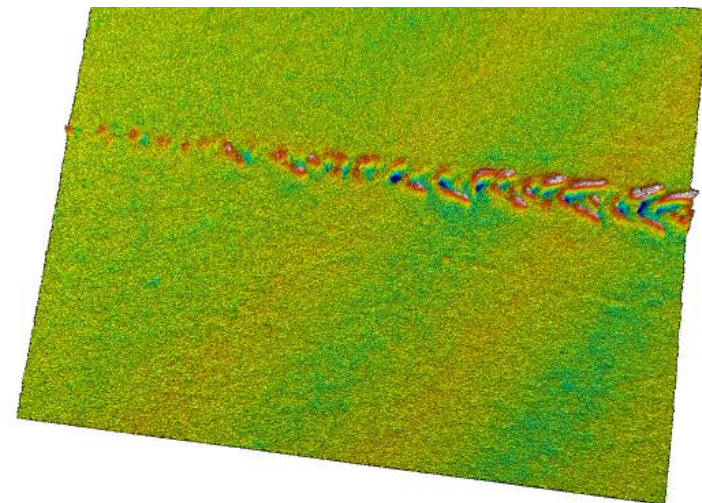
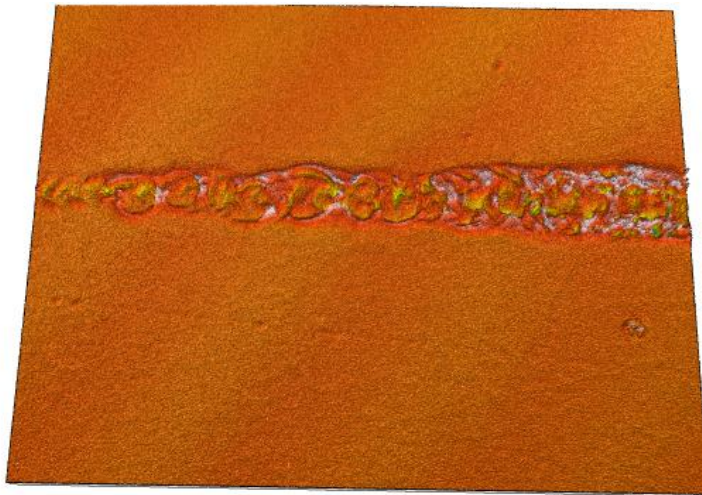
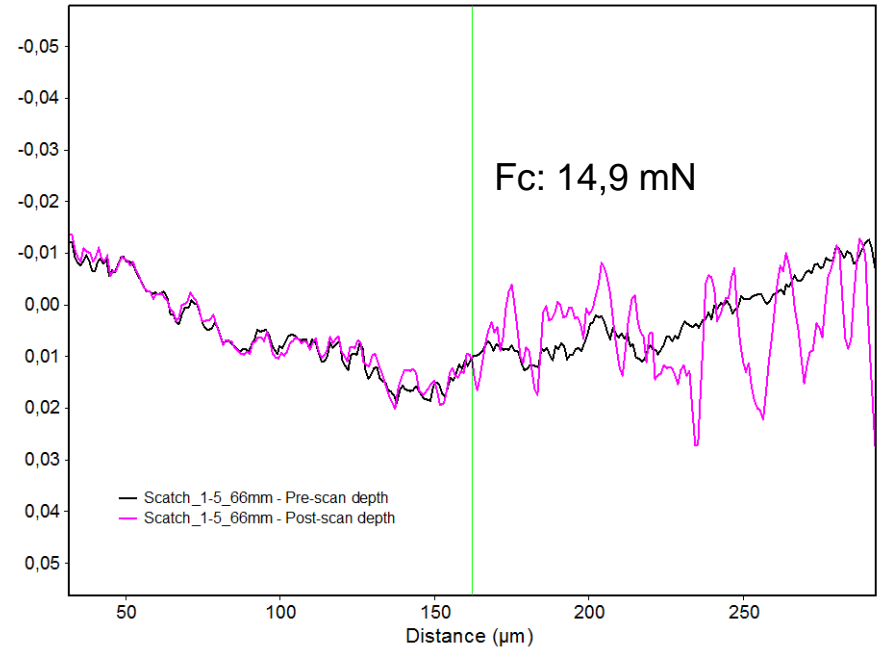
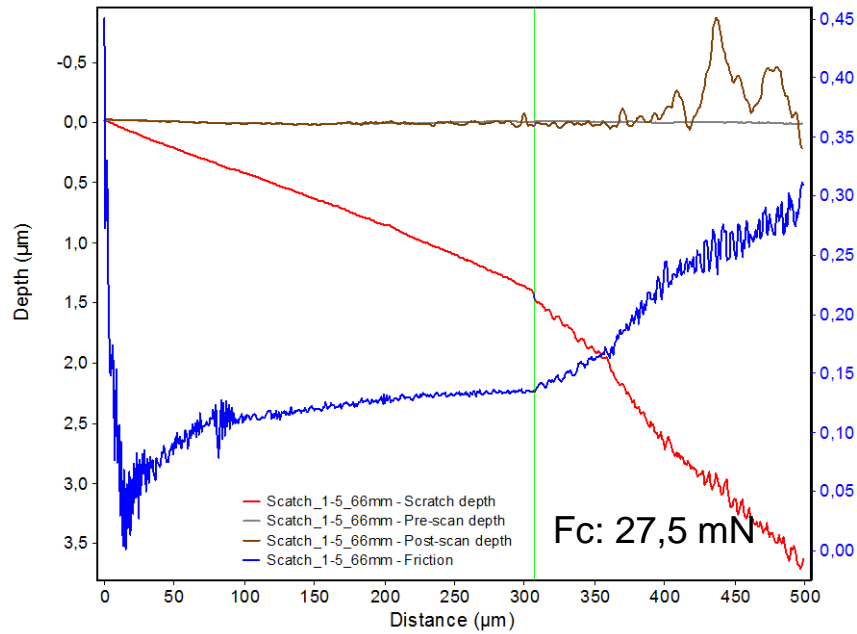
10,4  $\mu\text{m}$  tip radius

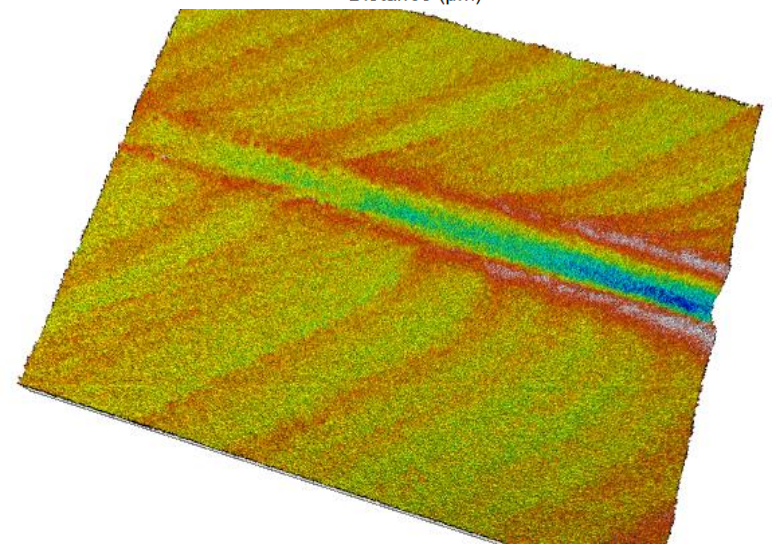
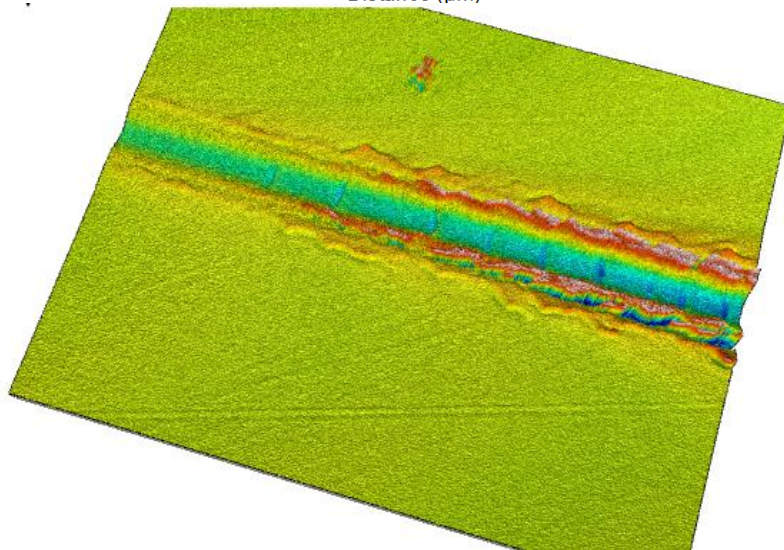
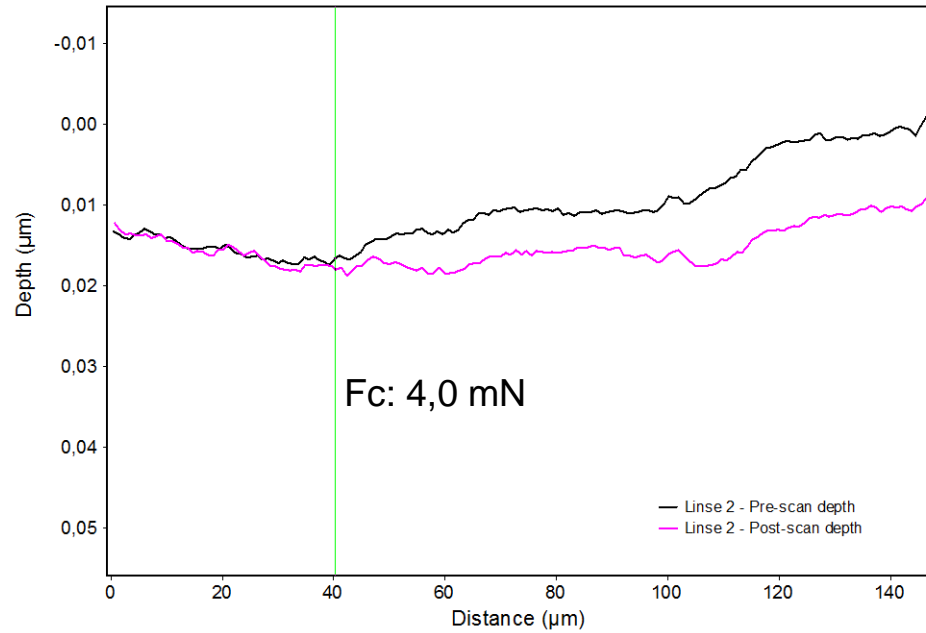
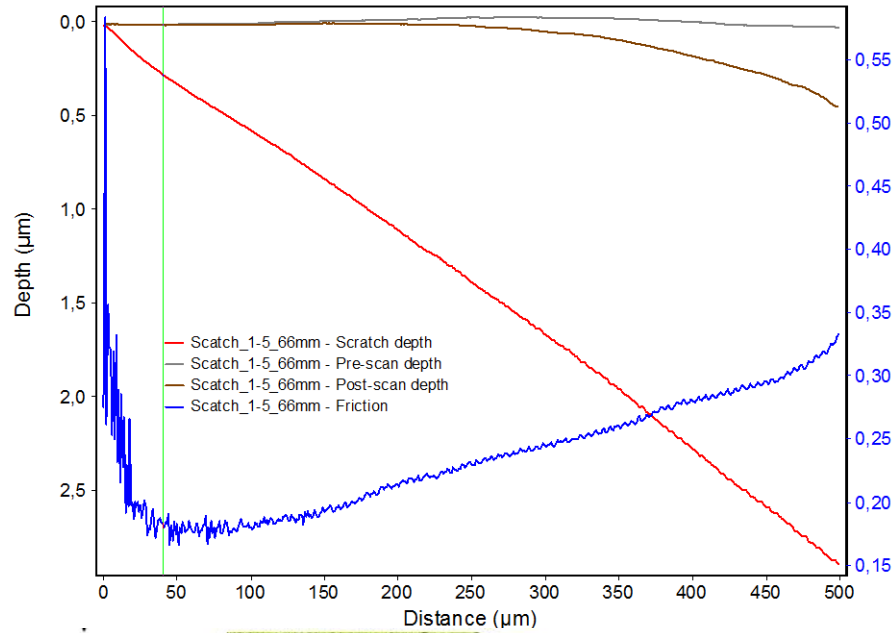
$F_{C3} = 136 \text{ mN}$



**Example 3:** Polymer lenses with hard varnish and anti reflective coating







## Mapping of mechanical properties

## Local versus global properties



The weakest link is deciding about the durability of a system.  
Global tests are therefore advantageous.

### ***Property***

Yield strength  
Young's modulus  
Adhesion

### ***Global tests***

Tensile test  
Tensile test  
Cavitation test

### ***Local tests***

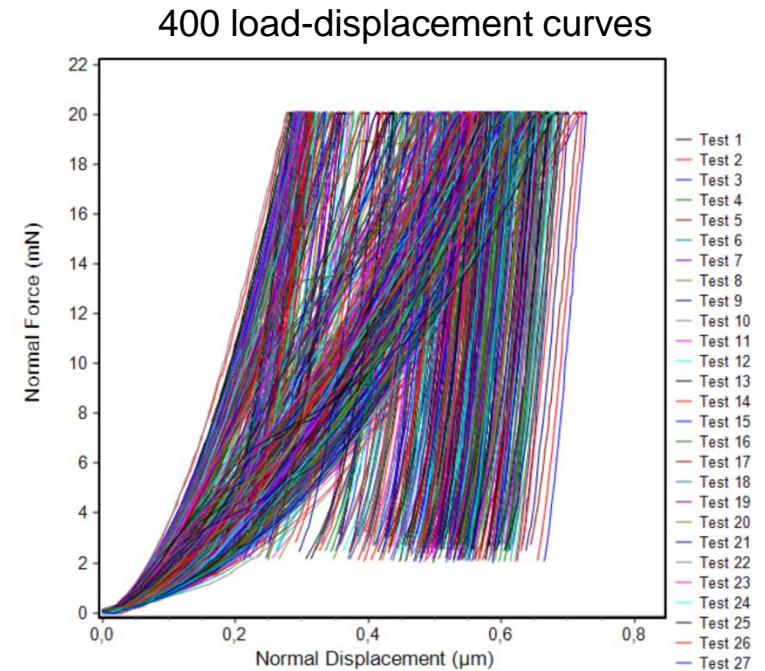
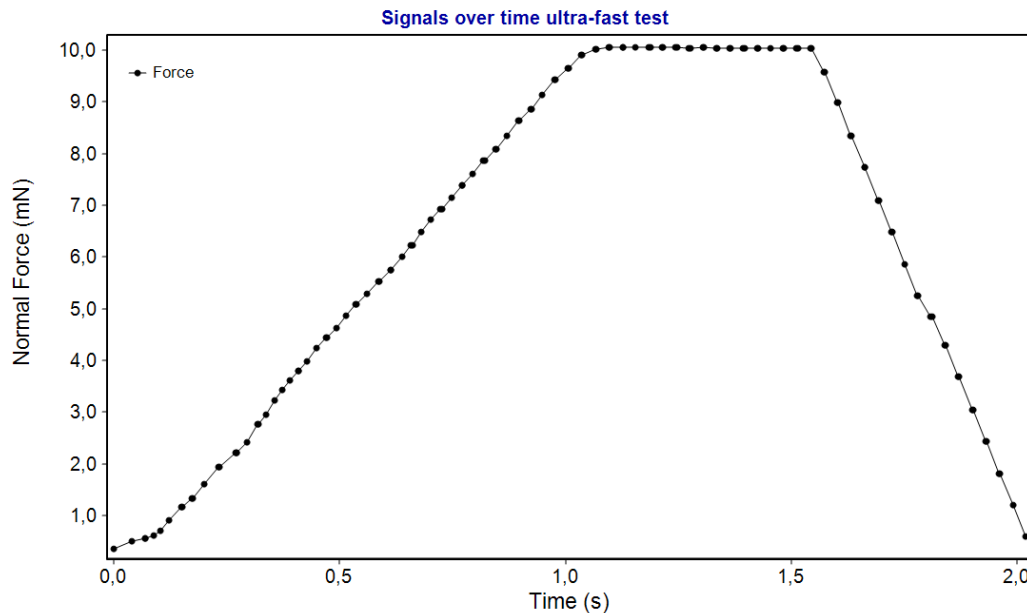
Indentation test  
Indentation test, Ultrasonic test  
Scratch test

The mapping of local mechanical properties is a step towards a global characterization.

## Ultra fast hardness tests with mapping function

- Test time recommended according standard: 130s + 30s Approach = 160s \* 400 = 1067min = 44h
- Test time for fast hardness test: 19s + 30s Approach = 49s \* 400 = 327min = 13,6h
- Test time for ultra fast hardness tests: 2s + 3s Approach = 5s \* 400 = 34min = 0,5h

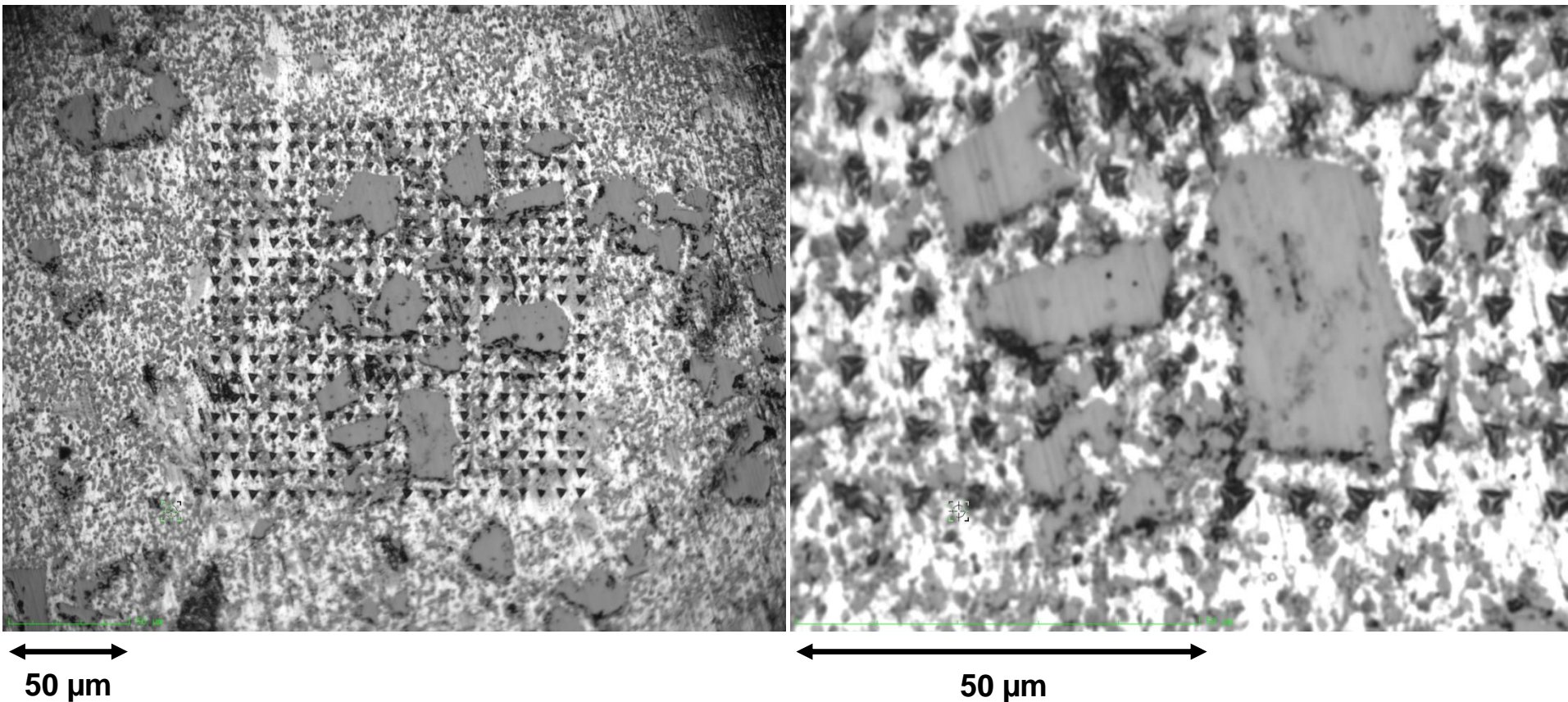
Example: Ultra fast tests with a typical data rate of 64 Hz



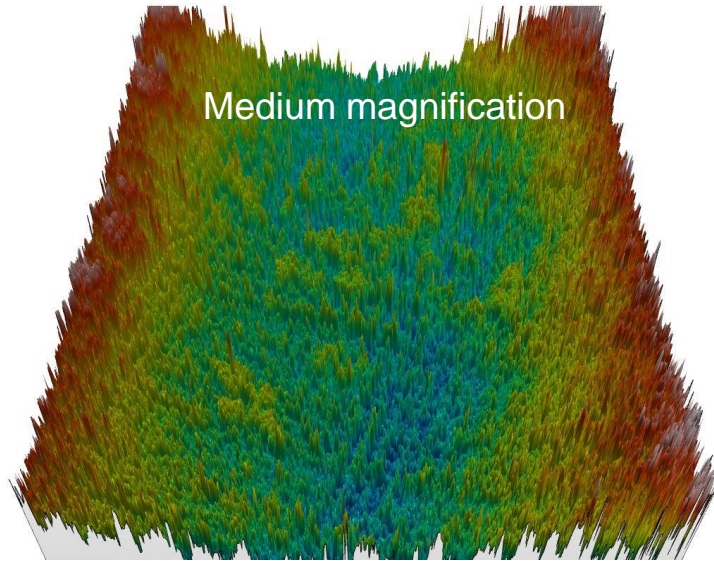
## Example 1: Hardness and modulus mapping

Sample: Al-Si composite (Al matrix with Si particles) , experimental cylinder liner from Yamaha

Test conditions: Maximum force: 20mN, Test time: 2s

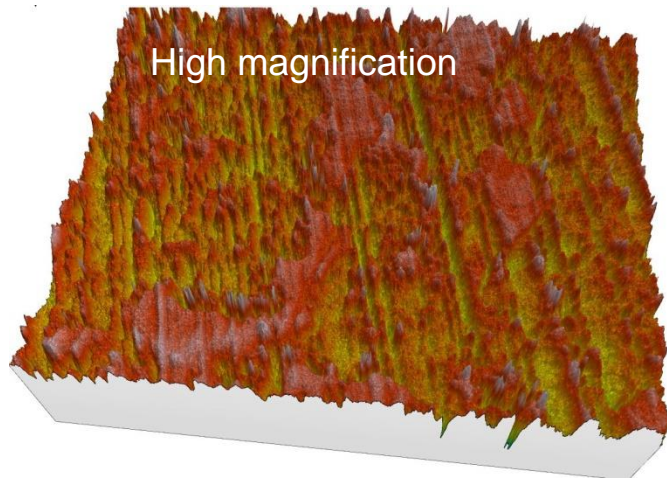


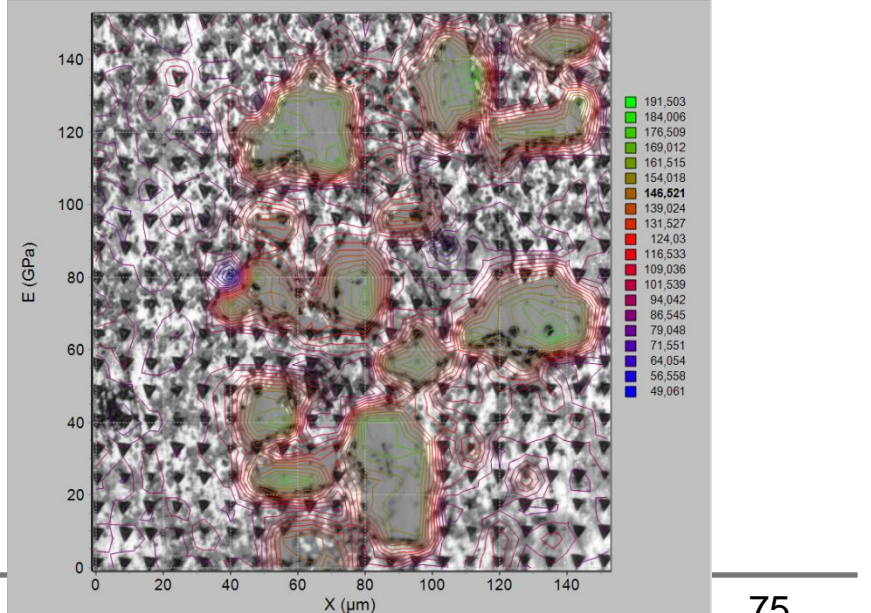
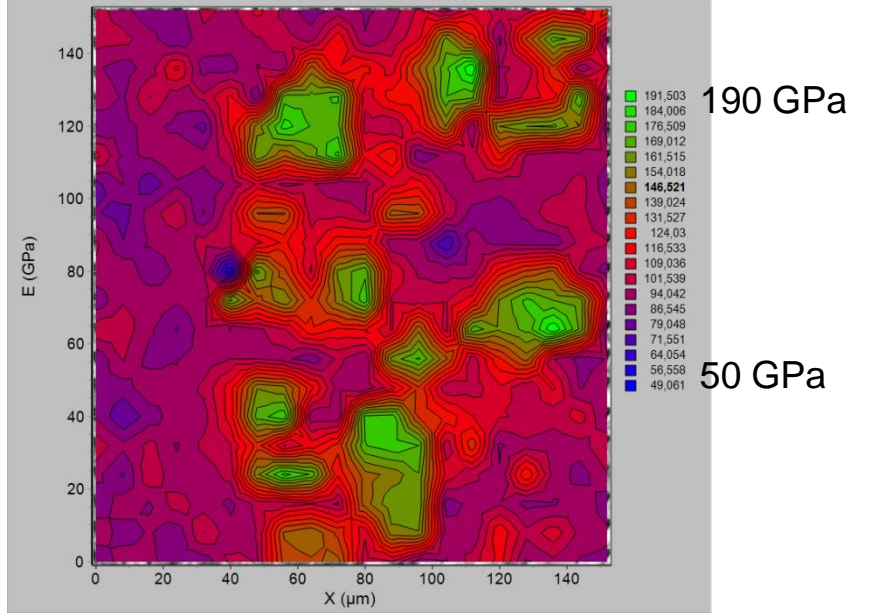
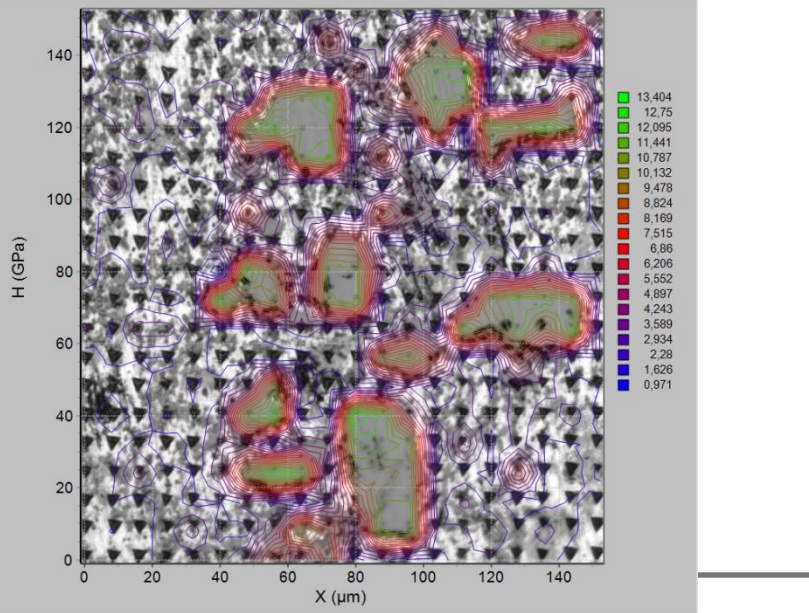
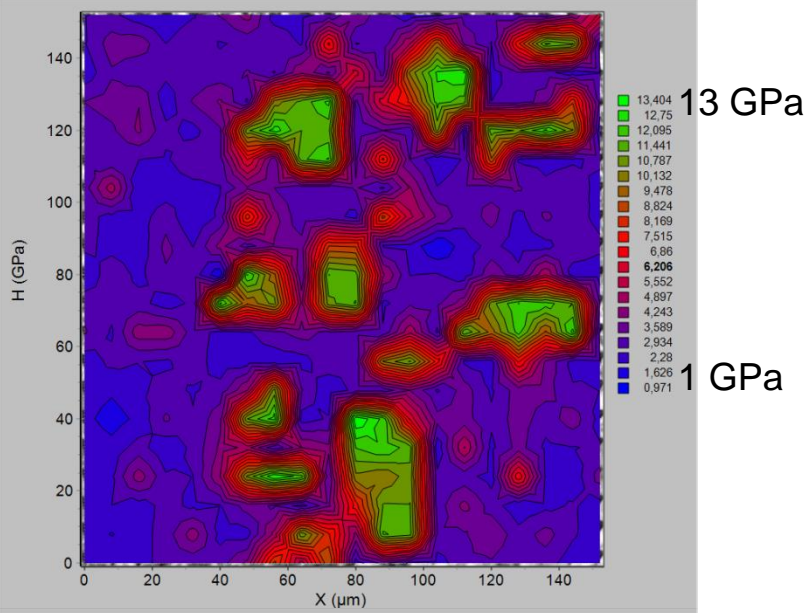




## Measurement of topography with green light interferometer

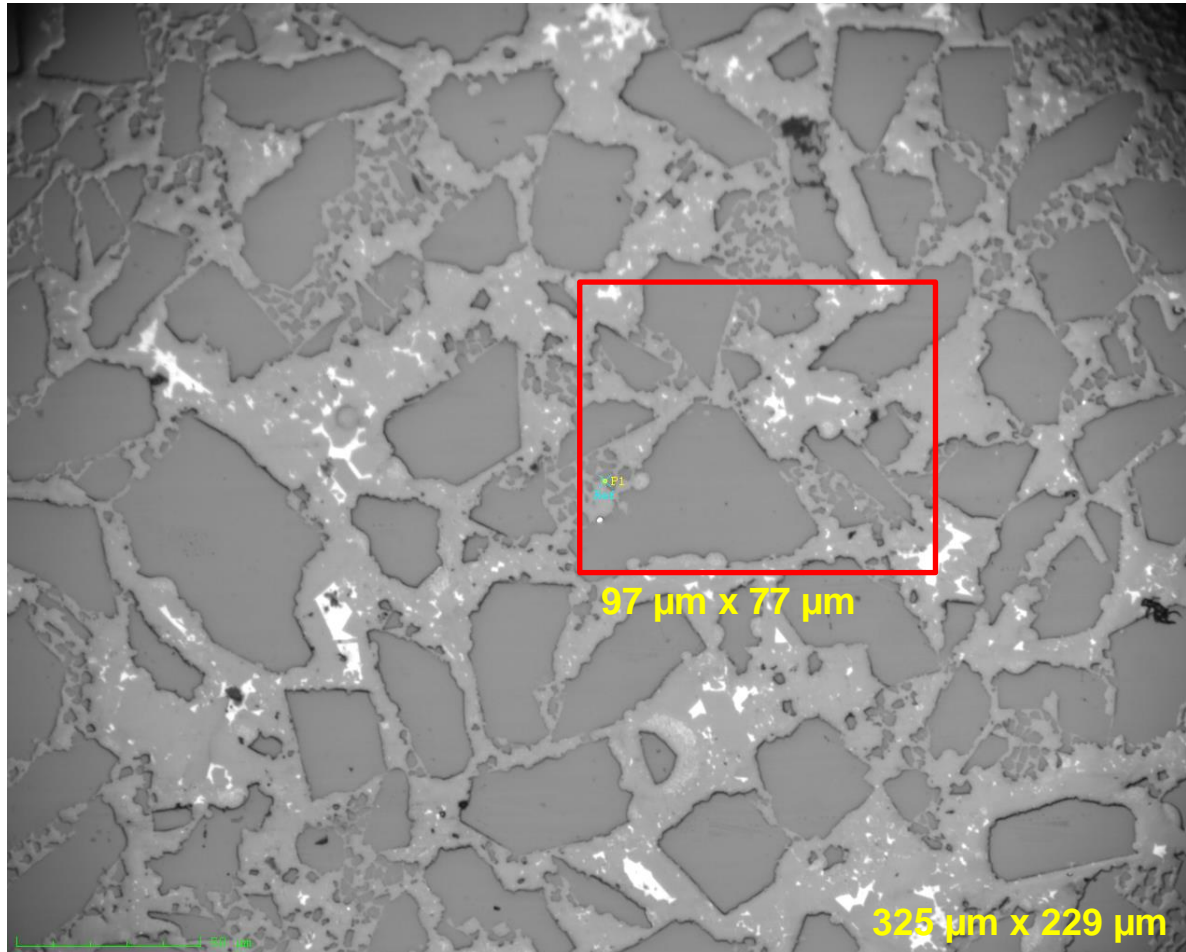
High surface roughness is making tests more difficult





## Example 2: Topography, modulus and friction mapping

Sample: Diamond - SiC composite



### Test parameters:

Indenter:

Sphere with 10,48 $\mu\text{m}$  radius

Contact diameter

on Diamond: 1,5  $\mu\text{m}$

on SiC: 1,8  $\mu\text{m}$

45 lines in a distance of 1.7 $\mu\text{m}$

Test time: 1600s

Scan time per line: 20s

Scan force: 30mN

Data rate 8 Hz

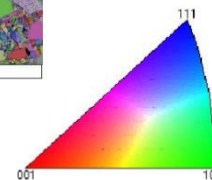
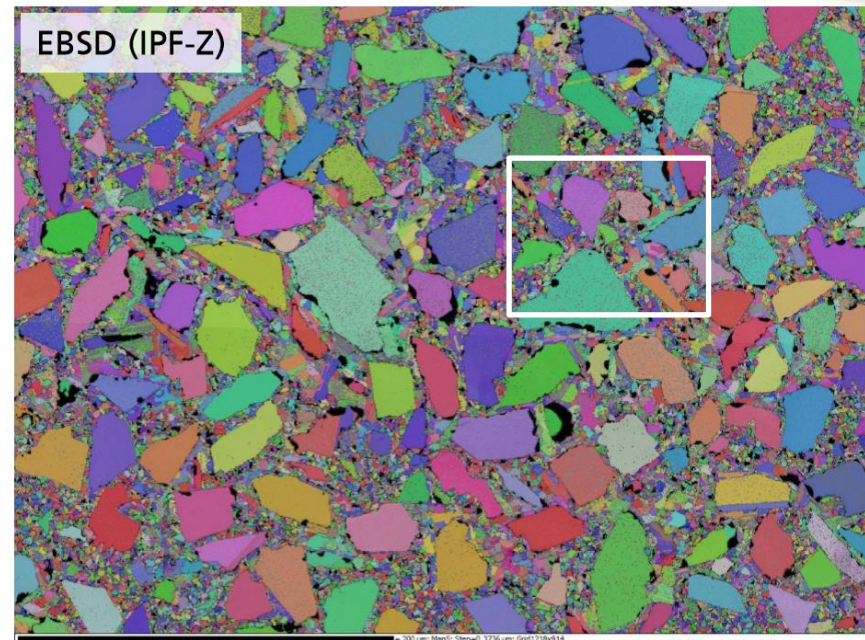
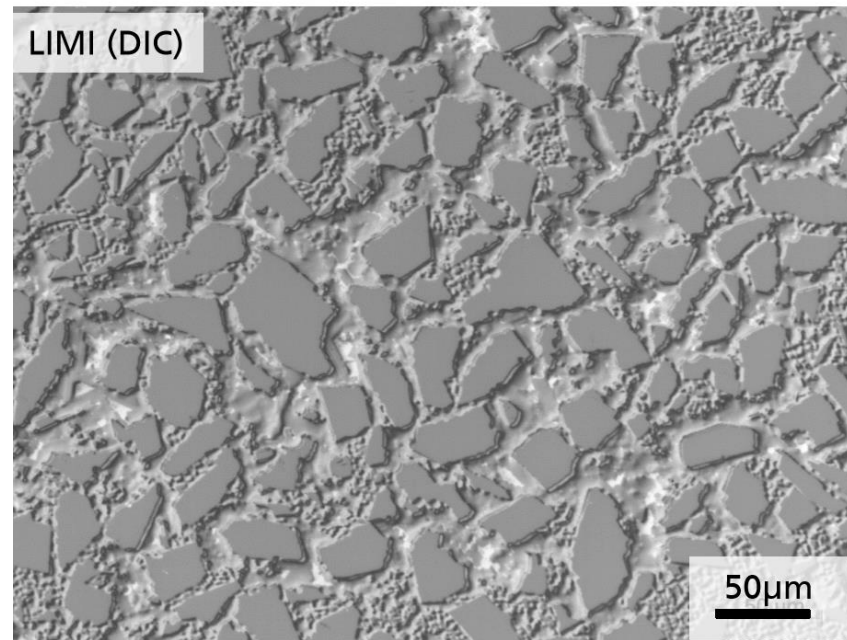
Oscillation frequency: 40 Hz

Amplitude: 0,22V  $\approx$  3nm

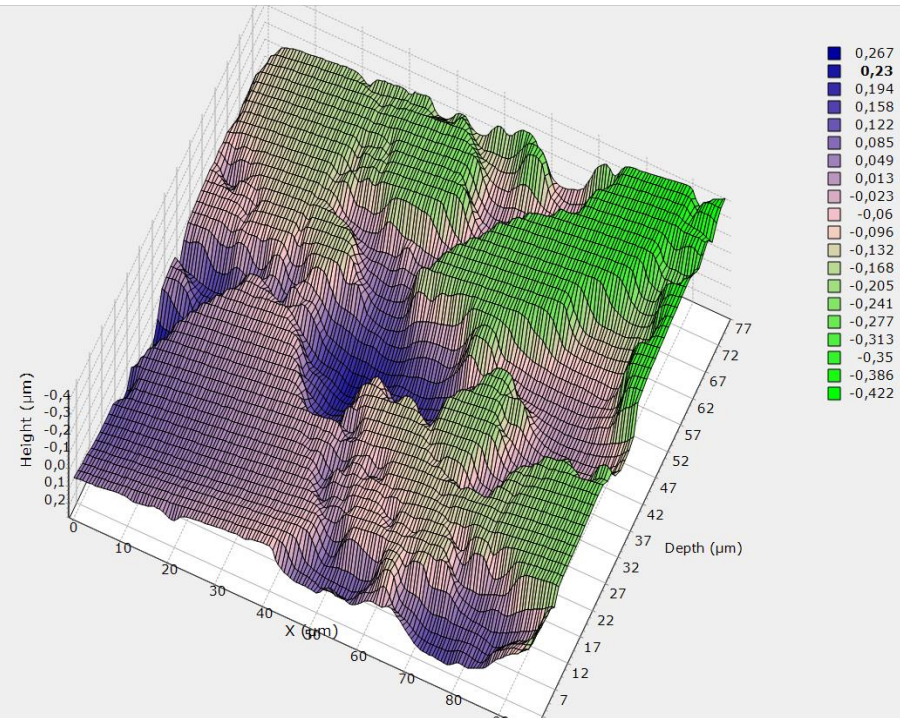
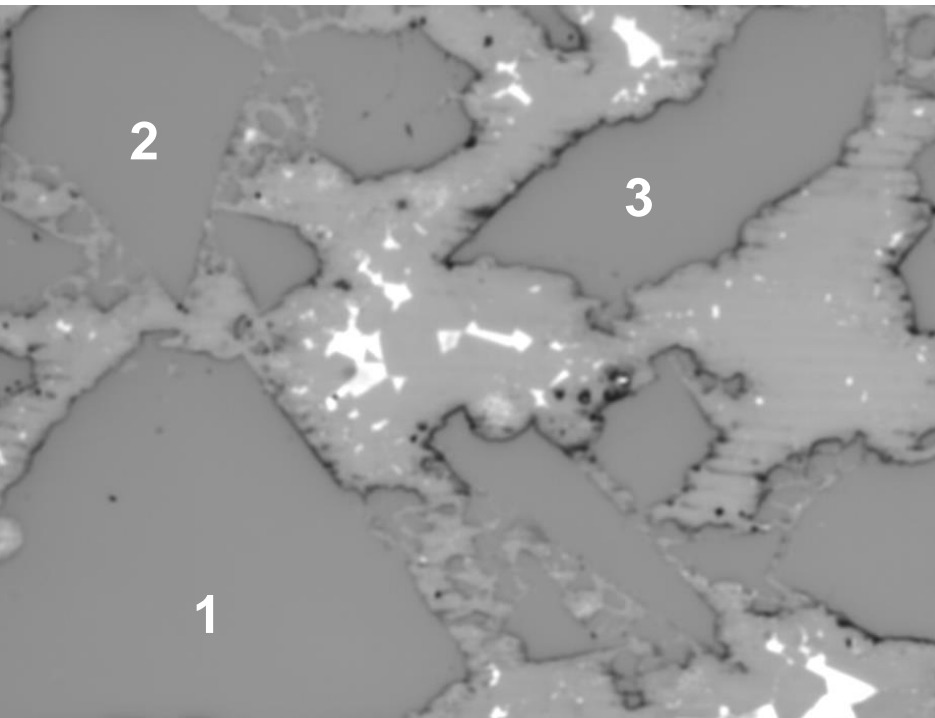
## EBSD measurements of diamond crystals done at Fraunhofer IKTS

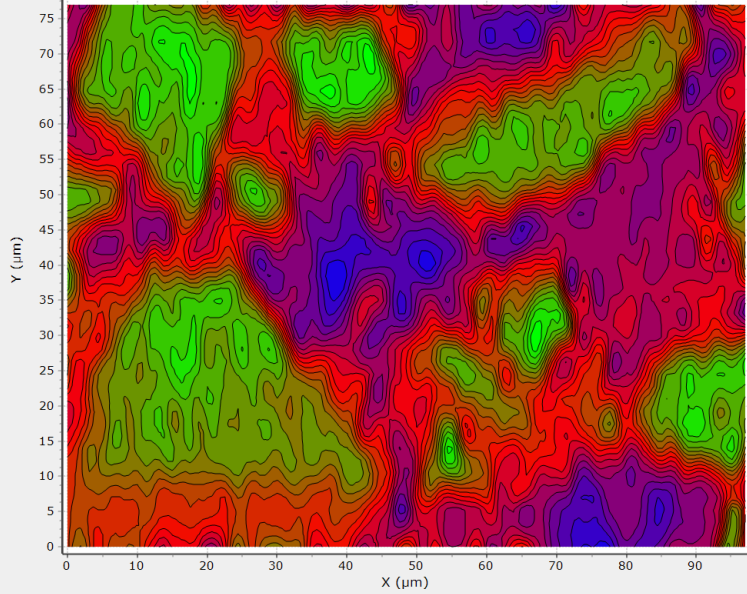
### Diamant - SiC

LIMI



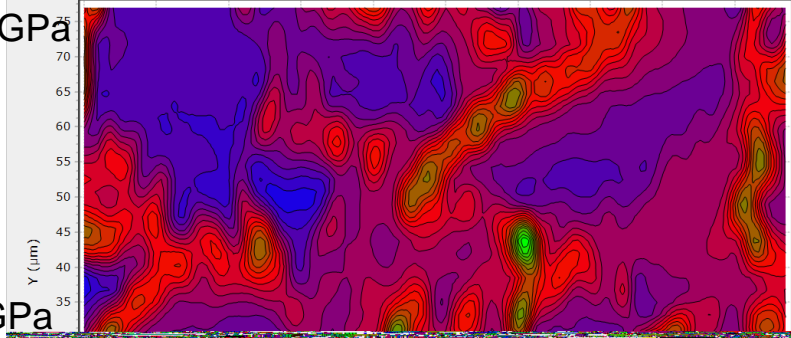
## Topography





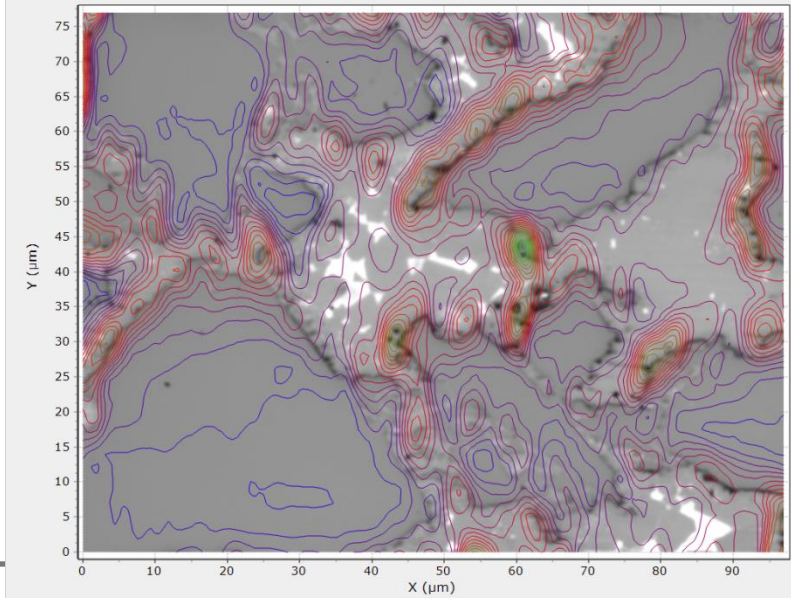
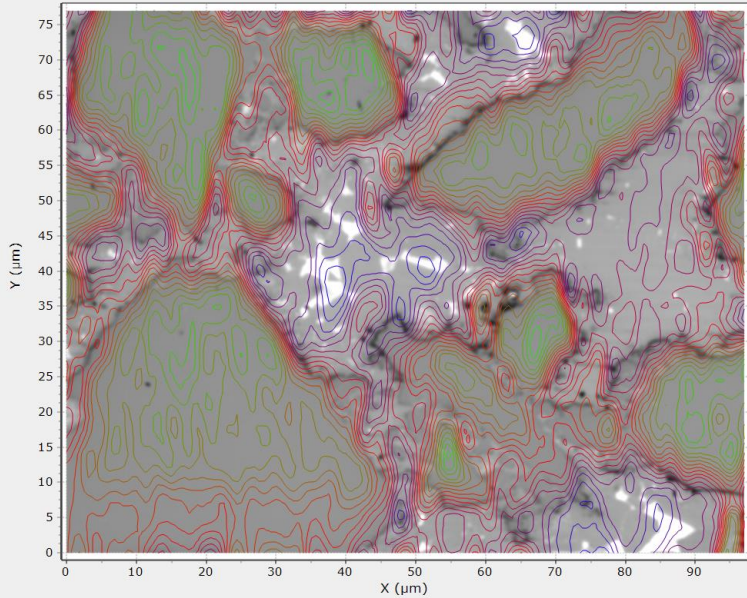
1132GPä

439GPä

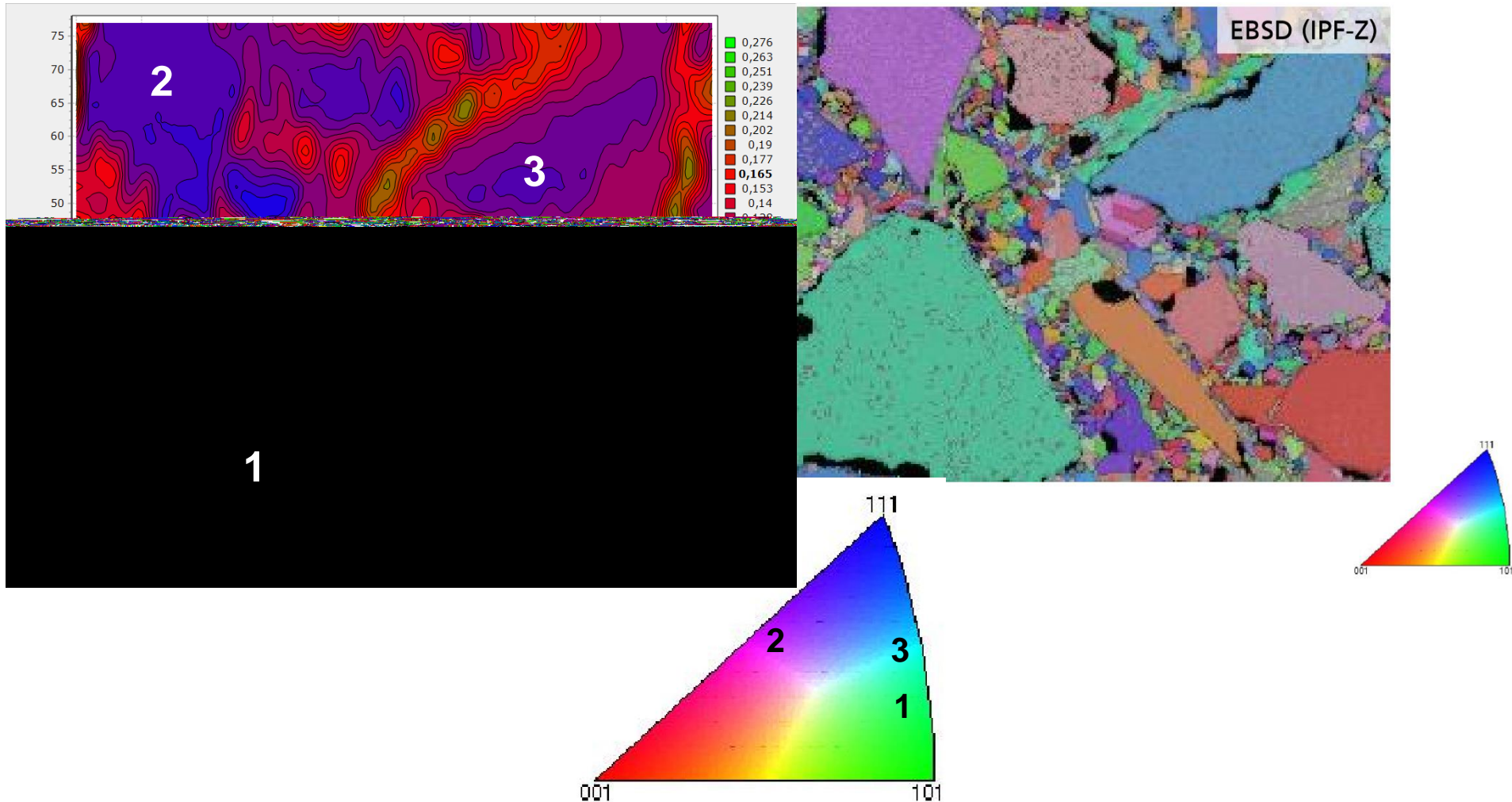


0,276

0,043



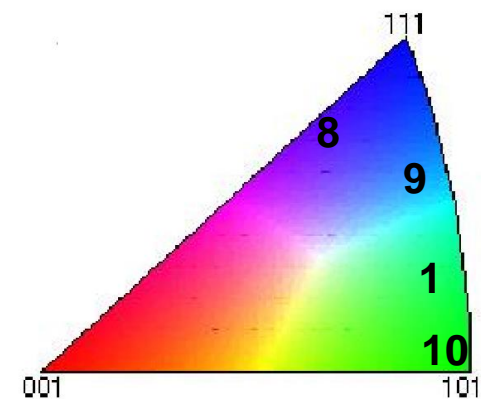
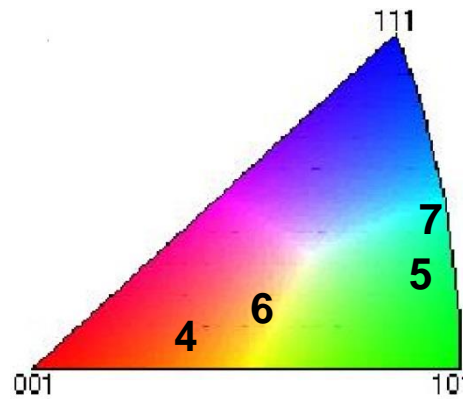
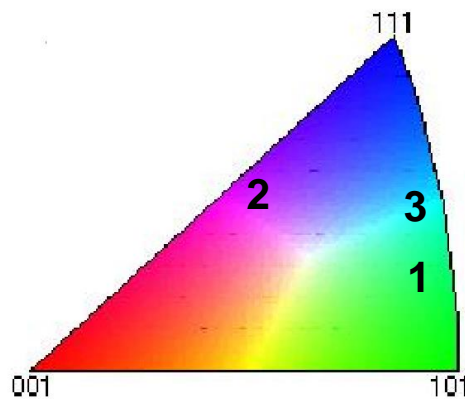
## Friction coefficient



## Friction coefficient in dependence on diamond crystal orientation

Measurement of 9 different grains

<b>1</b>	<b>2</b>	<b>3</b>	
<b>0,064</b>	<b>0,071</b>	<b>0,085</b>	
(101)-(111)	(001)-(111)	(101)-(111)	
<b>4</b>	<b>5</b>	<b>6</b>	<b>7</b>
<b>0,065</b>	<b>0,051</b>	<b>0,077</b>	<b>0,072</b>
(001)-(101)	(101)-(111)	(001)-(101)-(111)	(101)-(111)
<b>1</b>	<b>8</b>	<b>9</b>	<b>10</b>
<b>0,046</b>	<b>0,058</b>	<b>0,081</b>	<b>0,027</b>
(101)-(111)	(111)	(001)-(111)	(101)





## Conclusions

- Finding the mechanically best behaving coating-substrate combination for a certain application is time consuming and expensive.
- The quantification of the mechanical behavior is still a challenge. Not all necessary mechanical parameters of coatings can be measured with standard test methods up to now.
- For the understanding of failure mechanisms it is necessary to measure with nanometer resolution. A lot of different micro or nano-mechanical test methods are available now in one and the same instrument.
- A larger inclusion of modelling tools (FE, analytical) can considerably shorten the development process. For the calculations accurate and relevant mechanical parameters are necessary.
- A better reproduction of the conditions in an application is necessary in laboratory tests. This requires the inclusion of lateral force-displacement measurements with high precision
- In the future the significance of multi-axial testing in combination with stress calculation will increase.

## Thank you for your attention !

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